TOPICAL REVIEW — Topological 2D materials

Progress on 2D topological insulators and potential applications in electronic devices*

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(Received 22 May 2020; revised manuscript received 14 July 2020; accepted manuscript online 28 July 2020)

Two-dimensional topological insulators (2DTIs) have attracted increasing attention during the past few years. New 2DTIs with increasing larger spin–orbit coupling (SOC) gaps have been predicted by theoretical calculations and some of them have been synthesized experimentally. In this review, the 2DTIs, ranging from single element graphene-like materials to bi-elemental transition metal chalcogenides (TMDs) and to multi-elemental materials, with different thicknesses, structures, and phases, have been summarized and discussed. The topological properties (especially the quantum spin Hall effect and Dirac fermion feature) and potential applications have been summarized. This review also points out the challenge and opportunities for future 2DTI study, especially on the device applications based on the topological properties.

Keywords: two-dimensional materials, topological insulators, quantum spin Hall effect, dissipation-less devices, nanoelectronics

PACS: 73.63.-b, 73.43.Nq, 03.65.Vf, 85.35.-p

1. Introduction

Topological insulator is one of the most emergent research areas in condensed matter physics, especially for the novel physical phenomena. As a classic example, graphene was successfully exfoliated from graphite bulk to single-atomthickness layer in 2004^[1] and was predicted to be a twodimensional topological insulator (2DTI), dubbed as quantum spin Hall insulator (QSHI), by C. L. Kane and E. J. Mele in 2005.^[2,3] Since then, the exploration of 2DTI has gained increasing interest due to its scientific importance as a novel quantum state and related applications.^[4–6]

The 2DTIs hold several significant advantages due to the unique electronic structures. First, electronic structures of 2DTI have gaps in the bulk and gapless edge states (Fig. 1(a)). Such edge states have the spin-momentum locking property (Fig. 1(b)), which can suppress electronic backscattering against nonmagnetic impurity due to the time-reversal (TR) symmetry. Second, the edge states of 2DTI have the Dirac fermions feature, that is, a linear dispersion relationship in the energy space (Fig. 1(a)) which results in high mobility carriers.^[7,8] Compared with three-dimensional (3D) materials, the 2D materials have quantum confinement in the vertical direction, resulting a distinctly different band structure.^[9] Moreover, 2DTIs have significant advantages in the tunability of the energy band, valley, and spin properties as they can be easily manipulated by external fields like magnetic,^[10] electric,^[11] optic,^[12,13] and strain fields.^[14] Therefore, the

DOI: 10.1088/1674-1056/aba9c5

2DTI is very promising for dissipationless electronic devices, quantum computation, ultrafast response optoelectrical devices, *et al.*^[7,8]

The prediction of quantum spin Hall effect (QSHE) in graphene has inspired exploration of other 2DTI materials with stronger SOC strength. In 2006, B. A. Bernevig *et al.* theoretically predicted that QSHE can be achieved in HgTe/CdTe quantum well,^[15] which was soon realized experimently in 2007 by Molenkamp's group.^[16] Since then, there was a wave of scientific exploring research on 2DTI. Later, people quickly spread it to other quantum well systems theoretically^[17–19] and experimentally.^[20–22] The initial discovery and subsequent development of 2DTI in semiconductor quantum well systems have played a pivotal role, and have been discussed in previous papers.^[7–9]

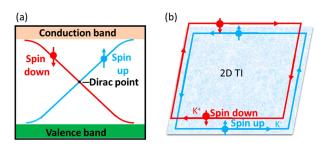


Fig. 1. (a) Band structure of 2DTI, which has the gap in the bulk denoted by the separation between conduction band and valence band, and gapless states at the edge denoted by the blue and red curves. The blue and red branches carry spin up and down currents, respectively. (b) Spin-momentum locking in the edge states of 2DTI in real space.

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^{*}Project supported by the Beijing Natural Science Foundation, China (Grant Nos. Z190006 and 4192054), the National Natural Science Foundation of China (Grant Nos. 61971035, 61901038, and 61725107), Strategic Priority Research Program of the Chinese Academy of Sciences (Grant No. XDB30000000), and Beijing Institute of Technology Research Fund Program for Young Scholars (Grant No. 3050011181814).

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This review will focus on the brilliant progress of 2DTI in the recent years, especially on the relationship between the geometric structures and the topological properties, as well as the related applications. In the end, we will give the perspective on the challenge and opportunities for future study on 2DTI.

2. Topological properties and application of novel 2D materials

During the last decade, new members of 2DTI with increasing spin–orbit coupling (SOC) gap have been predicted theoretically and prepared experimentally, such as silicene, germanene, bismuthene, Bi_4Br_4 , BiX/SbX (X = CI, Br, I), and 1T' transition metal chalcogenides (TMDs). The typical theoretically predicted values of SOC gaps are listed in Table 1. We further analyze the development of each of these materials, the related topological properties, and the potential applications.

Table 1.	The	values	of	SOC	gan	in	different	2DTL
Table I.	THU	values	O1	000	Sup		uniterent	2011.

2DTI	graphene	silicene	germanene	bismuthene	BiX/SbX	Bi_4Br_4	1T'-TMD
SOC gap/meV	$0.8 imes 10^{-3}$ [2]	1.55 ^[23]	23.9 ^[23]	470 ^[24]	1080 ^[25]	180 ^[26]	150 ^[27]

2.1. Silicene and germanene

Silicene, the silicon analogue of graphene, was proposed to host quantum spin Hall effect by Liu C. C. *et al.*^[28] in the year of 2011. Compared to graphene, silicene has a larger SOC gap due to the heavier atomic mass and the buckled geometry. According to the DFT calculations, silicene has a SOC gap of 1.55 meV (Fig. 2(a)) with helical gapless edge states. Although silicene has a buckled structure, the band

structure also has a massless Dirac fermion behavior with a Fermi velocity of $\sim 10^6$ m/s, as fast as that of graphene. Due to the compatibility with the current established siliconbased semiconductor industry, it seems that silicene can be a good material for field effect transistor (FET). However, the bandgap of silicene is still small (the corresponding QSH critical temperature is ~ 18 K) and may hinder its applications in the electronic devices. Various methods have been tested to induce a larger gap (a topological trivial band gap)

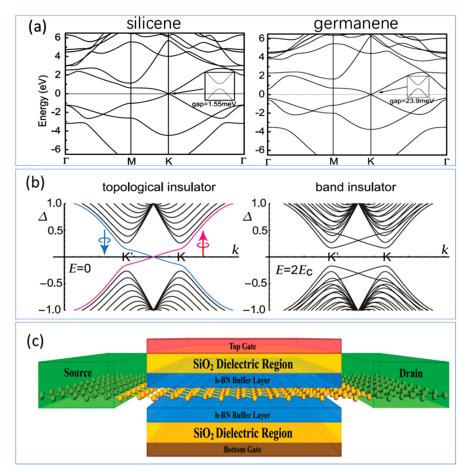


Fig. 2. (a) Calculated electronic band structures of buckled silicene and germanene. (b) Band structures of silicene or germanene under different vertical electric field *E*, which turns a topological insulator to a band insulator. (c) Schematic model of dual-gated silicene FET. (a) Reprinted with permission from Ref. [23]. Copyright 2011, American Physical Society. (b) Adopted with permission from Ref. [38]. (c) Reprinted with permission from Ref. [39]. Copyright 2012, American Chemical Society.

in silicene layer, such as adsorption or intercalation of metal atoms,^[29–31] hydrogenation,^[32,33] and oxidization.^[34] Similar to graphene nanoribbon, 1D silicene nanoribbon, made of silicene by cutting, has a tunable bandgap up to about 0.4 eV, which was predicted by DFT calculations^[5,35] and realized in experiments.^[36,37]

Apart from the above chemical or structural modification methods, which may affect the fast response of topological property-based devices, another method by applying a vertical electric field has been proposed.^[38] The vertical electric field can break the symmetry of the two sublattices of silicene and induce a topological phase transition between the QSH edge state (ON) and trivial insulator state (OFF), as shown in Fig. 2(b). Utilizing this feature, silicene could be a candidate for a new type of quantum mechanical switch, so-called topological insulator field effect transistor (TI-FET), as shown in Fig. 2(c).^[39] Thus silicene provides a flexible platform for low-power consumption and unconventional electronic devices based-on topological property.

In the periodic table of elements, another group-IVA element, germanium also is predicted to form a single layer stable structure, named as germanene. Germanene is also 2DTI with a low-buckled honeycomb structure similar to silicene, but a much larger spin–orbit gap of 23.9 meV^[23] was predicted (Fig. 2(a) right panel), corresponding to the QSH T_c of 276 K. Thus QSHE, as well as germanene-based TI-FETs, are expected to be observed experimentally.

Shortly after the theoretic prediction of silicene and germanene, there have been great progresses on the experimentally findings, both on characterization of geometric structures and electronic structures. The high-quality samples have been synthesized by molecular beam epitaxy (MBE) method on different substrates.^[40,46–48] For example, high-quality silicene with a $(\sqrt{3} \times \sqrt{3})$ silicene/ $(\sqrt{7} \times \sqrt{7})$ Ir(111) superstructure on Ir(111) (Fig. 3(a)),^[46] and silicene with both T and H phases on Ag(111) were independently reported (Fig. 3(c)).^[41] Besides, germanene with buckled configuration was reported on Pt(111) and Cu(111) substrates (Figs. 3(b) and 3(f)).^[40,43] Calculated electron localization function (ELF) showed the covalent interaction existing between adjacent germanium atoms, while the electrostatic interaction between germanium atom and substrate, thus supporting the 2D nature of the germanene lattice (Fig. 3(e)).

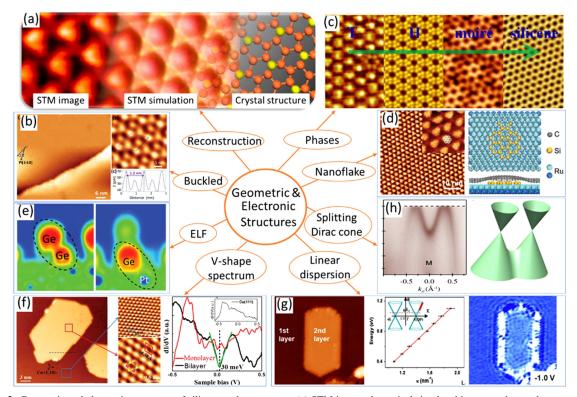


Fig. 3. Geometric and electronic structures of silicene and germanene. (a) STM image, theoretical simulated image, and crystal structure of silicene on Ir(111) from left to right. (b) STM image of buckled germanene on Pt(111). (c) T and H phases of silicene on Ag(111) substrate. (d) STM images of silicene nanoflake intercalated between graphene and Ru(0001). (e) Calculated electronic localization function (ELF), showing the covalent interaction existing between adjacent germanium atoms and electrostatic interaction between germanium atom and substrate. (f) *dI/dV* on monolayer (red) and bilayer (black) germanene. (g) *dI/dV* mapping on the 2nd layer silicene island which can derive the linear dispersion relationship of Dirac fermion. (h) Splitting of Dirac cone observed by angle resolved photoemission (ARPES) on 3×3 -silicene on Ag(111). (a) Reprinted with permission from Ref. [40]. Copyright 2014, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim. (c) Reprinted with permission from Ref. [41]. Copyright 2012, American Chemical Society. (d) Reprinted with permission from Ref. [42]. Copyright 2018, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim. (f) Reprinted with permission from Ref. [43]. Copyright 2017, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim. (g) Reprinted with permission from Ref. [44]. Copyright 2012, American Physical Society. (h) Reprinted with permission from Ref. [45]. Copyright 2019, American Physical Society. (h) Reprinted with permission from Ref. [45]. Copyright 2019, American Physical Society. (h) Reprinted with permission from Ref. [45]. Copyright 2019, American Physical Society.

Because of the easy oxidation of silicene in air, usage of silicene for device application becomes a challenge. To solve this problem, Gao's group intercalated silicon between graphene and Ru(0001) to protect silicene layer by covering of graphene layer. Depending on the silicon coverage, silicene nanoflake, single layer and multilayer layer silicene were prepared underneath graphene, as shown in Fig. 3(d).^[42] As for device, silicene-based FET was reported in 2015, which demonstrated theoretical expectations of its ambipolar Dirac charge transport and showed a high carrier mobility of ~ $100 \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$ at room-temperature.^[49]

Regarding the electronic structures, studies mainly focus on the linear dispersion feature of Dirac fermions. For example, in 2017, high-quality bilayer germanene islands on Cu(111) were reported, which revealed a V-shape dI/dV spectrum, indicating a signature of Dirac-type electrons (Fig. 3(f)).^[43] By dI/dV mapping on bilayer silicene islands at different bias voltages around the Fermi level, Wu's group has successfully described the linear dispersion relationship of the edge states, which is a hallmark of Dirac fermions.^[44] More recently, they reported a new physical phenomenon, the splitting of Dirac cones measured by ARPES technique, in the system of 3×3 -silicene on Ag(111), ascribed to external periodic potentials from the substrate–overlayer interaction (Figs. 3(g) and 3(h)).^[45]

2.2. Antimonene

Antimonene, a 2D honeycomb lattice made of antimony atoms, is another attractive mono-elemental candidate for 2D topological material.^[53–55] Two structures of antimonene have been found: buckled honeycomb and flat one (Fig. 4(a)), which were successfully fabricated on PdTe₂ and Ag(111) substrates by MBE, respectively.^[50,51] Figures 4(b) and 4(c) show atomic-resolution STM images of buckled antimonene monolayer (BAM) and flat antimonene monolayer (FAM). The calculated electron localization function of Sb–Sb pair is higher than that between Sb and the substrate, which proves

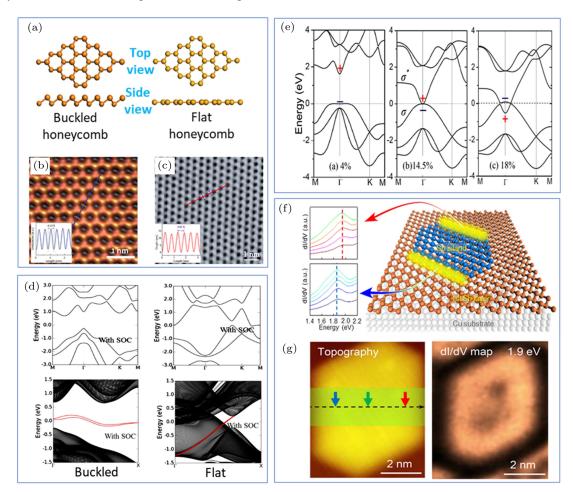


Fig. 4. Geometric and electronic structures of antimony monolayer. (a) Atomic structures of the buckled and flat honeycombs of antimony. (b), (c) Atomic resolution STM image of the buckled antimonene monolayer (BAM) and flat antimonene monolayer (FAM). Inset: line profile corresponding to the line in panel. (d) Band structures and the edge states of antimony monolayer. FAM has a topologically nontrivial edge state and BAM has a topologically trivial edge state. Red lines denote the edge states. (e) Inverted band of antimony monolayer induced by increasing strain field. (f) Edge states of antimonene island on top of ML antimonene. (g) STM image and *dl/dV* mapping of the antimonene island. (a), (c), (d) Reprinted with permission from Ref. [50]. Copyright 2018, American Chemical Society. (b) Reprinted with permission from Ref. [51]. Copyright 2017, WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim. (e) Reprinted with permission from Ref. [14]. Copyright 2015, Nature Publishing Group. (f), (g) Reprinted with permission from Ref. [52]. Copyright 2019, American Chemical Society.

that antimonene of both structures are in 2D nature with minor interaction with the substrate. Band calculations show that FAM is a topological semimetal with topologically non-trivial edge state, while BAM has a topologically trivial edge state (Fig. 4(d)). A theoretical work reported that the bulk bandgap of buckled antimonene can be tuned under different tensile, and above tensile of 14.5%, the bands are inverted, creating a topological gap (Fig. 4(e)).^[14] Thus, the flat and buckling structures provide a feasible way to regulate the electronic properties of antimonene and extend related nanodevice applications. Very recently, the antimonene nanoislands in the second layer with topologically nontrivial one-dimensional edge states were reported (Figs. 4(f) and 4(g)).^[52] This work not only provides hallmark to topologically nontrivial states in the BAM but also opens a novel route for tailoring the topological edge states by external strain of the supporting substrate.

2.3. Bismuthene, Bi_4Br_4 , and BiX/SbX (X = H, F, Cl, Br)

Following the trend of searching for large SOC 2DTI, monolayer or few-layer materials with the high-Z element bismuth (Z = 83) are explored.^[56–58] In 2014, Yazdani's group reported one-dimensional topological edge states of bismuth bilayers on top of bulk bismuth.^[59] Importantly, one type of edge showed spectra with a density of states that has inversesquare-root singularity inherent to a 1D state. Moreover, by visualizing the quantum interference of edge-mode quasiparticles in the confined geometries, they demonstrated the coherent propagation along the edge with properties consistent with strong suppression of backscattering. In 2017, Reis *et al.*^[60] reported the bismuthene on SiC as a candidate for a high temperature quantum spin Hall material. Figures 5(a) and 5(b) present the geometric structure and band structure of bismuthene on SiC, respectively. The line dI/dV across the step edge reveals a large bulk gap of ~ 0.8 eV with local metallic edge state (Fig. 5(c)). As the bismuthene was prepared on a semiconductor SiC substrate, this work pushed forward the future application for realizing various device proposals with topological edge modes.

Besides the 2D TI materials made of single element, the search for 2D TI with a larger gap is also expanded to bielement materials like BiX/SbX (X = H, F, Cl, and Br) and Bi₄Br₄. Because Bi and Sb are well known for their strong SOC that can drive and stabilize the topological states, it is intuitive to search for large-band-gap QSH insulators based on Bi/Sb-related materials. In 2014, Yao's group predicted a group of 2DTI BiX/SbX (X = H, F, Cl, and Br) monolayers with extraordinarily large bulk gaps based on firstprinciples calculations^[61] and tight-binding calculations,^[25] respectively. Figures 4(d) and 4(e) show the unit cell and the typical band structures of BiH and BiF. 2D BiX has a honeycomb structure like silicon hydride (silane), with a threefold rotational symmetry like that in graphene. The side view exhibits the nearest neighbor Bi-X distances and buckling heights, a low-buckled configuration is more stable for the BiF, BiCl, and BiBr monolayers.

Interestingly, the giant-gaps are due to the strong SOC related to the p_x and p_y orbitals of the Bi/Sb atoms, in contrast

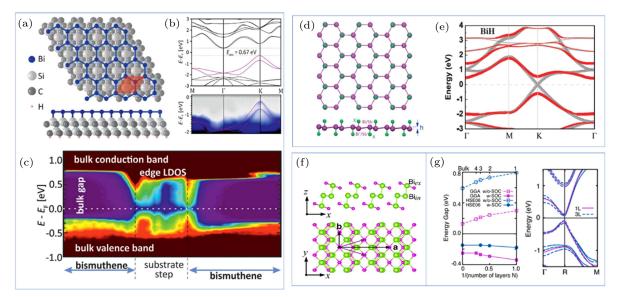


Fig. 5. Geometric and electronic structures of bismuthene and BiX layers. (a) Sketch of a bismuthene layer placed on the threefold-symmetric SiC(0001) substrate. (b) DFT calculated electronic band structures and ARPES experimental electronic band dispersion of single layer bismuthene. (c) Spatially resolved dI/dV data across the step, the dI/dV signal of the in-gap states/peaks at the edges of bismuthene film. (d) Atomic structure of monolayer BiX/SbX (X = H, F, Cl, Br), by top view and side view. (e) Electronic band structure of BiH and BiF with large gap. Red line means SOC is considered, but the gray line is not. (f) Side view and top view of the crystal structure of single-layer Bi₄Br₄. (g) The edge state and spin polarization of different layers of Bi₄Br₄. The layer dependence of energy gap of few-layer Bi₄Br₄ (left panel), and band structures of single-layer Bi₄Br₄ (right panel). (a)–(c) Reprinted with permission from Ref. [60]. Copyright 2017, American Association for the Advancement of Science. (d), (e) Reprinted with permission from Ref. [61]. Copyright 2014, Nature Publishing Group. (f), (g) Reprinted with permission from Ref. [62].

to p_z orbital in graphene and silicene. The topological characteristic of BiX/SbX monolayers is confirmed by the calculated nontrivial Z_2 index. The bulk band gap of greater than 1.0 eV in BiH and BiF monolayers is probably the largest bulk band gap of all the TIs predicted theoretically. Moreover, BiX/SbX monolayers in vertical electrical field become a quantum valley Hall insulator, exhibiting valley-selective circular dichroism.

While many 2DTIs with large SOC gap have been proposed and realized in the experiment, the lack of appropriate insulating substrates is another crucial issue. Single-layer Bi_4Br_4 may break this obstruction because its bulk crystal is an insulator.^[62] Single-layer Bi_4Br_4 can be obtained by mechanical exfoliated from its bulk material (see Fig. 5(f) for the layered crystal structure), and was recently demonstrated to be a QSH insulator with a sizable gap (0.18 eV).^[26] The energy gap and band structures of few-layer Bi_4Br_4 do not change much with increasing layers (Fig. 5(g)). Furthermore, at the boundary of multilayer Bi_4Br_4 , the topological edge states stemming from different single-layers are weakly coupled, and can be fully decoupled by constructing a stair-stepped edge. Thus the decoupled topological edge states are very suitable for multi-

channel dissipationless transport.

2.4. 1T'-TMD materials

The 2D layered transition metal chalcogenides (TMDs, MX_2) (M = transitionmetal, X = chalcogen) have received increasing attention in recent years. As shown by geometric schematics in Fig. 6(a), TMDs have a variety of phases such as 2H, 1T, 1T'. Among them, 2H and 1T phases have been studied in 2D semiconductor systems such as 2H-MoS₂ and 1T-PtSe₂. Recently, the 1T' phase is gaining increasing attention due to the type-II Weyl semimetal in the bulk WTe₂. While the properties of ML $1T'-MX_2$ have not been unveiled until in 2014, Qian *et al.*^[63] studied $1T'-MX_2$ (with M = tungsten or molybdenum and X = tellurium, selenium, or sulfur), and predicted that it changed from a type-II Weyl semimetal in the bulk form to a class of large-gap QSH materials in the monolayer form. The structural distortion from 1T to 1T' doubles the lateral period and causes an intrinsic band inversion, and SOC interaction further opens a gap of 0.08 eV, rendering the topological properties. A vertical electric field breaks the inversion symmetry and introduces a strong Rashba splitting of the doubly degenerate bands near the fundamental gap E_{g} .

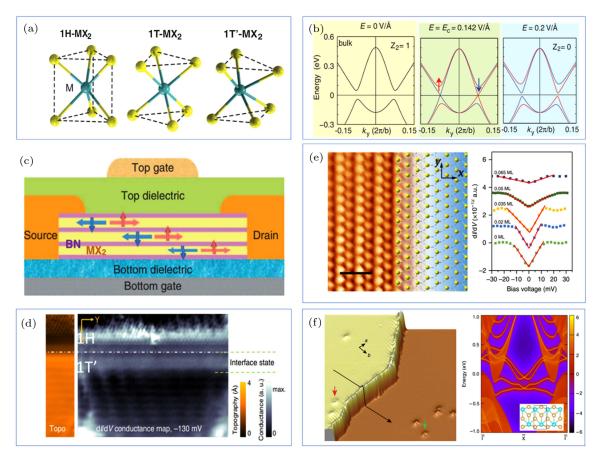


Fig. 6. Geometric and electronic structures of TMDs. (a) Three typical atomic configurations of transition metal dichalcogenides (TMDs, MX_2). (b) Band inversion of $1T'-MX_2$ induced by vertical electric fields. (c) A vdW-FET by stacking several layers of $1T'-MX_2$ and 2D dielectric layers (h-BN). (d) Experimental dI/dV map taken at the 1T'-1H interface. (e) Atomic resolution STM image of the single-layer $1T'-WTe_2$ surface and dI/dV fitted with a linear function near the Fermi energy. (f) Pseudo 3D image of WTe₂ showing a step edge and the calculated topological edge state of the monolayer $1T'-WTe_2$ bulk electronic structure. (a)–(c) Reprinted with permission from Ref. [63]. Copyright 2014, American Association form Ref. [65]. Copyright 2018, Nature Publishing Group. (f) Reprinted with permission from Ref. [66]. Copyright 2017, Nature Publishing Group.

As the field increases, E_g first decreases to zero at a critical field strength of 0.142 V/Å and then reopens (Fig. 6(b)). This gap-closing transition induces a topology change to a trivial phase. Similar to Bi₄Br₄, the 2D 1T'-*MX*₂ have shown possibility in constructing van-der Waals devices like topological FET (T-FET) by stacking multilayer together to increase the conducting channels (Fig. 6(c)).

This theoretic prediction has inspired a wave of experimental study on the preparation and topological study on the monolayer $1T'-MX_2$. In 2017, Crommie's group^[67] reported the study of electronic structure of the monolayer 1T'-WTe₂ on BLG/SiC(0001) by a combination of different techniques like ARPES, DFT, and STM. They observed topological band inversion, band gap opening, and the edge states that were consistent with expectations for a QSH insulator. In addition, this group found similar QSH behavior in the system of ML 1T'-WSe₂ (Fig. 6(d)).^[64] Using the 1D interfaces between the topologically non-trivial 1T' and 1H phases, they confirmed the predicted penetration depth (2 nm) of the one-dimensional interface states into the 2D bulk of a quantum spin Hall insulator (QSHI). Li et al.^[65] used high-resolution quasiparticle interference (OPI)-STS/STM to study the monolayer 1T'-WTe₂ also on BLG/SiC(0001), and found that it had a semimetal bulk band structure without a full SOC-induced band gap. Importantly, there is a Coulomb gap induced by electron-electron interactions at $E_{\rm F}$ (Fig. 6(e)). Not only monolayer WTe₂ but also the bulk step edges can exhibit the QSHE. In 2018, Peng et al.^[66] reported 1D step edge state of WTe₂ observed by STM (Fig. 6(f)). Importantly, the edge state shows significant robustness against edge imperfections and is unaffected by the presence of the substrate, which provides evidence for the topological origin.

2.5. Magnetic topological insulators

The 2D topological insulators were originally proposed in time-reversal invariant systems, however, the interplay between magnetic order and nontrivial topology can provide fertile ground to explore new physics and applications such as spintronics with low energy consumption, dissipationless topological electronics, and topological quantum computation. Recently, van der Waals layered AB_2C_4 materials (such as MnBi₂Te₄ and SnSb₂Te₄) have been predicted to host intriguing topological properties. For example, in 2019, Yong Xu's group^[68] predicted MnBi₂Te₄ to show intralayer ferromagnetic and interlayer antiferromagnetic, as well as rich topological quantum states such as a type II magnetic Weyl semimetal, intrinsic axion insulators, and QAH insulators in even- and odd-layer films, respectively. Using ARPES, Hong Ding's group^[69] and E. V. Chulkov's group^[70] independently reported the observation of surface state in MnBi₂Te₄, respectively, and provided clear evidence for nontrivial topology of the antiferromagnetic TIs. Very recently, various of exciting properties such as quantum anomalous Hall effect,^[71] robust axion insulator, and Chern insulator phases^[72] were observed in MnBi₂Te₄ thin flakes by method of quantum transport. These MnBi₂Te₄ family materials can be expected to integrate with other 2D materials to form van der Waals heterostructures and provide great opportunities for exploring exotic topological phenomena.

2.6. Quasi-2D topological materials and the optoelectronic coupling

While the optoelectronic coupling of 2DTI utilizing the topological properties has been limited, there are many examples of 3DTIs which have been explored for this purpose. For example, in 2009, Shen's group reported the investigation of the surface state of Bi2Te3 with angle-resolved photoemission spectroscopy, proving its 3DTI nature with a single Dirac cone on the surface (Fig. 7(a)). The large bulk gap of Bi₂Te₃ is promising for potential applications in high-temperature spintronics. Hajlaoui et al.^[74] used time-resolved ARPES to study the transient electronic dynamics of Bi2Te3, and provided a direct real-time visualization of the transient carrier population of both the surface states and the bulk conduction band (Fig. 7(b)). Mclver *et al.*^[77] found that by illuminating Bi₂Se₃ with circularly polarized light, a helicity-dependent photocurrent which originated from topological Dirac fermions can be generated. Moreover, a photocurrent which is controlled by the linear polarization of light may also have a topological surface state origin. Bi2Se3 and Bi2Te3 in these devices are layered nanoscale materials and can be regarded as stacked 2D materials, thus we call them quasi-2D materials. The opto-electronic coupling of 3DTI also extendeds to 3D Weyl semimetal. Weyl semimetals are transformed from Dirac materials by breaking the space inversion or time reversal symmetry. The break of symmetry splits the degeneracy of Dirac point into two Weyl points with opposite chiralities, the electronic excitations of which can be described by the Weyl equation.^[78,79] In 2018, Sun's group reported the realization of a broadband self-powered photodetector based on TaIrTe₄ (Fig. 7(c)). Compared with graphene and Cd₃As₂, TaIrTe₄ improved the response time from ps to µs order. In 2019, Sun's group^[80] further reported the realization of nonlinear optical correspondence on the type-II Weyl semimetal TaIrTe₄ sheet. Figure 7(d) illustrates the mechanism of generation of photocurrent in Weyl systems. 3D Weyl semimetals can generically support significant photocurrents due to the inversion symmetry breaking and also finite tilts of the Weyl spectra for type-II Weyl semimetal. Compared with the traditional photocurrent based on PN junction, 3D Weyl semimetals need neither applied electric field nor control of the depletion layer interface, and have a faster response on fs order, which will be very suitable for the next generation of ultrafast response photoelectric devices.

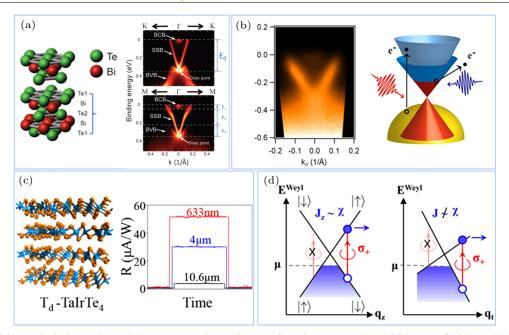


Fig. 7. Quasi-2D topological materials and the optoelectronic coupling. (a) Crystal structure and ARPES images of Bi_2Te_3 . (b) ARPES image and schematic of infrared laser pump excited electrons from the valence band (orange) into the conduction band of Bi_2Te_3 . (c) Crystal structure of the type-II WSM TaIrTe₄ and broadband photoresponse of TaIrTe₄ photodetector. (d) Schematics of photocurrent generations in type I (left panel) and type II (right panel) Weyl systems. (a) Reprinted from Ref. [73]. Copyright 2009, American Association for the Advancement of Science. (b) Reprinted with permission from Ref. [74]. Copyright 2012, American Chemical Society. (c) Reprinted with permission from Ref. [75]. Copyright 2018, American Chemical Society. (d) Reprinted with permission from Ref. [76]. Copyright 2017, American Physical Society.

3. Summary and perspectives

As mentioned above, there have been enormous progress on the study of 2DTI, from discovering of new members (including new phases) with increasing larger SOC gaps to exploring novel properties. For the application using topological properties, such as QSH effect and the Dirac fermion feature, many obstacles have been removed. For example, the largest SOC gap is over 1 eV, corresponding to a temperature much higher than room temperature so that application based on the QSH effect is possible. 2DTI can be prepared not only on metal substrates, but also on semiconductors and insulators, thus the interaction with substrates can be tuned and a back gate can be applied for the device purpose. With van-der Waals layered materials like ML Bi₄Br₄ and 1T' TMD, the number of conducting channels $(e^2/h \text{ per edge}, \text{ where } e \text{ is the elemen-}$ tary charge and h is Planck's constant) can be increased by stacking several ML materials together, resulting in a larger signal-to-noise ratio.

However, there are still some challenges and open questions for future developments of 2DTI. (a) The intrinsic topological characteristics of 2D systems need to be further understood. (b) More 2D topological systems combining several advantages (with high topological critical temperatures, on different substrates, and can be stacked to increase conducting channels) still need to be discovered. (c) Efficient methods of fast on/off switching are still lacking. Device preparation to tune the 2DTI with magnetic, electric, optic, and strain fields needs to be further explored. (d) Although a lot of theoretical papers predicting the application of 2DTI, till now the practical realization of 2DTI has been limited. A lot of the 2DTIs are made into devices utilizing the semiconducting or valley properties, instead of the topological properties. Meanwhile, the applications of topological systems mainly focus on the 3D systems (for example, 3DTI Bi₁₄Rh₃I₉ which hosts the helical edge states and offers the opportunity to design spin filters^[81]), while the 2D system is much fewer. Although these materials are 3DTIs, the extension to 2DTI can be expected. Thus, there is a great future to expand the area of 2DTI and make devices based on the topological properties into practice.

References

- Novoselov K S, Geim A K, Morozov S V, Jiang D, Zhang Y, Dubonos S V, Grigorieva I V and Firsov A A 2004 Science 306 666
- [2] Kane C L and Mele E J 2005 Phys. Rev. Lett. 95 226801
- [3] Kane C L and Mele E J 2005 *Phys. Rev. Lett.* **95** 146802
- [4] Guzman-Verri G G and Voon L L Y 2007 Phys. Rev. B 76 075131
- [5] Cahangirov S, Topsakal M, Akturk E, Sahin H and Ciraci S 2009 Phys. Rev. Lett. 102 236804
- [6] Lebegue S and Eriksson O 2009 Phys. Rev. B 79 115409
- [7] Hasan M Z and Kane C L 2010 Rev. Mod. Phys. 82 3045
- [8] Qi X L and Zhang S C 2011 Rev. Mod. Phys. 83 1057
- [9] Bansil A, Lin H and Das T 2016 Rev. Mod. Phys. 88 021004
- [10] Filatova T G, Gurin P V, Kloo L, Kulbachinskii V A, Kuznetsov A N, Kytin V G, Lindsjo M and Popovkin B A 2007 J. Solid State Chem. 180 1103
- [11] Li J and Chang K 2009 Appl. Phys. Lett. 95 222110
- [12] Mouri S, Miyauchi Y and Matsuda K 2013 Nano Lett. 13 5944
- [13] de Juan F, Grushin A G, Morimoto T and Moore J E 2017 Nat. Commun. 8 1
 - [14] Zhao M, Zhang X and Li L 2015 Sci. Rep. 5 16108
 - [15] B Andrei Bernevig, Hughes L and Zhang S C 2006 Science 314 1757
 - [16] Markus König, Steffen Wiedmann, Christoph Brüne, Andreas Roth, Hartmut Buhmann, Molenkamp W, Xiao Liang Qi and Zhang S C 2007 *Science* 318 766
 - [17] Liu C, Hughes, L T, Qi X L, Wang K and Zhang S C 2008 Phys. Rev. Lett. 100 236601

- [18] Miao M S, Yan Q, Van de Walle C G, Lou W K, Li L L and Chang K 2012 Phys. Rev. Lett. 109 186803
- [19] Zhang D, Lou W, Miao M, Zhang S and Chang K 2013 *Phys. Rev. Lett.* 111 156402
- [20] Knez I, Du R R and Sullivan G 2011 Phys. Rev. Lett. 107 136603
- [21] Pan W, Dimakis E, Wang G T, Moustakas T D and Tsui D C 2014 Appl. Phys. Lett. 105 213503
- [22] Zhou L, Dimakis E, Hathwar R, Toshihiro Aoki, Smith J, Moustakas T D, Goodnick S M and McCartney M R 2013 *Phys. Rev. B* 88 125310
- [23] Liu C C, Feng W X and Yao Y G 2011 Phys. Rev. Lett. 107 076802
- [24] Guo Y, Pan F, Ye M, Sun X T, Wang Y Y, Li J Z, Zhang X Y, Zhang H, Pan Y Y, Song Z G, Yang J B and Lu J 2017 ACS Appl. Mater. Interfaces 9 23128
- [25] Liu C C, Guan S, Song Z, Yang S A, Yang J B and Yao Y G 2014 *Phys. Rev. B* 90 085431
- [26] Zhou J J, Feng W X, Liu C C, Guan S and Yao Y G 2014 Nano Lett. 14 4767
- [27] Putungan D B, Lin S H and Kuo J L 2015 Phys. Chem. Chem. Phys. 17 21702
- [28] Liu C C, Jiang H and Yao Y G 2011 Phys. Rev. B 84 195430
- [29] Quhe R, Fei R X, Liu Q H, Zheng J X, Li H, Xu C Y, Ni Z Y, Wang Y Y, Yu D P and Gao Z X 2012 *Sci. Rep.* 2 853
- [30] Ni Z Y, Zhong H X, Jiang X H, Quhe R, Luo G F, Wang Y Y, Ye M, Yang J B, Shi J J and Lu J 2014 Nanoscale 6 7609
- [31] Liu H S, Han N N and Zhao J J 2014 J. Phys.: Condens. Matter 26 475303
- [32] Houssa M, Pourtois G, Afanas'ev V and Stesmans A 2010 Appl. Phys. Lett. 97 112106
- [33] Lew Yan Voon L, Sandberg E, Aga R and Farajian A 2010 Appl. Phys. Lett. 97 163114
- [34] Wang R, Pi X D, Ni Z Y, Liu Y, Lin S S, Xu M S and Yang D R 2013 Sci. Rep. 3 3507
- [35] Ding Y and Ni J 2009 Appl. Phys. Lett. 95 083115
- [36] Padova P D, Quaresima C, Ottaviani C, Sheverdyaeva P M, Moras P, Carbone C, Topwal D, Olivieri B, Kara A, Oughaddou H, Aufray B and Lay G L 2010 Appl. Phys. Lett. 96 261905
- [37] Feng B J, Li H, Meng S, Chen L and Wu K H 2016 Surf. Sci. 645 74
- [38] Ezawa M 2012 New J. Phys. 14 033003
- [39] Ni Z Y, Liu Q H, Tang K C, Zheng J X, Zhou J, Qin R, Gao Z X, Yu D P and Lu J 2012 *Nano Lett.* **12** 113
- [40] Li L, Lu S Z, Pan J, Qin Z, Wang Y Q, Wang Y, Cao G Y, Du S and Gao H J 2014 Adv. Mater. 26 4820
- [41] Feng B J, Ding Z J, Meng S, Yao Y G, He X Y, Cheng P, Chen L and Wu K H 2012 Nano Lett. 12 3507
- [42] Li G, Zhang L, Xu W, Pan J, Song S, Zhang Y, Zhou H, Wang Y, Bao L, Zhang Y Y, Du S, Ouyang M, Pantelides S T and Gao H J 2018 Adv. Mater 30 1804650
- [43] Qin Z H, Pan J B, Lu S Z, Yan S, Wang Y L, Du S X, Gao H J and Cao G Y 2017 Adv. Mater. 29 1606046
- [44] Chen L, Liu C C, Feng B, He X, Cheng P, Ding Z, Meng S, Yao Y and Wu K 2012 Phys. Rev. Lett. 109 056804
- [45] Feng B J, Zhou H, Feng Y, Liu H, He S L, Matsuda I, Chen L, Schwier E F, Shimada K and Meng S 2019 Phys. Rev. Lett. 122 196801
- [46] Meng L, Wang Y L, Zhang L Z, Du S X, Wu R T, Li L F, Zhang Y, Li G, Zhou H T, Hofer W A and Gao H J 2013 Nano Lett. 13 685
- [47] Zhu F F, Chen W J, Xu Y, Gao C L, Guan D D, Liu C H, Qian D, Zhang S C and Jia J F 2015 *Nat. Mater.* 14 1020
- [48] Yang Z Q, Jia J F and Qian D 2016 Chin. Phys. B 25 117312
- [49] Tao L, Cinquanta E, Chiappe D, Grazianetti C, Fanciulli M, Dubey M, Molle A and Akinwande D 2015 Nat. Nanotechnol. 10 227
- [50] Shao Y, Liu Z L, Cheng C, Wu X, Liu H, Liu C, Wang J O, Zhu S Y, Wang Y Q, Shi D X, Ibrahim K, Sun J T, Wang Y L and Gao H J 2018 *Nano Lett.* 18 2133
- [51] Wu X, Shao Y, Liu H, Feng Z, Wang Y L, Sun J T, Liu C, Wang J O, Liu Z L, Zhu S Y, Wang Y Q, Du S X, Shi Y G, Ibrahim K and Gao H J 2017 Adv. Mater. 29 1605407
- [52] Zhu S Y, Shao Y, Wang E, Cao L, Li X Y, Liu Z L, Liu C, Liu L W, Wang J O, Ibrahim K, Sun J T, Wang Y L, Du S and Gao H J 2019 *Nano Lett.* **19** 6323

- [53] Zhang S L, Yan Z, Li Y F, Chen Z F and Zeng H B 2015 Angew. Chem. 127 3155
- [54] Zhang S, Zhou W, Ma Y, Ji J, Cai B, Yang S A, Zhu Z, Chen Z and Zeng H 2017 Nano Lett. 17 3434
- [55] Zhang S, Xie M, Cai B, Zhang H, Ma Y, Chen Z, Zhu Z, Hu Z and Zeng H 2016 Phys. Rev. B 93 245303
- [56] Zhang S, Xie M, Li F, Yan Z, Li Y, Kan E, Liu W, Chen Z and Zeng H 2016 Angew. Chem. Int. Ed. 55 1666
- [57] Zhou W H, Cai B, Guo S Y, Zhang S L, Hu X M, Qu H Z and Zeng H B 2019 Appl. Mater. Today 15 163
- [58] Zhou W H, Chen J Y, Bai P X, Guo S Y, Zhang S L, Song X F, Tao L and Zeng H B 2019 *Research* 2019 1
- [59] Drozdov I K, Alexandradinata A, Jeon S, Nadj-Perge S, Ji H W, Cava R J, Bernevig B A and Yazdani A 2014 Nat. Phys. 10 664
- [60] Reis F, Li G, Dudy L, Bauernfeind M, Glass S, Hanke W, Thomale R, Schafer J and Claessen R 2017 Science 357 287
- [61] Song Z G, Liu C C, Yang J B, Han J Z, Ye M, Fu B T, Yang Y C, Niu Q, Lu J and Yao Y G 2014 NPG Asia Mater. 6 e147
- [62] Zhou J J, Feng W X, Liu G B and Yao Y G 2015 New J. Phys. 17 015004
- [63] Qian X F, Liu J W, Fu L and Li J 2014 Science 346 1344
- [64] Ugeda M M, Pulkin A, Tang S, Ryu H, Wu Q, Zhang Y, Wong D, Pedramrazi Z, Martin-Recio A, Chen Y, Wang F, Shen Z X, Mo S K, Yazyev O V and Crommie M F 2018 *Nat. Commun.* 9 3401
- [65] Song Y H, Jia Z Y, Zhang D, Zhu X Y, Shi Z Q, Wang H, Zhu L, Yuan Q Q, Zhang H, Xing D Y and Li S C 2018 Nat. Commun. 9 4071
- [66] Peng L, Yuan Y, Li G, Yang X, Xian J J, Yi C J, Shi Y G and Fu Y S 2017 Nat. Commun. 8 659
- [67] Tang S J, Zhang C F, Wong D, Pedramrazi Z, Tsai H Z, Jia C J, Moritz B, Claassen M, Ryu H, Kahn S, Jiang J, Yan H, Hashimoto M, Lu D H, Moore R G, Hwang C, Hwang C, Hussain Z, Chen Y L, Ugeda M M, Liu Z, Xie X M, Devereaux T P, Crommie M F, Mo S K and Shen Z X 2017 Nat. Phys. 13 683
- [68] Li J H, Li Y, Du S Q, Wang Z, Gu B L, Zhang S C, He K, Duan W H and Xu Y 2019 Sci. Adv. 5 eaaw5685
- [69] Li H, Gao S Y, Duan S F, Xu Y F, Zhu K J, Tian S J, Gao J C, Fan W H, Rao Z C, Huang J R, Li J J, Yan D Y, Liu Z T, Liu W L, Huang Y B, Li Y L, Liu Y N, Zhang G B, Zhang P, Takeshi Kondo, Shik Shin, Lei H C, Shi Y G, Zhang W T, Weng H M, Qian T and Ding H 2019 *Phys. Rev. X* 9 041039
- [70] Otrokov M M, Klimovskikh I I, Bentmann H, et al. 2019 Nature 576 416
- [71] Deng Y J, Yu Y J, Shi M Z, Guo Z X, Xu Z H, Wang J, Chen X H and Zhang Y B 2020 Science 367 895
- [72] Liu C, Wang Y C, Li H, Wu Y, Li Y X, Li J H, He K, Xu Y, Zhang J S and Wang Y Y 2020 Nat. Mater. 19 522
- [73] Chen Y L, Analytis J G, Chu J H, Liu Z K, Mo S K, Qi X L, Zhang H J, Lu D H, Dai X, Fang Z, Zhang S C, Fisher I R, Hussain Z and Shen Z X 2009 *Science* 325 178
- [74] Hajlaoui M, Papalazarou E, Mauchain J, Lantz G, Moisan N, Boschetto D, Jiang Z, Miotkowski I, Chen Y P, Ibrahimi A T, Perfetti L and Marsi M 2012 Nano Lett. 12 3532
- [75] Lai J, Liu Y, Ma J, Zhuo X, Peng Y, Lu W, Liu Z, Chen J and Sun D 2018 ACS Nano 12 4055
- [76] Chan C K, Lindner N H, Refael G and Lee P A 2017 Phys. Rev. B 95 041104
- [77] McIver J W, Hsieh D, Steinberg H, Jarillo-Herrero P and Gedik N 2012 Nat. Nanotechnol. 7 96
- [78] Rao S 2016 J. Indian Inst. Sci. 96 145
- [79] Zhu Z, Yan D, Nie X A, Xu H K, Yang X, Guan D D, Wang S y, Li Y Y, Liu C h, Liu J W, Luo H X, Zheng H and Jia J F 2019 *Chin. Phys. B* 28 077302
- [80] Ma J C, Gu Q Q, Liu Y N, Lai J W, Yu P, Zhuo X, Liu Z, Chen J H, Feng J and Sun D 2019 Nat. Mater. 18 476
- [81] Pauly C, Rasche B, Koepernik K, Liebmann M, Pratzer M, Richter M, Kellner J, Eschbach M, Kaufmann B, Plucinski L, Schneider Claus M, Ruck M, van den Brink J and Morgenstern M 2015 *Nat. Phys.* 11 338