# Managing excess Pbl<sub>2</sub> for efficient perovskite solar cells

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Perovskite solar cells (PSCs) have attracted much attention due to their low cost, high efficiency and easy processing. Recently, You *et al.* achieved a record efficiency of 26.1% (certified 25.6%) in PSCs with excellent stability<sup>[1]</sup>. Excess Pbl<sub>2</sub> in perovskite films was converted into inactive (Pbl<sub>2</sub>)<sub>2</sub>RbCl to avoid its detrimental effect to device stability, while maintaining its positive effect to device efficiency.

Most highly efficient PSCs were achieved with excess Pbl<sub>2</sub> in either perovskite bulk or film surface. For example, Seo et al. fabricated PSCs with >25% efficiency by using perovskite precursor solution with excess PbI2<sup>[2]</sup>. Besides, You et al. achieved planar PSCs with efficiency close to 20% in 2016 through annealing perovskite films at high temperature to induce its partial decomposition and form excess Pbl<sub>2</sub> spontaneously<sup>[3]</sup>. The increased efficiency in PSCs with excess Pbl<sub>2</sub> resulted from Pbl<sub>2</sub> passivation. Owing to the larger bandgap of  $Pbl_2$ , a favorable type I band alignment was established between Pbl<sub>2</sub> and perovskite, forming energy barriers to block both electron and hole transport to grain boundaries (GBs) and repel them into perovskite grains<sup>[4, 5]</sup>. As a result, potential carrier recombination induced by defects at GBs was effectively inhibited, thus improving device efficiency. In addition, the excess Pbl<sub>2</sub> at perovskite/charge-transport layer interface can block possible interfacial recombination, further enhancing device efficiency.

However, scientists found that excess Pbl<sub>2</sub> in perovskite films could cause severe stability issues of PSCs especially under illumination, despite increased efficiency<sup>[6, 7]</sup>. Pbl<sub>2</sub> tended to decompose into metallic Pb<sup>0</sup> and l<sub>2</sub> under illumination, yielding Pb<sup>0</sup> defects and thus decreasing device efficiency<sup>[8]</sup>. I<sub>2</sub> product could further promote perovskite decomposition, even destroy charge-transporting layer<sup>[9, 10]</sup> and corrode metal electrode<sup>[11, 12]</sup>, accelerating device degradation. In addition, excess Pbl<sub>2</sub> can absorb adjacent MA<sup>+</sup> (or FA<sup>+</sup>) and I<sup>-</sup> during perovskite aging, yielding MA<sup>+</sup> and I<sup>-</sup> vacancies<sup>[13]</sup>. Theses vacancies can act as channels for ion migration in perovskite films, also leading to device degradation.

To reduce the negative effect of excess Pbl<sub>2</sub>, Zhu *et al.* reported a chemical polishing method to optimize perovskite films. They introduced excess Pbl<sub>2</sub> during perovskite growth to obtain high-quality perovskite films. Then they washed away excess Pbl<sub>2</sub> on perovskite surface through its

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reaction with ammonium salts in the polishing agents (Fig. 1(a))<sup>[14]</sup>, avoiding Pbl<sub>2</sub>-induced perovskite degradation. The PSCs exhibited over 24% efficiency with good light and air stability. In addition, Luo *et al.* reported a ligand-modulated method (Fig. 1(b)) to regulate the shape and distribution of excess Pbl<sub>2</sub> in perovskite films via introducing cetyltrimethylammonium bromide (CTAB)<sup>[15]</sup>. CTAB caused Pbl<sub>2</sub> into 2D nanosheet, vertically sticking among perovskite grains. As a result, over 22% efficiency was obtained. Besides, Li *et al.* modulated excess Pbl<sub>2</sub> with ionic liquid [BMIM]X (Fig. 1(c))<sup>[16]</sup>. [BMIM]X could interact with Pbl<sub>2</sub> and form supramolecular structures, which could relax lattice distortion and release residual tensile strain in perovskite films, thus improving device stability.

Recently, You *et al.* managed excess Pbl<sub>2</sub> through RbCl doping in perovskite films<sup>[1]</sup>. RbCl reacted with Pbl<sub>2</sub> to form  $(Pbl_2)_2RbCl$  (Fig. 2(a)).  $(Pbl_2)_2RbCl$  promoted the formation and stabilization of black FAPbl<sub>3</sub> phase through Pb-Cl bonding. On the other hand,  $(Pbl_2)_2RbCl$  was inactive and could not react with FA<sup>+</sup> or I<sup>-</sup> of perovskite grains, thus inhibiting the formation of FA<sup>+</sup> or I<sup>-</sup> vacancies and suppressing ion migration (Fig. 2(b)). Besides,  $(Pbl_2)_2RbCl$  is an intercalated compound and is more favorable to block ion migration in perovskite. As a result, a record efficiency of 26.1% (certified 25.6%) was realized in PSCs with good stability (Fig. 2(c)). The devices retained 96% of initial efficiency after aging in N<sub>2</sub> for 1000 h and 80% of initial value after aging at 85°C for 500 h.

In short, future efforts on PSCs will focus on improving the module efficiency and stability for real commercialization.

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Fig. 1. (Color online) (a) Chemical polishing of perovskite surface to eliminate excess Pbl<sub>2</sub>. Reproduced with permission<sup>[14]</sup>, Copyright 2022, American Chemical Society. (b) Ligand-modulated Pbl<sub>2</sub> nanosheet in PSCs. Reproduced with permission<sup>[15]</sup>, Copyright 2020, Wiley. (c) Supramolecular engineering to modulate excess Pbl<sub>2</sub> in PSCs. Reproduced with permission<sup>[16]</sup>, Copyright 2022, Wiley.



Fig. 2. (Color online) (a)  $(Pbl_2)_2RbCl$  and its crystal structure in PSCs. (b) Ion-migration activation energy in control perovskite or perovskite with  $(Pbl_2)_2RbCl$ . (c) *J–V* curves for PSCs with  $(Pbl_2)_2RbCl$ . Reproduced with permission<sup>[1]</sup>, Copyright 2022, the American Association for the Advancement of Science.

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