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lon migration in perovskite solar cells

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The past few years have witnessed a remarkable pro-671.4 gress of perovskite solar cells (PSCs), which can be attributed 659.4 to the high light absorption coefficient, tunable bandgap, 547.4 long carrier diffusion length, solution processability at low temperature and relatively low cost of perovskite materials. Up to now, the highest certified power conversion efficiency (PCE) of PSCs has reached 25.5% (https://www.nrel.gov/pv/cell-efficiency.html), shedding light on the commercialization of PSCs. Nevertheless, the stability issues of PCSs severely limit the field deployment of solar cell modules. Notably, ion migra-463.4 tion in organic–inorganic halide perovskite (OIHP) originat-451.4 ing from the ionic nature of perovskite materials has been suggested to be responsible for many instability problems and 427.4 cannot be eliminated through encapsulation. With crucial significance being attached to the long-term durability of PSCs dur-403.4 ing operation, it is of profound implication to clarify the fundamental mechanism and potential influence of ion migration 379.4 on device performance. 367.4

To date, numerous researchers have provided evidences for the existence of ion migration in hybrid PSCs, including dynamic electrical characterizations, such as the J–V hysteresis, 331.4 the giant dielectric constant, and the switchable photocurrent, and direct elemental mapping, e.g. TPIR, focused XPS, 807.4 TOF-SIMS, et al. However, ion migration in OIHP solar cells295.4 has not drawn much attention until Snaith *et al.* in 2013 MRS fall meeting reported the J–V hysteresis phenomenon and 271.4 speculated that ion migration might be one of the three possible origins^[1], and Xiao *et al.* in 2014 reported the giant switchable photovoltaic effect and firstly proved the signific-235.4 ant ion migration in perovskites at RT^[2]. In the giant switch-p23.4 able photovoltaic effect, the perovskite layer can be poled under external bias, making the direction of photoinduced cur-199.4 rent completely reversible with the applied bias (Fig. 1(a)).187.4 Schematically, Fig. 1(b) shows positively charged ions drift 175.4 along the direction of the field and accumulate at the cathode region, while negatively charged ions are just the opposite. Moreover, the perovskite material adjacent to anodense, turned transparent under constant poling, indicating the existence of ion drifting from anode^[2]. 15.4

The species of migrating ions are suggested to be either

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intrinsic ions, such as MA⁺, I[−], within the perovskite, or extrins-571.4 ic ions from interfacial layers, e.g. Li+, H+, Na+. Based on the first-principle calculation, Eames *et al.* simulated the corresponding activation energies of three conventional ion migration mechanisms, where I⁻ travels along an octahedron edge_{523.4} with a lowest activation energy, MA⁺ moves to the neighboring octahedral vacancy with a moderate activation energy and Pb²⁺ migrates along the diagonal direction <110> with $a_{487.4}$ highest activation energy^[3]. It reveals that the most easily oc-475.4 curred migration is I⁻, followed by MA⁺, and Pb²⁺ turns to be_{463.4} difficult to migrate. At the same time, Azpiroz et al. analyzed the activation energies of four typical point defects (I vacan-439.4 cies, MA vacancies, Pb vacancies and I interstitials) related to ion migration and raised four plausible diffusion pathways^[4] 415.4 which are illustrated in Fig. 1(c). Despite the different numeric-403.4 al values of activation energy from different computational 891.4 models, it can still be elucidated that I interstitials and I vacan-1979.4 cies are the easiest to migrate compared with other defects. 367.4

Experimentally, the fact of ion migration was confirmed by Yuan *et al.* in 2015, when they first observed the migra-343.4 tion of MA⁺ under a poling voltage in MAPbl₃ PSCs through PTIR^[5]. In addition, a p-i-n structured device (with switch-B19.4 able polarity) was discovered to form in-situly in perovskite layer due to the chemical doping effect induced by excessive ions. Furthermore, Li *et al.* guantitively monitored the distribu-p83.4 tion of I/Pb ratio (within a lateral structure device) after electrical poling process and found it went much higher around anode region^[6]. After withdrawing poling bias, I⁻ ion showed reversible diffusion and returned back into its original state, veri-p35.4 fying the drifting and accumulation of I- ion at anode under poling voltage. Except for intrinsic ions, Li *et al.*^[7] pointed out that extrinsic ions like Li⁺, H⁺, and Na⁺ can also penetrate^{199,4} through perovskite film and accumulate in the counter transport layer, as evidenced by the distribution profile of different elements in a common structured cell (TiO₂/perovskite/nead spiro-OMeTAD) derived from TOF-SIMS.

Recent studies have indicated that the migration of ionic species has critical impacts on the performance and longterm operational stability of PSCs.

The migration and accumulation of ions have been 103.4 demonstrated to affect the charge carrier transport and elec-14.4 tronic properties in the perovskite film. As show in Fig. 1(b),79.4 Xiao *et al.* reported the switchable photovoltaic effect^[2], the 67.4 ion migration under the applied electric filed induces a self-55.4 doping effect and local band bending, forming a p-i-n or 13.6 n-i-p type junction. The concentration of accumulated ions

93.6 91.2



Fig. 1. (Color online) (a) Giant switchable photocurrent curve in a non-selective contact structured device. Reproduced with permission^[2], Copyright 2014, Nature Publishing Group. (b) Schematic representation of ion migration in a perovskite solar cell when applying forward (left) and reverse bias (right). Reproduced with permission^[2], Copyright 2014, Nature Publishing Group. (c) The possible migrating paths for I vacancies, MA vacancies, Pb vacancies and I interstitial defects. Reproduced with permission^[4], Copyright 2015, The Royal Society of Chemistry. (d) Schematic diagrams indicating the influence of vacancy drift on the band energies of a p-i-n device at short circuit^[3].

could be 10¹⁷ cm⁻³ or higher, which severely quench the photoluminescence (PL) intensities. Eames^[3] proposed that the migration of ion species can generate a local electric field, ^{475,9} which partially shield the built-in electric field and influence the collection efficiency of photogenerated carriers (Fig. 1(d)), ^{451,9} A high concentration of opposite ions accumulated around ^{439,9} grain boundaries (GBs) would form tunneling junctions with ^{427,9} sharp surface potential change^[8]. Xu *et al.* reported that the formation of I–Pb antisite defects with excessive I⁻ ions on the surface of perovskite film could form deep level traps ac ording to the density functional (DFT) calculation, which are responsible for the non-radiative recombination loss of ^{267,9} charge carrier^[9].

A key reason for the mobile ions play diverse roles in perovskite is their very low threshold electrical field to migrate, e.g. 0.3 V/ μ m, which is much smaller than the build-in electrical field in solar cells or local band bending induced by buildup charges^[5]. Faster ion migration along GBs than bulk was experimentally proved by Shao *et al.* based on an in-situe mapping of I and Pb elements around jointed single-crystals with GB parallel to the electrical field direction^[10]. The dramatic morphology change and the I element redistribution along GB under applied bias demonstrated GBs as the priority ion migration channels.

The migration and redistribution of ion species also result in the instability of perovskite phase, eventually causing decomposition of the films and device failure. Yuan *et al.* disrowered the phase transformation between MAPbl₃ perovskite and Pbl₂ crystals under an external electric field due to the migration of mobile ions (MA⁺, I⁻)^[11]. This structural transformation can be recovered at 330 K. Moreover, the per-

ovskites with mixed halides compositions tend to be phasesegregated under light illumination or applied electric filed which was verified to lead to reversible performance loss for PSCs. As shown in Fig. 2(a), Hoke *et al.* discovered the reversible photoluminescence shift in MAPb(Br_xI_{1-x})₃ mixed per-ps1.9 ovskite^[12]. This indicated that light illumination can cause the ion redistribution and halide segregation into iodide-rich and bromide-rich domains. Zhang *et al.* studied the phase-segregation effect in CsPbBr_{1.2}I_{1.8} nanocrystal ensemble, suggesting that it is the local electric field to break the iodide bonds that triggers the ion migration process^[13]. More importantly, the mobile ion species migrate across the perovskite layer and re-per-9 act with the function layer, leading to irreversible performance degradation and severely stability issues of PSCs. Kato et al. observed that the mobile I⁻ could react with the elec-pain trode and form Agl interface layer, which progressively re-219.9 duced the PSCs performance^[14]. Li *et al.*^[7] highlighted the im-por.9 portant roles of extinct ions migration such as Li⁺, Na⁺ and H⁺_{195.9} dopant used in the hole transport layer (HTL) can also provoke hysteresis and intensified PSCs performance attenuation.

Remarkably, since the redistribution of mobile ions is driven by dynamic equilibrium, the detrimental effect of ion migration in stability can be healed when the loss of mobile ions is ignorable. The light induced ion redistribution was also found to be able to cure the performance of PSCs. deQuilettes^[15] illustrated the photo-induced 'clean effect' due to the migration and redistribution of iodide ions, which contributes to the reduction of the trap density in the perovskite films (Fig. 2(b)). Furthermore, Motti *et al.*^[16] investigated the defilms (Fig. 2(b)). Furthermore, Motti *et al.*^[16] investigated the defilms (Fig. 2(b)). Furthermore, Motti *et al.*^[16] investigated the defilms (Fig. 2(b)). Furthermore, Motti *et al.*^[16] investigated the de-

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Fig. 2. (Color online) (a) The 200 XRD peak of an x = 0.6 film before (black) and after (red) white-light soaking for 5 minutes at ~50 mW/cm². XRD patterns of an x = 0.2 film (dashed green) and an x = 0.7 film (dashed brown) are included for comparison. Reproduced with permission^[12], Copyright 2015, Royal Society of Chemistry. (b) Proposed mechanism of photo-induced cleaning by halide redistribution. Reproduced with permission^[15], Copyright 2016, Nature Publishing Group. (c) (d) Photoluminescence enhancement and quenching mechanisms. (c) lon dynamics in MAPbl₃ thin film promoting PLIE, when the probability of 1⁰ species encounters is small and Frenkel pair annihilation is boosted by electron trapping, and PLID (d) when the probability of 1⁰ species encounters is high, boosting l₂ molecule formation. (c/d) Reproduced with permission^[16], Copyright 2019, Nature Publishing Group.

various quality of perovskite films. The long-living halide carriers can heal the defect through Frenkel defects annihilation when the trap density of the perovskite film is relatively low, as Fig. 2(c) depicted. On the contrary, the high-density traps provide binging sites for photogenerated species, boosting 12423 molecule formation and eventually leading to material degradation (Fig. 2(d)). Consequently, covering perovskite with compact materials possessing surface defect passivation effect is an effective strategy to prevent ion loss as a promising direc-142.

Recently, significant research efforts have been devoted ^{70.3} to suppress the ion migration from the perspective of composi-^{58.3} tion, crystallization, and dimension engineering. Compared to ^{44.3} MAPbl₃, the ion migration tendency is lower in FA-based and ^{44.3} all-inorganic PSCs, indicating the sensitivity of material stabilit-^{42.3} ies on its compositions. Lin *et al.*^[18] applied the temperature ^{110.3} dependent conductivity experiment and demonstrated that ^{86.3} the ion migration in 2D (BA)₂(MA)₃Pb₄l₁₃ perovskite is dramatic-^{86.3} ally suppressed due to the confinement effect of organic ^{42.3} spacers. In addition, Liu *et al.* introduced the (2D) A₂Pbl₄ per-^{42.3} ovskite layer between the 3D perovskite layer and the HTL to ^{50.3} form the 2D/3D heterogeneous junction, meanwhile redu-

cing the ion migration in the device and protecting the persource of strain engineering on improving the intrinsic stabiltice strain plays the opposite role, which highlights the importance of strain engineering on improving the intrinsic stabiltice attain engineering on and enhance stability of PSCs^[21].

To summarize, ion migration is detrimental to perform-46.3 ance and long-term operational stability, which is still hinder-134.3 ing the large-scale and commercial development of PSCs. 1023 Therefore, systematical research on the mechanism of ion mi-1103 gration induced phase segregation can supply the theoretic-96.3 al foundation for effectively eliminating hysteresis and enhan-96.3 cing the stability of PSCs. However, more efforts will be re-74.3 quired to elucidate the ion migration phenomenon in halide 92.3 perovskite materials. If ion loss is prevented, the dynamic equi-90.3 librium nature of mobile ion can be utilized to design state of the art high performance and stable perovskite optoelectron-7675 [ic devices. 7455

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