Boosting anapole-exciton strong coupling in all-dielectric heterostructures

JINGYU WANG,^{1,†} WEIMIN YANG,^{1,†} GUOYA SUN,^{1,2} YONGLIN HE,¹ PEIWEN REN,¹ AND ZHILIN YANG^{1,3}

¹Department of Physics, Collaborative Innovation Center for Optoelectronic Semiconductors and Efficient Devices, Jiujiang Research Institute, Xiamen University, Xiamen 361005, China

²e-mail: gysun@xmu.edu.cn ³e-mail: zlyang@xmu.edu.cn

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The light manipulation beyond the diffraction limit plays an invaluable role in modern physics and nanophotonics. In this work, we have demonstrated a strong coupling with a large Rabi splitting of 151 meV between bulk WS_2 excitons and anapole modes in the WS_2 -Si nanodisk heterostructure array with nanoholes as small as 50 nm radius. This result is acquired by introducing anapole modes to suppress radiative losses to confine light into subwavelength volumes and large spatial overlapping between excitons and strong optical fields. Our work shows that anapole modes may serve as a powerful way to enhance the interaction between light and matter at nanoscales, and it should pave an avenue toward high-performance all-dielectric optoelectronic applications. © 2022 Chinese Laser Press

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1. INTRODUCTION

The light manipulation at nanoscales not only lays the foundation of a wealth of science, particularly in the fields of physics, chemistry, and communications, but can also promote a considerable number of extraordinary applications in modern physics and photonics [1-9]. However, conventional light manipulation methods are often limited by the optical diffraction limit, making them bottlenecked in miniaturization and further hindering their practical applications. In the past decades, surface plasmon resonance (SPR) in metal nanostructures has served as a promising candidate for overcoming this barrier due to its abilities in channeling the far-field radiation to subwavelength dimensions [10–19]. Nevertheless, due to inherent ohmic loss of plasmonic metals and radiative loss characteristics of conventional optical resonance modes, the use of plasmonic nanocavities in light manipulation is limited. Therefore, it is highly desirable for researchers to seek new optical resonance modes with subradiant characteristics and materials with intrinsically low optical losses.

More recently, distinct optical resonances in dielectric nanostructures have been developed as a new method to facilitate light manipulation in nanophotonics [20–23]. In this method, high-refractive-index materials such as Ge, GaAs, and Si are employed to reduce dissipative losses owing to the small imaginary parts of their dielectric functions. Meanwhile, specific resonant optical modes are selectively excited to diminish the total electric dipole moment of dielectric nanostructure to suppress radiation loss. Particularly, the anapole mode stands out from

the crowd in efficient light manipulation due to the nonradiant resonance characteristic [24–26]. The anapole mode originates from the destructive interference between the electric dipole (ED) and toroidal dipole (TD) that have the same amplitude but opposite phases in the far field [27,28]. So far, light manipulation based on the anapole mode has been widely investigated in the weak coupling regime, such as local field enhancement, nonlinear optical effects, and nanolasers [29-32]. In the strong coupling regime, pioneering researchers have already accomplished the interaction between the anapole mode and exciton in heterostructures [33-36]. However, these optical cavities feature larger volumes and relatively weak field enhancement. Moreover, strong light-matter interactions often come with the challenges of aligning the resonator cavities with the excitons, while the electromagnetic energy of the anapole mode in previously pioneering research is mostly distributed inside the nanostructures, which makes only a small number of molecules able to access the near-field enhancement directly and results in smaller Rabi splitting [37]. Therefore, it is still a challenge to obtain the spatial overlap between subwavelength optical cavity and excitons in an all-dielectric system in order to achieve efficient light manipulation based on the anapole mode.

Herein, we theoretically design a system capable of accomplishing the efficient light manipulation in nanoholes and further enhancing light-matter interactions by exploiting the anapole mode. It is noted that bulk WS_2 has indirect bandgaps and strong absorption peaks, which in turn can be coupled to and hybridize with optical cavity modes. For the purpose of high-efficiency coupling, a nanohole is introduced in the center of a Si nanodisk, and the bulk WS_2 is placed inside the hole. We found that anapole modes supported by the perforated Si nanodisk array can be strongly coupled to excitons of the bulk WS_2 , with a large Rabi splitting of 151 meV, which is much larger than cases where the excitons are outside the Si nanodisk [33–36]. This work is expected to pave the way for high-performance optoelectronic applications at nanoscales.

2. MODELS AND PRINCIPLE

The proposed model consists of a two-dimensional (2D) perforated Si nanodisk array filled with bulk WS₂ in the center, as shown in Fig. 1(a), in which the periodicity of system is given as $\Lambda = 500$ nm, with the radius and height of the Si nanodisk denoted by R and H, and those of the bulk WS_2 denoted by r and h, respectively. Electromagnetic characteristics of the abovementioned heterostructure array are investigated using commercial software Lumerical FDTD solutions. In calculations, we use a normally incident plane wave with polarization along the x axis and electric field amplitude of 1 V/m as the light source. A periodic boundary condition in the x-y plane and perfectly matched layer condition along the residual axis are adopted, while nonuniform spatial grids with a minimum Yee-cell size of 1 nm and simulation time of 1000 fs are taken to make sure of the spatial and temporal convergence, respectively. A time step of $dt \approx 0.0038$ fs is adopted to satisfy Courant-Friedrichs-Lewy stability criterion [38]. The optical constant for Si is taken from Ref. [39], and the permittivity of the bulk WS₂ is anisotropic with the in-plane components modeled as a Lorentz oscillator [33,40]:

$$\varepsilon_{\rm in} = \varepsilon_e + f_0 \frac{\omega_{\rm ex}^2}{\omega_{\rm ex}^2 - \omega^2 - i\gamma_{\rm ex}\omega},\tag{1}$$

where $\varepsilon_e = 18$ is the background permittivity, f_0 is the oscillator strength, and $\omega_{\rm ex} = 1.9$ eV and $\gamma_{\rm ex} = 90$ meV are the resonance frequency and full width of excitons, respectively. The out-of-plane permittivity is assumed to be a constant $\varepsilon_{\rm out} = 7$. The bulk WS₂ is treated as a background index-only material when $f_0 = 0$, and its excitonic effect is considered when $f_0 = 0.4$. Considering the actual experimental conditions, heterostructure arrays can be readily fabricated using top-down methods, such as electron beam lithography (EBL) or

focused ion beam (FIB), among others. The bulk WS_2 can be exfoliated on top of glass substrate and patterned disk via FIB milling, then a PMMA layer is coated on the substrate and treated by EBL procedures. The position of the PMMA mask is aligned to the WS_2 nanodisk using cross-shaped markers, then a Si layer can be deposited onto the substrate by magnet-ron sputtering, and the final sample is obtained after a following lift-off process.

The coupling between anapole modes in the Si nanodisk array and excitons in the bulk WS₂ can be understood using a simple physical model shown in Fig. 1(b). Generally, when planar incident light impinges on the structure, the ED moment is illustrated as a separation of positive and negative electrical charges, and the TD moment corresponds to currents flowing on the surface of a torus in the Si nanodisk array. Then the destructive interference between the enhanced TD resonance and the ED resonance happens while they have same amplitude but opposite phase, and consequently, the anapole state in the electromagnetic spectrum is realized. The unexcited state and resonant anapole state of Si nanodisks are noted as $|0\rangle$ and $|1\rangle$, respectively. At the same time, the filled WS₂ exciton will undergo a transition from the ground state $|g\rangle$ to the excited state $|e\rangle$ with a resonance energy of 1.9 eV under normal illumination. Generally, changing the radius of the Si nanodisk is an efficient optimization way to achieve the spectral overlap. Therefore, it is found that Si nanodisk-bulk WS₂ heterostructures in the confined electromagnetic environment can reach the strong coupling regime when the anapole mode in the Si nanodisks overlaps the absorption peak of the bulk WS₂ exciton by adjusting the radius of the Si nanodisk, enabling new hybrid energy states with upper branches (UB) and lower branches (LB) and coherent exchange of energy between the anapole mode and exciton in a reversible way. The energy exchange between light and matter will be periodical temporally, and when the exciton and anapole resonate, two new resonance frequencies can be observed from the transmittance spectrum [41].

3. RESULTS AND DISCUSSION

First, the optical responses of the Si nanodisk array under normal illumination with different structural parameters are studied. In Fig. 2(a), we plot the transmission contour map with respect to radii of Si nanodisks and wavelength of incident



Fig. 1. (a) Schematic illustration of the Si nanodisk-bulk WS_2 heterostructure array under normally incident illumination with a radius *R* and a height *H* of Si nanodisks, and a radius *r* and a height *h* of bulk WS_2 . (b) Schematic illustration of the strong interplay between the anapole state in the Si nanodisk and the bulk WS_2 exciton, leading to hybrid states with a large Rabi splitting.



Fig. 2. Analyses of the nonradiating anapole mode in the heterostructures. (a) Simulated transmission contour map as a function of radius of the Si nanodisk array (R = 100-200 nm), where the height of the Si nanodisk H = 70 nm. (b) Simulated transmission contour map as a function of height of the Si nanodisk array (H = 20-80 nm), where the radius of the Si nanodisk R = 130 nm. (c) Simulated transmission spectrum of the Si nanodisk array (blue solid curve); the Si nanodisk array with air hole, where the radius of air hole is 50 nm, the height is 70 nm, and radius of the Si nanodisk R = 130 nm (red solid curve); and the Si nanodisk-bulk WS₂ heterostructure array, where $f_0 = 0$ (black solid curve). The electric-field enhancement distribution in the *x*-*y* plane (upper panels) and the magnetic-field enhancement distribution in the *y*-*z* plane (lower panels) of the Si nanodisk array with air holes, and (f) the Si nanodisk-bulk WS₂ heterostructure array, where $f_0 = 0$. The black arrows represent the magnetic dipole *M* in lower panels of (d)–(f).

light, where the height of the Si nanodisk is H = 70 nm. It is noted that one strong resonance mode (transmission dip) can be clearly seen in Fig. 2(a), which displays a prominent redshift with increasing of the nanodisk radius. We have also calculated a transmission contour map as a function of height of the Si nanodisk, with R = 130 nm, as shown in Fig. 2(b), and one transmission minimum in the spectra is also found, which only displays a slight redshift with increasing of the height of the Si nanodisk. Considering the position of the bulk WS₂ exciton peak, we choose the Si nanodisk with a radius of 130 nm and a height of 70 nm, of which the resonance position is labeled as D1. In Fig. 2(c), we show the transmission spectrum of a Si nanodisk array without holes (blue solid curve), a Si nanodisk array with holes (red solid curve), and a Si nanodisk array with bulk WS_2 ($f_0 = 0$) inside holes (black solid curve), with R = 130 nm, H = h = 70 nm, and r = 50 nm. It is found that the D1 mode of the Si nanodisk array with nanoholes shows a more pronounced blueshift than that of the Si nanodisk array since the refractive index of air is significantly smaller than that of silicon, and the transmission spectrum of the Si nanodisk-bulk WS₂ ($f_0 = 0$) heterostructure array remains almost unchanged compared to that of the Si nanodisk array due to their similar refractive index (see Fig. 5 in Appendix A).

From Figs. 2(a)-2(c), it can be summarized that the spectroscopic behavior of D1 mode can be changed by varying structural parameters.

In Figs. 2(d)-2(f), we calculate near-field distributions and displacement vector maps under D1 mode. It is found that the displacement vector at the center cross section of the x-y plane forms two circles of opposite directions, which in turn generates a highly curved magnetic field and results in a state whose energy is confined inside the nanodisk, which closely matches those of anapole states from previous works [27,34,42–44]. More importantly, the maximum electric field of these systems is concentrated in the middle of the nanodisk, and the introduction of nanoholes and bulk WS₂ can result in more enhancement inside the structure than that of the Si nanodisk array as shown in Figs. 2(e) and 2(f). Consequently, the Si nanodisk-bulk WS₂ heterostructure array is chosen in this coupling study due to the larger electric field enhancement in the position of the bulk WS₂.

In addition, to reveal the effect of the periodicity on the D1 mode, optical properties for the different numbers of unit cells of the WS₂ ($f_0 = 0$)-Si nanodisk heterostructure are also explored with R = 130 nm, H = h = 70 nm, and r = 50 nm. In Fig. 6(a) (in Appendix B), we plot the scattering intensity of

the isolated, 3×3 -unit cell and 5×5 -unit cell for the WS₂ $(f_0 = 0)$ -Si nanodisk heterostructure, and it is noted that the lowest total scattering spectrum is located at about 650 nm, and the full width at half-maximum (FWHM) decreases significantly with the increase of the number of the unit cell. Figures 6(b)-6(e) show the *x*-*y*-plane electric field distributions at the resonance wavelength, which all have characteristic features of an anapole state. It is found that the FWHM of D1 mode of the heterostructure array decreases significantly, and the electromagnetic enhancement of the heterostructure array is improved considerably compared to that of the isolated heterostructure. Therefore, the periodicity of the structure can boost the near-field properties of the D1 mode. In Fig. 7 (in Appendix B), we plot the transmission spectrum and the near-field distribution of the WS_2 ($f_0 = 0$)-Si nanodisk heterostructure array with different periodicity size. It is noted that the transmission dip displays a prominent redshift with increasing of the periodicity. Moreover, the heterostructure array with a periodicity of 500 nm exhibits maximum electromagnetic field enhancement inside the bulk WS2, and its resonance wavelength of D1 mode is located near the position of the exciton absorption peak. Therefore, the periodicity of 500 nm is selected in the main text.

To investigate the physical origin of the D1 mode, we calculate the scattered power of ED, magnetic dipole (MD), and TD moments of the heterostructure array with $f_0 = 0$, utilizing the Cartesian multipole decomposition based on the induced current density. In this procedure, the oscillating charge current distribution inside the nanostructure, induced by an incident linearly polarized plane wave, is expanded into a series of electric, magnetic, and toroidal multipole moments. The Cartesian ED (*P*), MD (*M*), and TD (*T*) moments are performed by the following formula in the long-wavelength approximation [45]:

$$P = \frac{1}{i\omega} \int f(r) \mathrm{d}^3 r,$$
 (2)

$$M = \frac{1}{2c} \int r \times J(r) \mathrm{d}^3 r,$$
 (3)

$$T = \frac{1}{10c} \int \{ [r \cdot J(r)]r - 2(r \cdot r)J(r) \} d^3r,$$
 (4)

where *r* is the position vector, *c* is the light velocity, and ω is the frequency. *J*(*r*) indicates the charge current in the structure and can be evaluated with the FDTD. The scattered power of each dipole, *I*_{ED}, *I*_{MD}, and *I*_{TD}, can be obtained by the following formulas [46–48]:

$$I_{\rm ED} = \frac{2\omega^4}{3c^3} |P|^2,$$
 (5)

$$I_{\rm MD} = \frac{2\omega^4}{3c^3} |M|^2,$$
 (6)

$$I_{\rm TD} = \frac{2\omega^6}{3c^5} |T|^2.$$
 (7)

It can be found that the scattered power of ED and TD intersects at about 656 nm, as shown in Fig. 3(a), while the phase difference between them equals approximately π , in Fig. 3(b), indicating that the anapole mode is located at 656 nm. From Fig. 8 (in Appendix C), we can see that the transmission dip (D1) has a certain redshift compared to the intersection position of the ED and TD due to the array effect [49–52]. Therefore, strictly speaking, the anapole mode should be located at 656 nm of the transmission spectrum, but since the near-field distribution of D1 mode still satisfies the characteristics of the anapole mode, the D1 mode can be referred to as the weak anapole mode synergetic with the array effect (see Appendix C).

The realization of strong coupling requires the simultaneous spatial and spectral overlap, where the spatial coincidence has been accomplished by putting the bulk WS₂ into the maximum near-field region of the weak anapole mode in the Si nanodisk array. The spectral coincidence can be achieved by continuously redshifting the optical resonance mode across the excitonic transition by adjusting the radius of the Si nanodisk. In Fig. 9 of Appendix D, we first plot the transmission contour map of the Si nanodisk-bulk WS₂ ($f_0 = 0$) heterostructure array with respect to the radii of the Si nanodisks and



Fig. 3. (a) Normalized Cartesian multipole decomposition results for the Si nanodisk-bulk WS_2 heterostructure array, where the radius of center WS_2 is 50 nm and the height is 70 nm, which is treated as background index-only material. The inset presents the normalized scattered power in the wavelength range 630–670 nm. (b) Corresponding normalized phase of *P* and *T* for the structure of (a).

wavelength of incident light. It is found that the optical resonance mode redshifts from 580 to 825 nm as radius of the Si nanodisk increases from 100 to 200 nm. Therefore, changing the radius of the Si nanodisk is an efficient way to achieve the spectral overlap between the optical resonance mode and WS₂ exciton. In Fig. 4(a), we present the simulated radiusdependent transmission spectra of the hybrid nanostructure array with H = 70 nm, r = 50 nm, and $f_0 = 0.4$. It is noted that clear evidence of strong coupling spectral splitting is manifested in the transmission spectra as the weak anapole resonance energy shifts across the exciton. In order to obtain the coupling strength, we extract two dips of multiple transmission spectra in Fig. 4(a), as marked in Fig. 4(b) (blue and red spheres), and we use the classic coupled oscillator model to provide an accurate physical analyzer to reveal coupling behaviors in the hybrid nanostructure. In this model, the response of excitons can be viewed as an oscillator, and then the system can be modeled as two coupled harmonic oscillators:

$$\begin{pmatrix} E_{\text{anapole}} - i\Gamma_{\text{anapole}}/2 & g\\ g & E_{\text{exciton}} - i\Gamma_{\text{exciton}}/2 \end{pmatrix} \begin{pmatrix} \alpha\\ \beta \end{pmatrix}$$
$$= E \begin{pmatrix} \alpha\\ \beta \end{pmatrix}, \tag{8}$$

where E_{anapole} and E_{exciton} are energies of the uncoupled weak anapole resonance mode and the exciton, respectively, and g is the coupling strength. Γ_{anapole} and Γ_{exciton} represent dissipation rates of the uncoupled weak anapole resonance mode and the exciton, respectively; the former can be extracted to be ~100 meV from Fig. 2(c), and the latter is set as ~90 meV in Eq. (1). E stands for the energy corresponding to energies of the new quasiparticles, and α and β are eigenvector components, satisfying $|\alpha|^2 + |\beta|^2 = 1$. Because the widths of the uncoupled weak anapole and exciton are small compared to their energies, dissipations are ignored and eigenvalues are

$$E_{\pm} = (E_{\text{anapole}} + E_{\text{exciton}})/2 \pm \sqrt{g^2 + \frac{1}{4}\delta^2},$$
 (9)

where $\delta = E_{\text{anapole}} - E_{\text{exciton}}$ is the detuning energy between the weak anapole mode and the exciton in bulk WS₂. We then fit the transmission spectra using Eq. (9), as shown in Fig. 4(b) (solid curves), and it shows a clear anti-crossing behavior between the UB and LB with a Rabi splitting $\Omega = 2g = 151 \text{ meV}$ at $\delta = 0$, which satisfies strong coupling criterion ($\Omega > \frac{\Gamma_{\text{anapole}} + \Gamma_{\text{exciton}}}{2}$). To evaluate the coupling characteristic, we have investigated optical responses of the heterostructure array with R = 130 nm and r = 50 nm, as shown Figs. 4(c)–4(e).



Fig. 4. Analyses of the resonance coupling in the heterostructures, where the excitonic effect of bulk WS₂ is considered ($f_0 = 0.4$). (a) Simulated transmission contour map as a function of radius of the Si nanodisk (R = 80-200 nm), where the height of the Si nanodisk H = 70 nm, the radius of bulk WS₂ is r = 50 nm, and the height is h = 70 nm. (b) The energy of the upper branch (UB) (red sphere) and lower branch (LB) (blue sphere) as a function of detuning. The solid lines are fit to the coupled oscillator model, giving a splitting of 151 meV. (c) Simulated transmission spectrum of the Si nanodisk-bulk WS₂ heterostructure array when R = 130 nm and r = 50 nm, where the f_0 is set as 0 (black dash line) and 0.4 (red solid line). The electric-field enhancement distribution in the x-y plane (upper panels) and the magnetic-field enhancement distribution in the y-z plane (lower panels) at the (d) UB and (e) LB for the structure of (c).

When $f_0 = 0.4$, we observe the emergence of two transmission dips at 635 nm (labeled UB) and 693 nm (labeled LB) from Fig. 4(c). In addition to the dip splitting on transmission spectra, near-field patterns of the UB and LB will exchange around the resonance crossing, as shown in Figs. 4(d) and 4(e), where resonant hybridization is clearly shown. The electromagnetic field distributions of both the UB and LB resemble the typical anapole-like pattern.

Next, the resonance coupling for the heterostructure system with different radius r of WS₂ is demonstrated in Appendix E. For $f_0 = 0$, the weak anapole mode in the transmission spectra displays a weak redshift with increasing r as shown in Fig. 10(a) (in Appendix E). For $f_0 = 0.4$, we can see that the splitting between the LB and UB hybrid modes is broadening with the increase of r [see Fig. 10(b) in Appendix E]. When r is in the range from 60 to 100 nm, the middle transmission dip is attributed to the Mie resonance mode of bulk WS₂ due to its high refractive index. This observation can be understood by investigating the electromagnetic field enhancement of the heterostructure array with different WS₂ radius of 20, 50, and 100 nm, respectively, which is shown in Figs. 10(c)-10(e) (in Appendix E). It is found that the electromagnetic field enhancement inside the bulk WS_2 is much larger for r = 100 nm than for the other two. Moreover, a Rabi splitting of 260 meV can be acquired with r = 100 nm from Fig. 10, which is far greater than the previous work [33], revealing that the heterostructure array can enhance the interaction between bulk WS₂ excitons and the anapole mode compared with a single WS_2 nanodisk. In Fig. 11 (in Appendix E), we plot the transmission contour map with respect to oscillator strength f_0 and wavelength of incident light for the heterostructure array. It is found that the splitting between the LB and UB hybrid anapole modes is broadening with the increasing of f_0 . The refractive index is proportional to f_0 , and a stronger dipole moment would be generated for the bulk WS_2 with a larger f_0 [34]. Therefore, a stronger polarized electromagnetic field generated by bulk WS_2 can be felt by the Si nanodisk when f_0 is enlarged, which will lead to a stronger resonance coupling between the exciton and the anapole mode.

In Appendix F, the coupling behavior is studied when the bulk WS₂ is put outside the Si nanodisk. There exists no dip splitting in the transmission spectra while the WS_2 is placed on top of the Si nanodisk surface as shown in Fig. 12(a) (in Appendix F). A strong coupling regime with a smaller Rabi splitting can be achieved when the WS₂ is put on the side of the Si nanodisk as shown in Fig. 12(b) (in Appendix F). We can reasonably deduce that the very weak coupling strength between excitons and anapole modes is due to the fact that electromagnetic field enhancement caused by the weak anapole mode is away from the WS₂. Moreover, in Fig. 13 (in Appendix G), we studied the coupling characteristic of the Si nanodisk-J-Aggregate heterostructure array, where the radii of the J-Aggregate and the Si nanodisk are 40 nm and 150 nm, respectively. We found that the anapole mode supported by perforated Si nanodisks can be strongly coupled to the J-Aggregate with a Rabi splitting of 200 meV, which is far greater than in previous works [34]. It is noted that the Rabi splitting in our strategy can be largely amplified, compared with

systems where the J-Aggregates are placed outside the Si nanodisk directly.

4. CONCLUSIONS

In summary, we have studied the coupling characteristics of excitons and anapole modes in a Si nanodisk-bulk WS_2 heterostructure array using FDTD. By opening up a dedicated nanohole at the center of the Si nanodisk, the inside electrical field generated by the anapole modes reaches its maximum, i.e., the near-field enhancement can be significantly amplified. At the same time, the nanohole structure makes it feasible to realize spatial overlap between excitons in bulk WS_2 and the confined electromagnetic energy. Strong coupling with a large Rabi splitting of 151 meV is obtained. The present work may provide theoretical contributions to potential applications in optoelectronic devices, light harvesting, and photocatalysis.

APPENDIX A: REFRACTIVE INDEX OF BULK WS₂

To compare the refractive index of the Si and bulk WS₂, we give the specific refractive index of bulk WS₂ including the case of $f_0 = 0$ and $f_0 = 0.4$ in Fig. 5.



Fig. 5. Wavelength dependence of the (a), (b) real and (c), (d) imaginary parts of the in-plane complex index of refraction for a Lorentz model dielectric of bulk WS₂ with (a), (c) $f_0 = 0$ and (b), (d) $f_0 = 0.4$.

APPENDIX B: ADVANTAGES OF THE PERIODICITY

1. Spectra of Anapole Modes in the Isolated Heterostructure

In Fig. 6(a), we plot the simulated scattering spectrum of the isolated, 3×3 -unit cell and 5×5 -unit cell for WS₂ ($f_0 = 0$)-Si



Fig. 6. (a) Simulated scattering spectrum of isolated, 3×3 finite number of unit cell and 5×5 finite number of unit cell for Si nanodisk-bulk WS₂ heterostructure with R = 130 nm, H = 70 nm, r = 50 nm, and $f_0 = 0$. (b) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of the isolated heterostructure. (c) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 3×3 finite number of unit cells for the heterostructure. (d) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure. (e) The electric-field enhancement distribution in the *x*-*y* plane at the resonance wavelength of 5×5 finite number of unit cells for the heterostructure.

nanodisk heterostructure. It is noted that the lowest total scattering spectrum is located at about 650 nm and FWHM decreases significantly with the increase of the number of the unit cell. In Figs. 6(b)-6(d), the field distribution and field vector diagram at the resonance wavelength of the finite number of unit cell is simulated. In Fig. 6(e), we plot the electric-field enhancement distribution in the x-y plane at the resonance wavelength of the heterostructure array with $\Lambda = 500$ nm and it is found that opposite circular displacement currents in the up and low hand sides of the disk generate a circular magnetic moment distribution that is perpendicular to the disk surface, which resemble the typical anapole-like pattern. **Research Article**



Fig. 7. (a) Simulated transmission spectrum for Si nanodisk-bulk WS_2 ($f_0 = 0$) heterostructure arrays with different periodicities. The near-field distribution of WS_2 ($f_0 = 0$)-Si nanodisk heterostructure arrays with periodicity of (b) 300 nm, (c) 400 nm, (d) 500 nm, (e) 600 nm, and (f) 700 nm at the position of the exciton absorption peak of around 656 nm.

2. Periodicity-Dependent Transmission Spectra and Near-Field Distributions

In Fig. 7, we plot the transmission spectrum and near-field distribution of WS₂ ($f_0 = 0$)-Si nanodisk heterostructure arrays with different periodicity size. It is noted that the transmission dip displays a prominent red shift with increasing of the periodicity, while the scattered powers of ED and TD moments of all systems always intersect at about 656 nm, which indicates that the resonance wavelength of anapole mode of the system is located near 656 nm. Moreover, the heterostructure array with a periodicity of 500 nm exhibits maximum electromagnetic field enhancement inside the bulk WS₂ based on the anapole mode and the resonance wavelength of D1 mode with a periodicity of 500 nm is near the position of the exciton absorption peak.

APPENDIX C: MULTIPOLES AND TOTAL SCATTERING POWER

In Fig. 8(a), we plot the normalized Cartesian multipole decomposition results for the Si nanodisk-bulk WS₂ heterostructure array, where the radius of the center WS_2 is 50 nm and the height is 70 nm, which is treated as background index-only material. It can be found that the scattered powers of ED and TD moments intersect at about 656 nm, while the transmission dip has a certain redshift compared to the intersection position of the ED and TD due to the array effect. To clearly identify the contributions to the far-field radiation, we calculate the scattered power of each multipole and the corresponding total scattered power in Fig. 8(b). It can be noted that a dip appears in the total scattered power spectrum when the scattered powers of the ED and TD are equivalent, which ultimately creates a weak anapole mode. Additionally, from Fig. 2(f), we can see that the displacement vector at the center cross section of the x-y plane forms two circles of opposite directions, which in turn generate a highly curved magnetic field and result in a state that closely matches the characteristics of anapole states. Therefore, strictly speaking, the anapole mode



Fig. 8. (a) Normalized corresponding transmission spectra and contributions from different multipoles for the Si nanodisk-bulk WS_2 ($f_0 = 0$) heterostructure array. (b) Total scattering power and contributions from different multipoles for the Si nanodisk-bulk WS_2 ($f_0 = 0$) heterostructure array.

should be located at 656 nm of the transmission spectrum, but since the near-field distribution of D1 mode in the transmission spectrum still satisfies the characteristics of the anapole mode, for the Si nanodisk-bulk WS₂ ($f_0 = 0$) heterostructure array with R = 130 nm, r = 50 nm, and h = 70 nm, the D1 mode can be referred to as the synergy between the weak anapole mode and the array effect.

APPENDIX D: RADIUS-DEPENDENT TRANSMISSION SPECTRA

In Fig. 9, we plot the transmission contour map of the Si nanodisk-bulk WS₂ ($f_0 = 0$) heterostructure array with respect to radii of Si nanodisks and wavelength of incident light.



Fig. 9. Simulated transmission contour map as a function of radius of the Si nanodisk (R = 100-200 nm) for the Si nanodisk-bulk WS₂ heterostructure array, where bulk WS₂ is treated as background index-only material.

APPENDIX E: RADIUS-DEPENDENT AND f-DEPENDENT OF THE WS₂ TRANSMISSION SPECTRA AND NEAR-FIELD DISTRIBUTION

In Fig. 10, the resonance coupling for the heterostructure system with different radius r of WS_2 is demonstrated. The splitting between the LB and UB hybrid modes is broadening with



Fig. 10. (a) Simulated transmission contour map as a function of radius of the WS₂ (r = 40-100 nm) for the Si nanodisk-bulk WS₂ heterostructure array, where bulk WS₂ is treated as background index-only material, the radius of Si nanodisk is R = 130 nm, and the height H = 70 nm. (b) Simulated transmission contour map as a function of radius of the WS₂ (r = 40-100 nm) for the Si nanodisk-bulk WS₂ heterostructure array, where the exciton of bulk WS₂ is considered, the radius of Si nanodisk is R = 130 nm, and the height H = 70 nm. The electric-field enhancement distribution in the x-y plane for an individual nanodisk of the heterostructure array with WS₂ radius (c) 20 nm, (d) 50 nm, and (e) 100 nm.



Fig. 11. (a) Simulated transmission contour map as a function of f for Si nanodisk-bulk WS₂ heterostructure array. (b) In-plane (Re[n]) dielectric functions of bulk WS₂ with different f.

the increasing of r. In Fig. 11, we discuss the strong coupling characteristics under different f values. The splitting between the resulted LB and UB hybrid anapole modes is broadening with the increasing of the oscillator strength.

APPENDIX F: RADIUS-DEPENDENT TRANSMISSION SPECTRA OF DIFFERENT POSITIONS OF WS₂

In Fig. 12, we study the coupling behavior when the bulk WS_2 is put outside the Si nanodisk. There exists no dip splitting in the transmission spectra while the WS_2 is placed on top of the Si nanodisk surface and a strong coupling regime with a smaller Rabi splitting can be achieved when the WS_2 is put on the side of the Si nanodisk.



Fig. 12. (a) Simulated transmission contour map as a function of radius of the Si nanodisk (R = 50-200 nm) for the Si nanodisk-bulk WS₂ heterostructure array, where bulk WS₂ is positioned on top of the Si nanodisk surface, the radius of WS₂ is r = 50 nm, and f_0 is set as 0.4. (b) Simulated transmission contour map as a function of radius of the Si nanodisk (R = 50-200 nm) for Si nanodisk-bulk WS₂ heterostructure array, where bulk WS₂ is positioned on the side surface of the Si nanodisk, the width of WS₂ is 50 nm, and f_0 is set as 0.4.

APPENDIX G: TRANSMISSION SPECTRA OF THE SI NANODISK-J-AGGREGATE HETEROSTRUCTURE ARRAY

In Fig. 13, we study the coupling characteristic of the Si nanodisk-J-Aggregate heterostructure array. It is found that the anapole mode supported by our perforated Si nanodisks can be strongly coupled to the J-Aggregate with a Rabi splitting of 200 meV.



Fig. 13. (a) Simulated transmission spectrum of the Si nanodisk array, where the radius of the Si nanodisk is R = 150 nm, the height is 70 nm, and the period is $\Lambda = 500$ nm. The inset shows the electric-field enhancement distribution of the anapole mode. (b) Simulated transmission contour map of the Si nanodisk-J-Aggregate heterostructure array as a function of radius of the Si nanodisk (R = 100-200 nm), where the height of the Si nanodisk H = 70 nm and the radius of the J-Aggregates is 40 nm.

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[†]These authors contributed equally to this work.

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