

Heterogeneous integrated phase modulator based on two-dimensional layered materials

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Silicon nitride, with ultralow propagation loss and a wide transparency window, offers an exciting platform to explore integrated photonic devices for various emerging applications. It is appealing to combine the intrinsic optical properties of two-dimensional layered materials with high-quality optical waveguides and resonators to achieve functional devices in a single chip. Here we demonstrate a micro-ring resonator-based phase modulator integrated with few-layer MoS₂. The ionic liquid is employed directly on the surface of MoS₂ to form a capacitor configuration. The effective index of the composite MoS₂-SiN waveguide can be modulated via adjusting bias voltages to achieve different charged doping induced electro-refractive responses in MoS₂ film. The maximum effective index modulation of the composite MoS₂-SiN waveguide can be achieved to 0.45×10^{-3} . The phase tuning efficiency is measured to be 29.42 pm/V, corresponding to a $V_{\pi}L$ of 0.69 V·cm. Since the micro-ring resonator is designed near the critical coupling regime, the coupling condition between the bus waveguide and micro-ring resonator can also be engineered from under-coupling to over-coupling regime during the charged doping process. That can be involved as a degree of freedom for the coupling tailoring. The ability to modulate the effective index with two-dimensional materials and the robust nature of the heterostructure integrated phase modulator could be useful for engineering reliable ultra-compact and low-power-consumption integrated photonic devices. © 2022 Chinese Laser Press

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1. INTRODUCTION

Optical phase control is a key functionality to modulate optical signals in photonic integrated circuits (PICs) for various applications. In particular, integrated phase modulator-based micro-ring resonators (MRRs) are becoming increasingly important due to their compact footprint, high Q -factor, and low cost [1–4]. In contrast to the traditional silicon on insulator (SOI), the silicon nitride (SiN) platform has stimulated great interest because of its ultralow propagation loss and broadband transparency window. Moreover, a thick SiN platform is expected for high optical confinement and dispersion engineering to enhance nonlinear optical properties [5,6]. The MRR-based on-chip modulator as a most important device for the optical link shows great potential in the field of high-volume optical signal processing [7–11]. The optical signal can be effectively controlled through engineering the effective index of the optical waveguide. State-of-the-art silicon modulators rely on the phase modulation through the plasma dispersion effect, either by

injection, accumulation, or depletion of free carrier to control the effective index of the waveguide. Despite being relatively fast and efficient, these devices suffer from a complex fabrication process and high insertion loss [12]. Alternative approaches are based on heterogeneous integration with materials such as III-V semiconductors, electro-optic organic layers, or epitaxial barium titanate (BTO) [13–17]. However, most of the above-mentioned solutions are not feasible using the SiN platform. Specifically, thanks to its insulating nature (~ 5 eV bandgap), plasma dispersion effects and many approaches based on integrated III-V semiconductors and organic layers, which rely on the conductivity of doped silicon waveguides, cannot be used in SiN-based photonic devices. Recently, the integration of various electro-refractive and electro-absorption materials such as graphene and transition metal dichalcogenides (TMDs) has opened additional routes toward performant SiN-based modulators, which promise to extend the performance beyond the limits set by the physical properties of SiN. For instance,

on-chip modulators integrated with graphene on the SiN platform have been demonstrated due to their high carrier mobility and broadband absorption ability [18–22]. While graphene exhibits various exceptional properties, the lack of an electronic bandgap due to its Dirac cone band structure limits its practical applications and encourages related research to explore alternative two-dimensional layered materials (2DLMs) with semiconducting characteristics [7,10]. As a member of 2DLMs, TMDs with fascinating properties such as layer-dependent bandgap, large exciton binding energies, and high optical nonlinearities have been widely studied on field effect transistors, photodetectors, and optical switches [23–32]. Among the TMDs, molybdenum disulfide (MoS_2) is a typical two-dimensional semiconductor material with a direct bandgap of ~ 1.8 eV for monolayer and a bulk indirect bandgap of ~ 1.3 eV. It shows the characteristics of electrically regulating the dielectric constant and the ability of broadband absorption ranging from visible to near-infrared (NIR) [33–37]. While previous studies have been devoted to investigating the photoelectric characteristics of TMDs based on-chip photonic devices, few studies have explored the electrical control of the optical response in few-layer TMDs on a thick SiN platform, which is highly desired for further applications.

In this work, we fabricate a composite MoS_2 -SiN waveguide integrated MRR phase modulator, and ionic liquid is employed to develop a capacitor configuration. The chemically assisted wet transfer method is used to transfer few-layer MoS_2 on the surface of the SiN waveguide. The MRR-based phase modulator shows a high tuning efficiency of 29.42 pm/V at 1549.8 nm and the maximum effective index modulation of 0.45×10^{-3} . The half-wave-voltage-length product ($V_\pi L$) is calculated to be 0.69 V·cm. Additionally, the influences of different types of charged doping (cations and anions) are systematically studied. Interestingly, the change of coupling states between the bus waveguide and MRR from under-coupling to over-coupling is observed in our experiment, which can be involved as a degree of freedom for the coupling tailoring. Such a large tunability in effective index arises from the strong

electro-refractive responses in few-layer MoS_2 film, which allows for the electrical control of the effective index of heterostructure integrated photonic devices and, thus, nonlinear optical responses for future on-chip optoelectronics.

2. RESULTS AND DISCUSSION

We deposited 3 μm thermally oxidized SiO_2 by plasma enhanced chemical vapor disposition (PECVD) as buried oxide on the 500 μm silicon substrate. The SiN layer with 780 nm thickness was then deposited on the SiO_2 by low pressure chemical vapor deposition (LPCVD). The typical CMOS process, such as electron beam lithography (EBL) and inductively coupled plasma-reactive ion etching (ICP-RIE), was used to pattern the SiN waveguide with 1800 nm \times 780 nm cross section (corresponding to width and height) for the bus waveguide and MRR and 350 nm \times 780 nm cross section for the inverse taper. We deposited Cr/Au (5 nm/50 nm) film as the metallic electrode. Few-layer MoS_2 was grown on the sapphire substrate by chemical vapor deposition (CVD). The poly(methyl methacrylate) (PMMA) was spin-coated on the surface of MoS_2 as auxiliary transfer material. The MoS_2 /PMMA stack was then transferred onto the SiN waveguide and one of the electrodes. After the transfer, the PMMA was completely removed by soaking in acetone. The MoS_2 used in our device was directly integrated with SiN waveguides and metallic electrode, which contributes to a low power consumption and strong interaction between light and materials. Finally, we introduced ionic liquid ($\text{DEME}^+ - \text{TFSI}^-$) cladded on the surface of the composite MoS_2 -SiN waveguide to form a capacitor configuration. The schematic of the composite MoS_2 -SiN waveguide integrated MRR phase modulator is shown in Fig. 1(a). The optical field is well confined in the SiN waveguide because of the large refractive index difference between the waveguide and external medium. The large waveguide width and excellent perpendicularity of the sidewall ($\sim 89.4^\circ$ in our experiment) can effectively reduce the instinct sidewall scattering loss caused by the fabrication process [38]. The coupling distance between the MRR

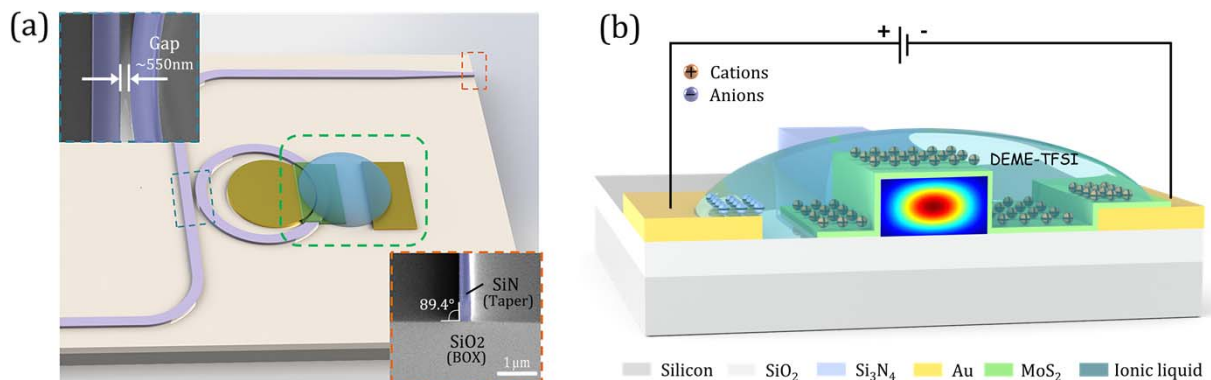


Fig. 1. (a) Schematic of a few-layer MoS_2 -based phase shifter. The part of the blue dotted square represents the coupling region between the bus waveguide and micro-ring resonator, corresponding to a coupling gap of ~ 550 nm. The part of the red dotted square shows the inverse taper waveguide with 350 nm \times 780 nm cross section and $\sim 89.4^\circ$ perpendicularity of the sidewall for a high coupling efficiency. The part of the green dotted square is the active modulation region. The detail illustration is shown in (b), which is the MoS_2 -SiN integrated structure cladded with ionic liquid ($\text{DEME}^+ - \text{TFSI}^-$) to form a capacitor configuration. The bias voltage is used to induce the accumulation of charged carriers at the interface of MoS_2 and then change its optical properties.

and bus waveguide is designed to ~ 550 nm, which makes our resonator near critical coupling regime. Figure 1(b) illustrates the composite MoS_2 -SiN waveguide cladded with the ionic liquid to achieve the charged doping process. Benefitting from the chemically assisted wet transfer technique, few-layer MoS_2 is uniformly cladded on the surface of the SiN waveguide by van der Waals force. Both sides of SiN waveguide are combined with inverse taper to enhance the coupling efficiency, which works as a mode transformer to transform the fiber mode to the waveguide mode. We coupled external light to the input port of the SiN waveguide using a tapered single-mode fiber that is then collected at the output port of the SiN waveguide using a similar tapered fiber. The transmission spectrum of the MRR under different bias voltages was recorded by an optical spectrometer (AQ63700, Yokogawa) to measure the phase shift and absorption change.

As shown in Fig. 2(a), the broadband linear absorption spectrum of few-layer MoS_2 film is measured ranging from 500 nm to 2000 nm. The A and B peaks around visible regime belong to the A and B exciton absorption of MoS_2 and indicate a high quality of MoS_2 film. The linear transmission of MoS_2 film at

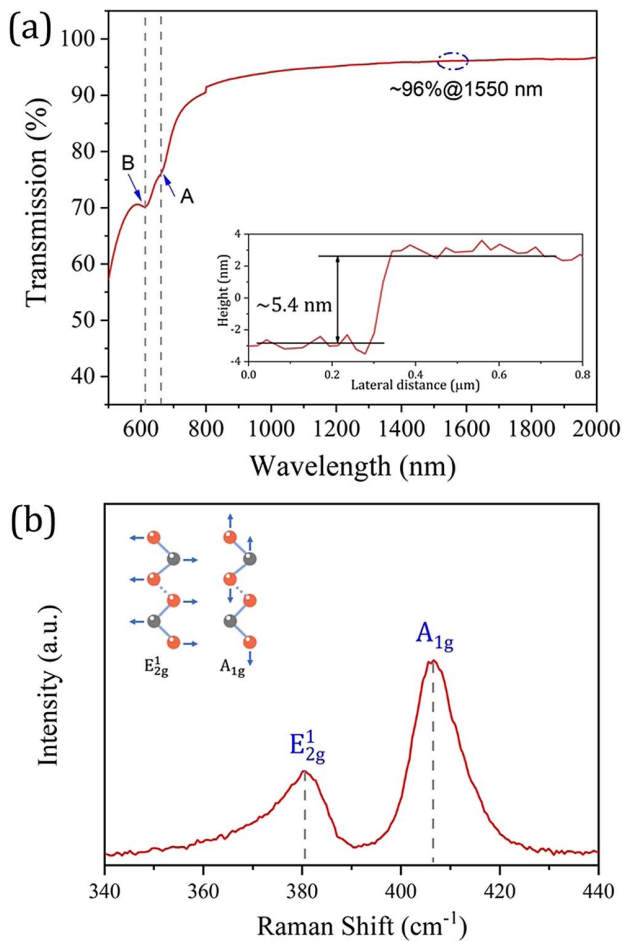


Fig. 2. (a) Broadband linear absorption spectrum of few-layer MoS_2 film. The inset image is the thickness of film measured by an AFM. (b) Raman spectrum of transferred few-layer MoS_2 , corresponding to the typical Raman vibrational modes.

1550 nm wave band is measured to be $\sim 96\%$. The small linear absorption contributes to decreasing the insertion loss of MoS_2 film when transferred on the surface of the resonator waveguide. The thickness of MoS_2 used in our experiment is measured to be ~ 5.4 nm [Fig. 2(a), inset] by an atomic force microscope (AFM), which indicates the number of layers is about 8 (considering ~ 0.65 nm thickness for single layer prepared by mechanical exfoliation method) [36]. The Raman spectrometer is used with 532 nm excitation wavelength to characterize the crystal quality of transferred MoS_2 film, and the results are shown in Fig. 2(b). The two peaks indicate the in-plane vibrational modes of the Mo and S atoms (E_{2g}^1) and the out-of-plane vibrational mode of S atoms (A_{1g}), respectively. The MoS_2 film with high crystal quality contributes to the light-matter interaction more efficiently and further improves the performance of heterostructure integrated photonic devices.

The bias voltage is applied across the two electrodes to achieve electrostatic doping of the MoS_2 film via ionic liquid ($\text{DEME}^+ - \text{TFSI}^-$). First, we set the electrode in direct contact with the MoS_2 film as the negative bias voltage, which results in the accumulation of cations at the interface of MoS_2 film. As shown in Fig. 3(a), the MRR has a resonance near 1549.8 nm. When the bias voltages are gradually increased, the resonance wavelength is redshifted with an offset of ~ 76.5 pm at 2.6 V, indicating a strong phase shift of the propagating mode in the MRR. Figure 3(b) shows that the resonance shift has a linear relationship with the applied voltage and the tuning efficiency is calculated to be 29.42 pm/V. According to the relationship of $V_{\pi}L = |\lambda_{\text{FSR}}L\Delta V/2\Delta\lambda|$, the half-wave-voltage length product $V_{\pi}L$ is calculated to be 0.69 V-cm, which is comparable to previous reported results [10,16]. Additionally, we also observe the regular intensity modulation in our MRR, which is mainly caused by the change in coupling conditions and the absorption modulation of few-layer MoS_2 and is discussed in detail in the following section. When the inversed bias voltage is applied, indicating the accumulation of anions at the interface of MoS_2 film [Fig. 4(a)], the normalized transmission response of the MRR is also redshifted with a linear relationship for the applied voltages, but the tuning efficiency is calculated to be as low as 10.35 pm/V at -2.6 V [Fig. 4(b)]. Interestingly, as shown in Fig. 5(a), when we continue to increase the bias voltage, the offset of the resonance wavelength loses its linearity at -2.7 V and tends to a similar saturation state. When the applied voltage is further changed from -2.7 V to -4 V, the maximum wavelength offset is measured to be ~ 7.3 pm relative to the resonance wavelength at -2.4 V [Fig. 5(b)]. The variation of the effective index at different bias voltages is extracted from the normalized transmission spectra [10]. When the interface of MoS_2 is cations doped, the effective index of the composite MoS_2 -SiN waveguide changes linearly during the voltage increase. The maximum offset of the effective index reaches 0.45×10^{-3} at a bias voltage of 3.8 V. When inversed bias voltages are applied, the effective index of the composite MoS_2 -SiN waveguide also changes linearly before -2.6 V with a maximum value of 0.12×10^{-3} . However, the effective index displays a saturable state when the bias voltage is further adjusted with a range from -2.7 V to -4 V.

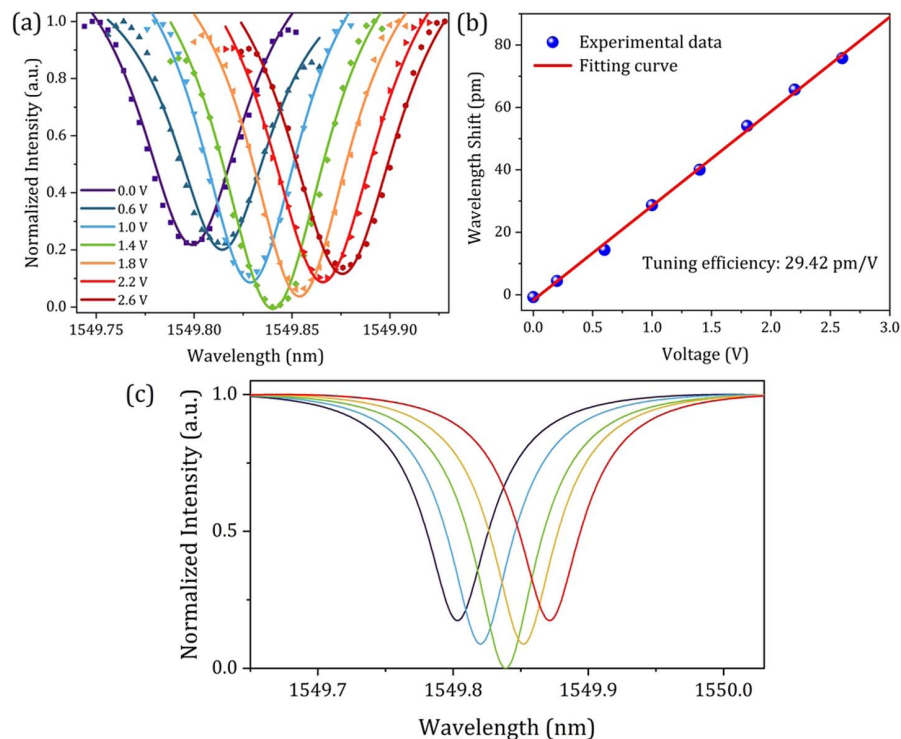


Fig. 3. (a) Normalized transmission response of the phase shifter as a function of applied bias voltage (cations doping on the interface of MoS₂ film). (b) Offset of resonance wavelength, corresponding to a tuning efficiency of 29.42 pm/V. (c) Simulation results of different coupling states for a micro-ring resonator.

Interestingly, besides the phase modulation, the output spectra under different types of charge doping also display regular intensity modulation, which is desired for enormous applications such as optical frequency combs and microwave photonics. Unlike the monolayer MoS₂ that is nearly transparent in the NIR band, few-layer MoS₂ has been demonstrated with a certain absorption capacity at NIR wavelengths [39,40]. Figure 6 shows the relationship between the effective index of the composite MoS₂-SiN waveguide and the applied voltages. The inset images are the energy-band diagram of few-layer MoS₂ under different types of charged doping. Note that few-layer MoS₂ used in our experiment is an n-type semiconductor, which is consistent with previously reported results [41,42]. As shown in Fig. 6, when MoS₂ film is charged-doped by cations under voltages ranging from 0 V to 3.8 V, the effective index of the composite MoS₂-SiN waveguide increases linearly with the increase of bias voltage, which redshifts the resonant wavelength. At the same time, because the Fermi level of MoS₂ film is adjusted to the conduction band, the absorption of light by the MoS₂ film becomes smaller. Finally, the coupling intensity of the MRR reaches the maximum, and the change of effective index is calculated to be $\sim 0.18 \times 10^{-3}$ at 1.4 V bias voltage shown in Fig. 3(a). When the bias voltage is further increased, because the absorption of the MoS₂ film has reached the minimum, the further increase of the effective index of the composite MoS₂-SiN waveguide causes the shift of the resonant wavelength and changes the critical coupling to the over-coupling, which leads the coupling intensity to decrease again. In contrast, since the density of states (DOS)

in the valence band of MoS₂ is less than that in the conduction band, the accumulated charge of the valence band is also smaller than that of conduction band [43–45]. As shown in Fig. 6, when the interface of MoS₂ film is doped with anions, the effective index of the composite MoS₂-SiN waveguide increases linearly with the increase of bias voltage in the range from 0 V to -1.4 V and redshifts the resonant wavelength. Since the Fermi level adjustment of the MoS₂ film tends to its valence band, the absorption induced from MoS₂ can be enhanced, thereby further introducing larger losses in the micro-ring cavity. The coupling state of the MRR remains at under-coupling. The coupling intensity of the MRR to the incident light is the lowest at the bias voltage ranging from -1.4 V to -1.8 V. However, when the voltage is further increased, the absorption due to the MoS₂ film has reached the maximum value. Meanwhile, the anion doping makes the effective index change of the composite MoS₂-SiN waveguide in the range from -2.4 V to -4 V approach saturation with a change amount of about 0.12×10^{-3} , which is less than the refractive index offset ($\sim 0.18 \times 10^{-3}$) of the same bias voltage by cation doping. Therefore, within this range, the MRR is always in an under-coupling state. Note that the coupling intensity of the MRR to incident light varies with a small change in the effective index of the composite MoS₂-SiN waveguide.

In Table 1, we compare the performance of the present hybrid integrated device with the state-of-the-art, including silicon-graphene-hybrid MZMs, silicon-LiNbO₃-hybrid MZMs, silicon-organic-hybrid MZMs, SiN-PZT-hybrid MRRs, SiN-LiNbO₃-hybrid MRRs, and SiN-TMD-hybrid MRRs.

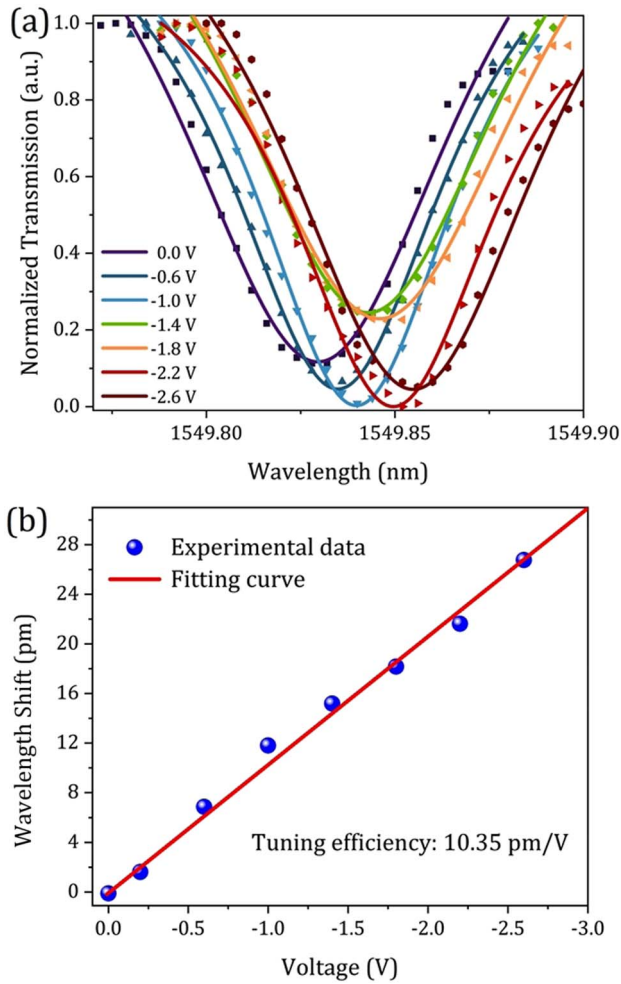


Fig. 4. (a) Normalized transmission response of a phase shifter as a function of applied bias voltage (anion doping on the surface of MoS₂ film). (b) The offset of resonance wavelength at different voltages, which shows a lower tuning efficiency than that of cation doping.

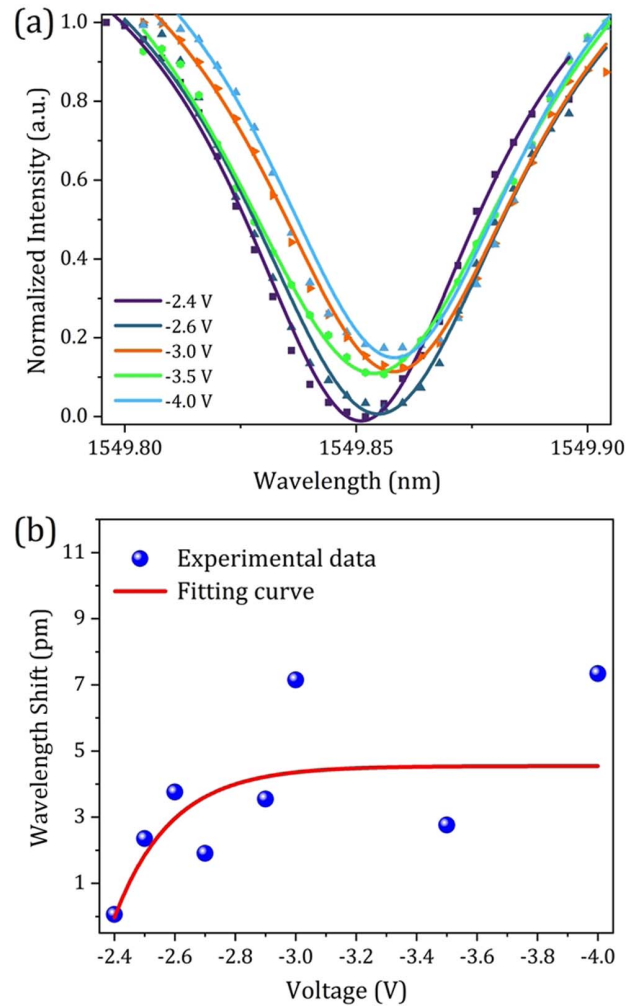


Fig. 5. (a) Normalized transmission response of a phase shifter at high bias voltages. (b) The offset of resonance wavelength loses its linearity at -2.7 V and tends to a similar saturation state when the voltage is further increased.

Here, we focus on the cross section of waveguide and the phase tuning efficiency of each device, which are key merits to represent the propagation loss and modulation efficiency. As shown in Table 1, our device is the only one to achieve low $V_{\pi}L$ at a thick SiN platform. In particular, a thick SiN waveguide possesses a lower propagation loss (typical ~ 0.1 dB/cm) than that of a thin SiN waveguide, which requires a SiO₂ cladding layer to reduce the insertion loss. In addition, a thick SiN waveguide enables us to achieve dispersion engineering, which can be used to integrate with other functional devices such as Kerr optical frequency combs and tunable delay lines at a single chip. The demonstrated $V_{\pi}L$ of our device is also promising, which is only surpassed by the silicon-graphene-hybrid MZMs with a much higher waveguide propagation loss and electrical power consumption. Our proposed device has a simple processing flow, which is more suitable for large-scale photonic systems with low electrical power dissipation and high performance. Moreover, in order to further reduce the insertion loss introduced to the MRRs after MoS₂ transfer, a CMOS-compatible patterning process can be used to precisely

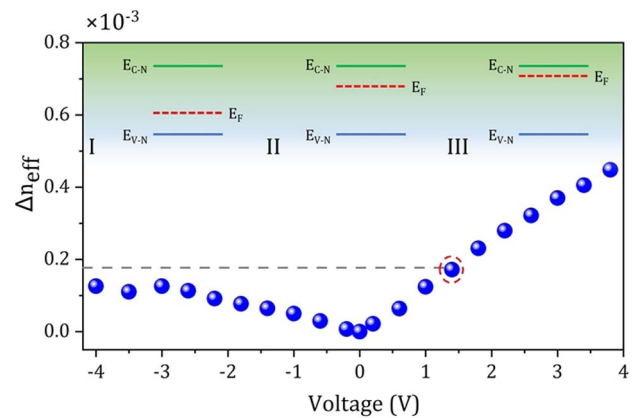


Fig. 6. The effective index of the composite MoS₂-SiN waveguide is exacted from the tuned resonance and plotted as a voltage dependent function. The inset images are the energy-band diagram of few-layer MoS₂ under different types of charged doping, where E_{C-N} and E_{V-N} are the conduction band edge and valence band edge of n-type MoS₂ and E_F is the Fermi energy level.

Table 1. Comparison of Phase Tuning Efficiency for the Hybrid Integrated Phase Modulator^a

Platform	Structure	Method	Cross Section of Waveguide	Insertion Loss (dB)	$V_{\pi}L$ (V·cm)	Refs.
SOI	MZM	Graphene–silicon	480 nm × 220 nm	/	0.28	[9]
SOI	MZM	LiNbO ₃ –silicon	500 nm × 220 nm	2.5	2.2	[46]
SOI	MZM	Organic–silicon	550 nm × 220 nm	5.4	1.4	[47]
SOI	MZM	ITO–silicon	500 nm × 220 nm	6.7	0.0095	[48]
SiN	MRR	PZT–SiN	1000 nm × 330 nm	/	3.3	[49]
SiN	Racetrack	LiNbO ₃ –SiN	1200 nm × 200 nm	13	5.1	[15]
SiN	MZM	MoS ₂ –SiN	1300 nm × 330 nm	~6.6	1.7	[10]
	MRR	WS ₂ –SiN	1300 nm × 330 nm	/	0.8	
SiN	MRR	MoS ₂ –SiN	1800 nm × 780 nm	3.2	0.69	This work

^aMZM, Mach–Zehnder modulator; PZT, piezoelectric lead zirconate titanate; Racetrack, racetrack resonator.

pattern the shape of the transferred material and remove excess material.

3. CONCLUSION

We have experimentally demonstrated a MoS₂–SiN heterostructure integrated MRR phase shifter with electrostatically doping ionic liquid. The chemically assisted wet transfer method is used to uniformly clad few-layer MoS₂ on the surface of the SiN waveguide. The effective index of the composite MoS₂–SiN waveguide is studied via adjusting bias voltages to achieve different charged doping induced electro-refractive responses in MoS₂ film. The experimental results demonstrate that our phase shifter has a phase tuning efficiency reaching 29.42 pm/V and $V_{\pi}L$ is calculated to be 0.69 V·cm. The maximum change in the effective index with cation doping MoS₂ film is measured to be 0.45×10^{-3} at 3.8 V. Additionally, when applying the anion doping to the interface of MoS₂ film, the maximum change in the effective index of composite MoS₂–SiN is about 0.12×10^{-3} at –3.8 V bias voltage, and the saturation phenomenon is also observed. Since the resonator is designed near the critical coupling regime, during the charged doping process, the coupling condition between the bus waveguide and MRR can be engineered from under-coupling to over-coupling. That can be involved as a degree of freedom for the coupling tailoring. The proposed phase shifter presents a facile scheme to be hybrid with 2DLMs and to be realized on a thick SiN platform, which is promising for compact MRR structures and might be used as a promising phase modulation scheme for large-scale optical links with ultralow loss and power consumption.

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Data Availability. Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

[†]These authors contributed equally to this paper.

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