PHOTONICS Research

Nonlinear optical properties of CsPbCl_xBr_{3-x} nanocrystals embedded glass

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All-inorganic perovskite has attracted significant attention due to its excellent nonlinear optical characteristics. Stable and low-toxic perovskite materials have great application prospects in optoelectronic devices. Here, we study the nonlinear optical properties of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) nanocrystals (NCs) glass by open-aperture Z-scan. It is found that the two- (2PA) and three-photon absorption (3PA) intensity can be adjusted by the treatment temperature and the ratio of halide anions. The perovskite NCs glass treated at a high temperature has better crystallinity, resulting in stronger nonlinear absorption performance. In addition, the value of the 2PA parameter of CsPbCl_{1.5}Br_{1.5} NCs glasses decreases when the incident pump intensity increases, which is ascribed to the saturation of 2PA and population inversion. Finally, the research results show that the 2PA coefficient (0.127 cm GW⁻¹) and 3PA coefficient (1.21 × 10⁻⁵ cm³ GW⁻²) of CsPbCl₁Br₂ NCs glass with high Br anion content are larger than those of CsPbCl₂Br₁ and CsPbCl_{1.5}Br_{1.5} NCs glasses. This is mainly due to the greater influence of Br anions on the symmetry of the perovskite structure, which leads to the redistribution of delocalized electrons. The revealed adjustable nonlinear optical properties of perovskite NCs glass are essential for developing stable and high-performance nonlinear optical devices.

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1. INTRODUCTION

All-inorganic perovskite has received widespread attention recently, owing to its tunable light-emitting bandgap [1,2], large exciton binding energy [3], and other excellent photoelectric properties [4,5]. Hence, perovskite materials are mostly used in light-emitting diodes (LEDs) [6,7], solar cells [8], photodetectors [9,10], lasers [4,11], and other devices. Halide perovskite nanocrystals have become one of the most promising optoelectronic materials due to their low-cost and easy synthesis [1]. Compared with bulk and layered materials, the specific surface area of CsPbCl_xBr_{3-x} nanocrystals (NCs) is greatly increased, thereby enhancing the NC optical performance [1,2,12]. Due to the strong multiphoton absorption (MPA) characteristics of halide perovskite NCs, they are very promising as a material for the development of multiphoton pump lasers [13]. The spherical NCs not only overcome the difficulty of large area growth of thin films, but also can combine with a variety of substrates or solutions for incorporation into optoelectronic devices [14].

Owing to the poor stability of bare perovskite and toxicity of lead halide perovskite, the application of perovskite materials in optoelectronic devices is greatly restricted [15,16]. So far, diverse approaches to improve the stability of perovskite materials have been reported, such as bonding of the organic ligands [17], establishing core/shell nanostructure [18], Mn-doping [19], silica coating [20,21], and infiltrating CsPbX₃ NCs into mesoporous matrices [22]. However, combining the thermal, chemical, and mechanical stability of the glass has been proved an effective method to improve the stability of perovskite NCs [23–25]. Hu et al. reported the well-designed arrangement of CsPbBr3 NCs glasses with reduced self-absorption emission and enhanced the quantum efficiency of solar cells [26]. Ye et al. investigated the versatile precipitation of CsPbX₃ NCs glasses, in which photoluminescence (PL) covering the whole visible range with high efficiency could be achieved [24].

Meanwhile, some researches indicated that the relative PL intensity of CsPbBr₃ quantum dots glass is still ~85% ~ 90% after being immersed in water for 120 h or exposed to UV light for 100 h, and even about 60% of PL intensity still remained for storage up to 45 days [27,28]. All these previous studies reveal that embedding CsPbX₃ quantum dots or NCs into glass has a great potential in improving its stability and fluorescence performance [26–28].

However, the nonlinear optical aspect of perovskite glass, especially the study of MPA, which is important to the application of these optoelectronic devices, has been rarely investigated [13,29,30]. Many studies have been reported on the nonlinear optical properties of pure perovskite [31]. For example, a two-photon absorption (2PA) cross section of CsPbBr₃ NCs in toluene as high as 10^6 GW has been reported [32]. Chen et al. demonstrated that the 2PA cross section of CsPbBr₃ NCs depends on the particle size [33]. Furthermore, due to the symmetry breaking of the perovskite octahedron structure, Li et al. confirmed that all-inorganic perovskites with different proportions of halogen atoms show greater MPA intensity than CsPbCl₃ and specially designed organic molecules [34,35]. Not only that, Chen et al. reported that the five-photon absorption cross section of Type-I core-shell halide perovskite NCs is 9 orders of magnitude higher than that of specially designed organic molecules [30]. However, the nonlinear optical aspect of perovskite glass, which is highly desired for applications in low-threshold lasing [36], optical data storage [37], photodetectors [38], and other nonlinear optical photoelectric devices [32], has been rarely investigated.

In the present work, the three-order nonlinear optical characteristics of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses have been investigated by open-aperture (OA) Z-scan measurements using femtosecond laser pulses. We observe the 2PA and threephoton absorption (3PA) phenomena of CsPbCl_xBr_{3-x} (x =1, 1.5, 2) NCs glasses at wavelengths of 800 and 1300 nm. The magnitude of the 2PA coefficient (β) of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses is about 10⁻¹-10⁻² cm GW⁻¹, and the magnitude of the 3PA coefficient (γ) is 10⁻⁵-10⁻⁶ cm³ GW⁻². The observed MPA behavior of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses could play a great role in the development of perovskite-based optoelectronic devices.

2. EXPERIMENT SETUP

In the OA Z-scan measurement, as shown as Fig. 1, the perovskite NCs glasses with thickness L are moved along the Gaussian light propagation direction, and the laser beam transmittance after the sample is measured. For 2PA measurement, we used a Ti:sapphire femtosecond laser (Spectra-Physics) operating at a central wavelength of 800 nm with a repetition frequency of 1 kHz and a pulse width of 35 fs. For 3PA measurements, the laser source consisted of an optical parametric amplifier (TOPAS-Prime), delivering a wavelength of 1300 nm with a pulse width of 200 fs. The repetition frequency of laser was 1 kHz. In order to amplify weak signals, a chopper (Thorlabs MC2000B) is inserted into the optical path. The focal length of the front lens of the sample is 10 cm, and the spot radius at the focal point is about 30 μ m. The transmitted intensity of each pulse after passing through the sample is measured by a Si-biased detector (Thorlabs DET10A2) using the lock-in amplifier (Signal Recovery Model 7270) technique.

3. RESULTS AND DISCUSSION

In this work, CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs are successfully embedded inside a glass sheet (radius = 10 mm, thickness = 0.56 mm) matrix via the melt-quenching and *in situ* crystallization method. The CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs are spherical structures with a diameter of ~35–45 nm [see the Appendix A, Fig. 6(a)] [39]. Figure 2(a) shows the X-ray diffraction (XRD) patterns of CsPbCl_{1.5}Br_{1.5} NCs glass under different treatment temperatures. The diffraction peaks of perovskite glass are at 15.5°, 22°, 31°, 38.5°, and 44.6°, corresponding to (100), (110), (200), (211), and (220) phase, respectively [40,41]. Referring to the previous research results, these narrow diffraction peaks demonstrate that the CsPbCl_{1.5}Br_{1.5} NCs have good crystallization [25]. Figure 2(b) displays the PL emission spectra of CsPbCl_{1.5}Br_{1.5} NCs glass. The PL emission peak of



Fig. 2. (a) XRD patterns and (b) PL emission spectra of $CsPbCl_{1.5}Br_{1.5}$ NCs glasses under different treatment temperature excited by femtosecond pulses at 365 nm.



Fig. 1. Experimental setup for the Z-scan technique.



Fig. 3. OA Z-scan results of CsPbCl_{1.5}Br_{1.5} NCs glasses with different treatment temperatures. (a) Under pump intensity 25.5 GW/cm² at a wavelength of 800 nm and (b) under pump intensity 217 GW/cm² at a wavelength of 1300 nm. (The balls are the experimental data, and the solid lines are fitting curves.) The fitting results of (c) β and (d) γ of CsPbCl_{1.5}Br_{1.5} NCs glass with different treatment temperatures. (Inset, the schematic of a two-level model.)

CsPbCl_{1.5}Br_{1.5} NCs glass displays a slight redshift with the treatment temperature from 470°C to 530°C. The reason for the slight redshift of PL emission peak in the CsPbCl_{1.5}Br_{1.5} NCs glass is the increase of the crystal grains size, which is caused by the increase of the heat treatment temperature [41]. However, the surface defects increase due to the continuous increase in temperature, which affects the fluorescence quantum yield and causes the PL emission intensity of the CsPbCl_{1.5}Br_{1.5} NCs glass to decrease [41].

Figures 3(a) and 3(b) show the OA Z-scan curves of the CsPbCl_{1.5}Br_{1.5} NCs glass (bandgap ~ 2.58 eV) at the different treatment temperature excited by 800 nm (~1.55 eV) with pump intensity of 25.5 GW/cm² and by 1300 nm (~0.95 eV) with pump intensity of 217 GW/cm², respectively (see the Appendix A, Fig. 8). As shown in Figs. 3(a) and 3(b), the Z-scan curves are all valley shapes. When the CsPbCl_{1.5}Br_{1.5} NCs glass is close to the focus, the normalized transmission of the incident laser decreases. As shown in Figs. 3(a) and 3(b), the CsPbCl_{1.5}Br_{1.5} NCs glass with a heat treatment temperature of 530°C has the strongest nonlinear response.

In theory, the measured normalized transmission (T) for OA Z-scan results is given by the expression [42,43]

$$T_{\text{OA}(n\text{PA})} = \frac{1}{\{1 + (n-1)\alpha_{\text{NL}}L_{\text{eff}}\{I_0/[1 + (z/z_0)^2]\}^{n-1}\}^{1/n-1}}$$

(n = 1, 2, 3), (1)

where $L_{\rm eff}$ is the effective length of the sample. $\alpha_{\rm NL}$ is the nonlinear optical coefficient. z is the position of the sample in the light path, $z_0 = \pi \omega_0^2 / \lambda$ is the Rayleigh range of the Gaussian beam, and ω_0 is the beam waist at the focal point (z = 0).

Among them, the 2PA coefficient is represented by β . The imaginary part of third-order nonlinear susceptibility [44,45],

$$\mathrm{Im}\chi^{(3)} = \frac{c^2 \varepsilon_0 n_0^2 \beta}{\omega},$$
 (2)

where *c* is the speed of light, n_0 is the linear refractive index, and ε_0 and ω are the vacuum permittivity and angular frequency of the laser beam, respectively. The figures of merit (FOMs) are used to describe nonlinear absorption characteristics: FOM = $|\text{Im}\chi^{(3)}/\alpha_0|$, where α_0 is the linear absorption coefficient. The 3PA coefficient is represented by γ . The effective thicknesses of 2PA and 3PA are $L_{\text{eff}} = (1 - e^{-\alpha_0 L})/\alpha_0$ and $L'_{\text{eff}} = (1 - e^{-2\alpha_0 L})/2\alpha_0$, respectively [42,43].

Figures 3(c) and 3(d) display the 2PA and 3PA coefficients of CsPbCl_{1.5}Br_{1.5} NCs glasses at the different treatment temperatures obtained by fitting Eq. (1). The insets in Figs. 3(c) and 3(d) show the schematic diagram of the 2PA and 3PA processes, respectively. When CsPbCl_{1.5}Br_{1.5} NCs glass is excited by the femtosecond laser, electrons in the valence band need to absorb two (or three) photons at the same time to transition to the conduction band. By fitting with a Z-scan theory, 2PA coefficients of CsPbCl_{1.5}Br_{1.5} NCs glass with different treatment temperatures were calculated: $\beta = 0.87$ cm GW⁻¹ for 470°C treatment temperature, $\beta = 0.97$ cm GW⁻¹ for 500°C treatment temperature, and $\beta = 1.23$ cm GW⁻¹ for 530°C treatment temperature, respectively. Under the pump intensity of 25 GW/cm², the Im $\chi^{(3)}$ for CsPbCl_{1.5}Br_{1.5} NCs glasses are in the range of $(1.99-2.81) \times 10^3$ esu and the magnitudes of FOM are all at 10^3 esu cm obtained by fitting Eq. (2) (see the Appendix A, Table 1). 3PA coefficients of CsPbCl_{1.5}Br_{1.5} NCs glasses with different treatment temperatures were calculated: $\gamma = 2 \times 10^{-5}$ cm³ GW⁻² for 470°C, $\gamma = 2.17 \times$ 10^{-5} cm³ GW⁻² for 500°C, and $\gamma = 2.86 \times 10^{-5}$ cm³ GW⁻² for 530°C, respectively. As the processing temperature increases, the crystallinity is better. The larger the nanocrystal particles, the stronger the 2PA and 3PA performance of the CsPbCl_{1.5}Br_{1.5} NCs glasses [39].

Furthermore, the nonlinear properties of CsPbCl_{1.5}Br_{1.5} NCs glasses with different pump intensities are measured. The OA Z-scan curves of CsPbCl_{1.5}Br_{1.5} NCs glass under 500°C treatment temperature with various incident pump intensities at the wavelength of 800 nm are shown in Fig. 4(a). The normalized transmittance decreases when either increasing the pump intensity or placing the CsPbCl_{1.5}Br_{1.5} NCs glass closer to the focus point, z = 0, while as the pump intensity



Fig. 4. (a) OA Z-scan curves and (b) corresponding fitting results of β (black ball), Im $\chi^{(3)}$ (pink), and FOM (blue) of the CsPbCl_{1.5}Br_{1.5} NCs glass at the wavelength of 800 nm with different incident pump intensity.

increases, the normalized transmittance curve deepens. These results suggest the potential application of CsPbCl₁₅Br₁₅ NCs glass in optoelectronic devices, such as optical limiting devices [46]. Figure 4(b) summarizes the dependence of 2PA fitting results as a function of pump intensity. (Fitting results are summarized in Table 2). The β is 0.096 cm/GW of CsPbCl_{1.5}Br_{1.5} NCs glass at the pump intensity of 255 GW/cm² and 0.089 cm/GW at the pump intensity of 332 GW/cm², respectively. The result of the 2PA coefficient we obtained is with the same order of magnitude as the CsPbBr₃ NCs [36] and CsPbCl_xBr_{3-x} (x = 1, 2) quantum dots [47]. The perovskite NCs encapsulated in perovskite glass are isolated from the external environment. This gives them better stability and greatly improves the service life of perovskite NCs. The Im $\chi^{(3)}$ for CsPbCl_{1.5}Br_{1.5} NCs glasses are in the range of $(2.03-2.88) \times 10^2$ esu and the FOM is in the range of $(2.3-3.2) \times 10^2$ esu cm. It is shown that as the pump intensity increased, the value of β , Im $\chi^{(3)}$, and FOM decreased. The same phenomenon has been observed in other materials, such as PbS/glue nanocomposite [48] and few-layer WS₂ films [49]. As the incident light energy increases, a large number of carriers gather in the excited state, resulting in population inversion [50]. Hence, the electrons in the ground state cannot further absorb the photon and transition to the excited state [51]. The β of CsPbCl_{1.5}Br_{1.5} NCs decreases as the pump intensity increases.

To further explore the influence of different doped halogen anion ratios on the MPA and consider the influence of treatment temperature on the PL performance of all-inorganic perovskite glasses, we choose CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses with a treatment temperature of 500°C to explore their nonlinear response [52]. Figures 5(a) and 5(b) display the OA Z-scan response of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses under femtosecond laser of 800 nm and 1300 nm, respectively.

Figure 5(a) shows the OA Z-scan curves of $CsPbCl_xBr_{3-x}$ (x = 1, 1.5, 2) NCs glasses under pump intensity 178 GW/cm^2 at a wavelength of 800 nm. Comparing the normalized transmittance curves of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses with different Cl⁻ and Br⁻ ion ratios, it is found that the CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses with higher Br- ions doping ratio have a greater nonlinear response in the process of 2PA and 3PA. The bandgap width of CsPbCl₁Br₂ NCs glasses is 2.46 eV, slightly smaller than that of CsPbCl_{1.5}Br_{1.5} (2.58 eV) and CsPbCl₂Br₁ (2.7 eV) (see the Appendix A, Fig. 8). Due to the increase of Br^{-} ion content, the bandgap is narrowed, thereby promoting the carrier transition rate [34,53]. CsPbCl₁Br₂ NCs glasses exhibit a large MPA phenomenon. Correspondingly, the PL emission peak of CsPbCl₂Br₁, CsPbCl₁₅Br₁₅, and CsPbCl₁Br₂ NCs glasses with emission wavelength peaks at 454, 470, and 490 nm is shown in Fig. 5(c), respectively. The PL peak shifts to the lower energy direction as the proportion of Br- ions increases. It is shown that the luminescence can be effectively tuned by introducing the Cl⁻ and Br⁻ ion ratios. The width of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses at half-height (FWHM) of the PL emission is less than 30 nm (see the Appendix A, Fig. 7).

By fitting the Z-scan data in Figs. 5(a) and 5(b), the β of CsPbCl₁Br₂ NCs glass is ~0.087 cm GW⁻¹, 0.1 cm GW⁻¹



Fig. 5. OA Z-scan results of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses. (a) Under pump intensity 178 GW/cm² at a wavelength of 800 nm and (b) under pump intensity 535 GW/cm² at a wavelength of 1300 nm. (The balls are the experimental data, and the solid lines are fitting curves.) (c) PL emission spectra of CsPbCl_xBr_{3-x} (x = 1, 1.5, 2) NCs glasses for 500°C treatment temperature excited by femtosecond pulses at 365 nm; (d) fitting the results of 2PA coefficient (β , purple bar) and 3PA coefficient (γ , red bar) obtained in (a) and (b), respectively.

for $CsPbCl_{1.5}Br_{1.5}$ NCs glass, and 0.127 cm GW⁻¹ for CsPbCl₂Br₁ NCs glass. The 2PA coefficient of CsPbCl₁Br₂ NCs glass is 1 order of magnitude higher than that of CsPbCl₁Br₂ quantum dots [47]. Combined with Eq. (2), the Im $\chi^{(3)}$ for CsPbCl_xBr_{3-x} NCs glasses is in a range of $(1.99-2.9) \times 10^2$ esu, which is larger than that of CsPbBr₃ NC (see the Appendix A). The FOM value used to describe the nonlinear absorption characteristics is in the range of $(2.2-2.53) \times 10^2$ esu cm (see the Appendix A, Table 3). Meanwhile, we obtained the γ : 0.54×10^{-5} cm³ GW⁻² is for $CsPbCl_2Br_1$ NCs glass, 0.82×10^{-5} cm³ GW⁻² is for CsPbCl_{1.5}Br_{1.5} NCs glass, and 1.21×10^{-5} cm³ GW⁻² is for CsPbCl₁Br₂ NCs glass. The 3PA coefficient of CsPbCl₁Br₂ NCs glass is over 1 order of magnitude larger than that of the CsPbCl_{1.5}Br_{1.5} and CsPbCl₂Br₁ NCs glasses. This is mainly because the CsPbCl₁Br₂ NCs glass has a narrower bandgap and higher structural destabilization that will lead to the easier carrier transition and delocalized electrons redistribution compared with CsPbCl_{1.5}Br_{1.5} and CsPbCl₂Br₁ NCs



Fig. 6. (a) Transmission electron microscopy (TEM) images of $CsPbCl_xBr_{3-x}$ (x = 1, 1.5, 2) NCs and (b) high-resolution TEM images of $CsPbCl_{1.5}Br_{1.5}$ NCs.



Fig. 7. (a) Amplified spontaneous emission (ASE) measurement on $CsPbCl_{1.5}Br_{1.5} NCs$ glass under an 800 nm pulsed laser at room temperature and (b) corresponding full-width at half-maxima (FWHM) and output as a function of incident pump intensity.

glasses [24]. Due to the narrower bandgap and structural destabilization of the perovskite, the electron cloud is distorted, which promotes the transition of electrons from the ground state to the excited state [34]. Therefore, the carrier transition rate is further increased.

4. CONCLUSION

In summary, we measure the 2PA and 3PA properties of $C_sPbCl_xBr_{3-x}$ (x = 1, 1.5, 2) NCs glasses using the OA Z-scan method. The $CsPbCl_{1.5}Br_{1.5}$ NCs glass under 530°C treatment temperature exhibited the strongest 2PA and 3PA coefficients, which is mainly due to the better crystallization properties of perovskite NCs at higher temperatures. Furthermore, the



Fig. 8. $(\alpha hv)^2 - hv$ plot of CsPbCl_xBr_{3-x} (x = 1, 2, 3) NCs glass.

dependence of incident pump intensity shows that the nonlinear coefficient decreases when the incident pump intensity increases. We also found that the larger the proportion of Br anions, the stronger the MPA performance. These results for CsPbCl_xBr_{3-x} NCs glass may guide designs for their potential use in applications.

APPENDIX A

The CsPbCl₂Br_{3-x} (x = 1, 1.5, 2) nanocrystals are spherical structures with a diameter of ~35–45 nm enclosed in a glass sheet, as shown in Fig. 6(a). As shown as Fig. 9, CsPbCl₁Br₂ and CsPbCl₂Br₁ NCs glasses also showed the 2PA phenomenon that the normalized transmittance decreased greatly as the pump power increased under the wavelength of 800 nm. As the pump energy becomes stronger, more carriers are excited,

Table 2. 2PA Parameters of the CsPbCl_{1.5}Br_{1.5} NCs Glass Measured by an OA Z-Scan at 800 nm under Different Pump Intensity^a

Pump Intensi	ity		
(GW/cm ²)	β (cm GW ⁻¹)	$Im\chi^{(3)}$ (esu)	FOM (esu cm)
102	0.13 ± 0.0064	2.88×10^{2}	3.2×10^{2}
178	0.1 ± 0.0042	2.28×10^{2}	2.54×10^{2}
255	0.096 ± 0.0043	2.19×10^{2}	2.43×10^{2}
332	0.089 ± 0.0052	2.03×10^{2}	2.3×10^{2}

^{*a*}β, 2PA coefficient; $Im\chi^{(3)}$, imaginary part of third-order nonlinear susceptibility; FOM, figures of merit.

Table 1. Summary of the Measured 2PA and 3PA Parameters of CsPbCl_{1.5}Br_{1.5} NCs Glasses at Wavelengths of 800 nm and 1300 nm^a

Sample	Growth Condition	Т	$\alpha_0 \ (\mathrm{cm}^{-1})$	β (cm GW ⁻¹)	$\gamma ~(\times 10^{-5} ~{\rm cm}^3 ~{\rm GW}^{-2})$	$Im\chi^{(3)}$ (esu)	FOM (esu cm)
CsPbCl ₁₅ Br ₁₅	470°C	0.94	0.48	0.87 ± 0.015	2 ± 0.077	1.99×10^{3}	4.14×10^{3}
CsPbCl ₁₅ Br ₁₅	500°C	0.90	0.81	0.97 ± 0.016	2.17 ± 0.058	2.21×10^{3}	2.73×10^{3}
CsPbCl _{1.5} Br _{1.5}	530°C	0.88	0.99	1.23 ± 0.028	2.86 ± 0.031	2.81×10^{3}	2.84×10^{3}

"T, linear transmittance of perovskite glasses; α_0 , linear absorption coefficient; β , 2PA coefficient; γ , 3PA coefficient; $\text{Im}\chi^{(3)}$, imaginary part of third-order nonlinear susceptibility; FOM, figures of merit.

Table 3. Summary of the Measured 2PA and 3PA Parameters of CsPbCl_xBr_{3-x} NCs Glasses at Wavelengths of 800 nm and 1300 nm^a

Sample	Growth Condition	Т	$\alpha_0 \ (\mathrm{cm}^{-1})$	β (×10 ⁻² cm GW ⁻¹)	$\gamma ~(\times 10^{-5}~cm^3~GW^{-2})$	Im $\chi^{(3)}$ (esu)	FOM (esu cm)
CsPbCl ₂ Br ₁	500°C	0.89	0.9	8.7 ± 0.42	0.54 ± 0.017	1.99×10^{2}	2.2×10^{2}
CsPbCl _{1.5} Br _{1.5}	500°C	0.89	0.9	10 ± 0.59	0.82 ± 0.02	2.28×10^2	2.53×10^2
CsPbCl ₁ Br ₂	500°C	0.85	1.26	12.7 ± 0.52	1.21 ± 0.02	2.9×10^2	2.3×10^{2}

"T, linear transmittance of perovskite glasses; α_0 , linear absorption coefficient; β , 2PA coefficient; γ , 3PA coefficient; $\text{Im}\chi^{(3)}$, imaginary part of third-order nonlinear susceptibility; FOM, figures of merit.



Fig. 9. Open-aperture Z-scan curves of the (a) $CsPbCl_1Br_2$ and (b) $CsPbCl_2Br_1$ NCs glasses at the wavelength of 800 nm with different incident pump intensity.

thereby absorbing more photons. This performance can be applied to optoelectronic devices such as optical limiting.

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