PHOTONICS Research

In-depth investigation and applications of novel silicon photonics microstructures supporting optical vorticity and waveguiding for ultranarrowband near-infrared perfect absorption

Roy Avrahamy,¹ Moshe Zohar,² Mark Auslender,^{1,*} Benny Milgrom,³ Shlomo Hava,^{1,2} and Rafi Shikler^{1,4}

¹School of Electrical and Computer Engineering, Ben-Gurion University of the Negev, Beer-Sheva 84105, Israel ²Electrical and Electronics Engineering Department, Shamoon College of Engineering, Beer-Sheva 84100, Israel ³School of Electrical Engineering, The Jerusalem College of Technology, Jerusalem 91160, Israel ⁴Ilse Katz Institute for Nanoscale Science & Technology, Ben-Gurion University of the Negev, Beer-Sheva 84105, Israel *Corresponding author: marka@bgu.ac.il

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We propose a novel concept of designing silicon photonics metamaterials for perfect near-infrared light absorption. The study's emphasis is an in-depth investigation of various physical mechanisms behind the ~100% ultranarrowband record peak absorptance of the designed structures, comprising an ultrathin silicon absorber. The electromagnetic power transport, described by the Poynting vector, is innovatively explored, which shows combined vortex and crossed-junction two-dimensional waveguide-like flows as outcomes of optical field singularities. These flows, though peculiar for each of the designed structures, turn out to be key factors of the perfect resonant optical absorption. The electromagnetic fields show tight two-dimensional confinement: a sharp vertical confinement of the resonant-cavity type combined with a lateral metasurface supported confinement. The siliconabsorbing layer and its oxide environment are confined between two subwavelength metasurfaces such that the entire design is well compatible with silicon-on-insulator microelectronics. The design concept and its outcomes meet the extensive challenges of ultrathin absorbers for minimum noise and an ultra-narrowband absorptance spectrum, while maintaining an overall very thin structure for planar integration. With these materials and such objectives, the proposed designs seem essential, as standard approaches fail, mainly due to a very low silicon absorption coefficient over the near-infrared range. Tolerance tests for fabrication errors show fair tolerability while maintaining a high absorptance peak, along with a controllable deviation off the central-design wavelength. Various applications are suggested and analyzed, which include but are not limited to: efficient photodetectors for focal plane array and on-chip integrated silicon photonics, high-precision spectroscopic chemical and angularposition sensing, and wavelength-division multiplexing. © 2020 Chinese Laser Press

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1. INTRODUCTION

Enhancement of optical absorption is sorely sought in various light-management devices, such as solar photovoltaic and heatstorage cells and photodetectors (PDs), and it may also contribute to cutting-edge optical camouflage implementations. Silicon (Si) has such physical properties, including optical and electronic ones, that made it dominant in microelectronics and also a favorite for many photonic and optoelectronic applications, in particular those mentioned above. Broadly, Si photonics terms the field of prediction, design, fabrication, and implementation of photonic devices made of Si, its

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compounds with O and N, and with other IV-group elements. Often, this term is used in a narrower sense to refer to optical solutions to problems of connectivity and communication within a Si chip package by integrating Si-based optics and optoelectronics within the chip. Research on Si photonics began in the 1990s [1–5], proceeded through the 2000s [6–9], and has continued ever since [10–14]. On-chip optical interconnects require high-power lasers to convert the electrical data signals into light pulses, optical waveguides/fibers to propagate the light, and fast PDs to turn the optical signals back into electrical signals. Many breakthroughs have already been achieved

[10–14] in the realm of diverse Si-based photonic and optoelectronic components.

From now on, the term PD is used for brevity to present any photo-absorbing device. Assume for simplicity that each photon absorbed by the photoactive layer of the PD generates an electron-hole pair. Then $\eta = A$, where η and A are the quantum efficiency and optical absorptance, respectively, of the PD. In a monolithic PD, the detected light traverses the absorber once, so η may be increased only by increasing its thickness t_a , but the thicker the absorber, the stronger the thermal noise and the longer the transit time (in a diode PD). For some applications, e.g., PDs' focal plane arrays (FPAs) [15], high η and low noise are of paramount importance in order to show high responsivity while avoiding faulty high-density packaging. For others, e.g., the on-chip integrated Si-PDs (which should also be thin enough to be compatible with current planar-chip technology), the short transit time is also crucial.

A drastic decrease of t_a to decrease the noise and transit time while maintaining high η requires the confinement of incoming light around the absorber layer. An approach adopted to this end is based on the solid-state resonant Fabry–Perot cavity, in which this layer is sandwiched between two multilayered sections. In each of these, the distributed Bragg reflector (DBR) acts as a mirror and two layers adjacent to the absorber layer form an optical cavity. There, even very thin active layer may absorb most of the incident light power. Such a resonant-cavity enhanced PD (RCE-PD) [16] enables much more efficient (and faster, when relevant) detection, as well as operation at higher temperatures with a much thinner absorber than in its monolithic counterpart, although these benefits are traded off against a smaller spectral bandwidth.

The field of RCE-PDs has matured over the past decades regarding different spectral ranges, absorber materials, and physical devices; see, e.g., Refs. [3–6,17,18] for Si-PDs. However, the restriction on the DBRs' reflectances [16] for attaining maximal peak η may require deposition of too many layers, such that even with quite a thin absorber the entire device may prove too thick for on-chip integration. Such a high number of layers, as a rule, exceeds a technology-defined critical value, beyond which growth defects inevitably emerge and impair the device's performance even with lossless DBRs; see, e.g., Ref. [19] for the SiO₂/Si₃N₄ ones. The reported on-chip Si-PDs, fabricated with thin but absorbing SiO₂/Si DBRs and rather large t_a [3–6], showed very low NIR efficiency $\eta_{max} < 50\%$. Thus, reducing the Si RCE-PD thickness below some threshold clearly conflicts with the perfect- η objective.

Due to the reasons outlined above, to concurrently meet the challenges of a high η , thin Si layer, and small overall thickness of a NIR Si-PD, a new design solution, quite different from the RCE approach, is needed. We suggest using metasurfaces [20], which enable us to redesign the known photonic devices to make them thinner and planar while concurrently being of a highly reduced complexity and a high performance. For our purposes, we use the simplest kind of metasurface, a subwavelength (SW) grating-on-layer (GL) metasurface. So far, an upgrade of the RCE-PD with the replacement of the front DBR by a one-dimensional (1D) GL (not necessarily an SW one) was proposed, and proof-of-concept examples of the thus modified

HgCdTe-PDs were reported [21,22]. The round-trip phase control [16] extended to account for the presence of the grating and an optimization allowed for [21,22] fabrication-tolerant designs with peak $\eta \approx 100\%$ and a back DBR twice as thin as that of the best performing RCE-PD composed of the same materials. However, due to the remaining DBR, such a design solution proves yet insufficient for downscaling the Si-PD at goal.

In Si photonics, a 0, ± 1 diffraction order Si–SiO₂ grating was theorized [1] to bend normally incident light into the depletion region of the on-chip Si-PD. Another efficient light coupling into a CMOS-compatible Si-PD was demonstrated; see, e.g., Ref. [7], with a Si–air grating under oblique incidence, at which only one nonspecular diffraction order is supported. A similar strategy is followed [13,14,23] in contemporary Si photonics. In several important applications, the high-refractive-index (RI) contrast SW-GL, despite its structural simplicity, may function like a much more complex metasurface [20]. Due to this fact, such SW-GL nowadays attracts much interest [24] as a new platform for integrated optoelectronics, in particular, Si photonics [13,14].

In this study, novel ultrathin Si-photonics structures are proposed and designed to show strikingly efficient light absorption within a narrow portion of the NIR range, around the centerdesign wavelength (CDW) $\lambda_0 = 0.8 \ \mu\text{m}$. In theoretical studies [1], this wavelength was allocated as one from operation bandwidths of on-chip NIR Si photodiode PDs [3–5,7,9], implying that a short-haul light guiding with Si [10–13] is still feasible at $\lambda = \lambda_0$. Remarkably, these structures exhibit strong vortex and waveguide flows of the internal electromagnetic (EM) power as outcomes of the EM fields' singularities [25].

The rest of the paper is organized into six sections. Section 2 states the problem, outlines the suggested structures, and discusses their modeling. In Section 3, an optimization-based design procedure is described, and the considered structural parameters and computed absorptance spectra are presented. In Section 4, the simulations of the EM-fields-related physical quantities at the CDW are reported, and their results are discussed. Section 5 reports the designs' tolerance testing results. Section 6 considers possible applications of the designed structures, and Section 7 concludes the study.

2. STATEMENT OF PROBLEM, PROPOSED ENHANCED ABSORPTION STRUCTURES, AND PRINCIPLES OF MODELING

In the quest to thin the Si-photonics structure as much as possible, it will perfectly absorb light at the designated NIR CDW while surpassing the bounds on the mirrors' and cavity's thicknesses noted in the introduction. We redesign the RCE structure by replacing both DBR mirrors by 1D GL with the same grating period Λ . Since SiO₂ and Si₃N₄ are well technologically compatible with each other and Si for composing the DBR and the cavity [19], respectively, we employ these materials in the redesign, while benefiting from the previous design technique [21,22] to attain nearly perfect peak absorptance.

Figure 1 shows a schematic of the dual-GL structures under study. There, the cavity and substrate material is SiO₂, and the



Fig. 1. Sketch of the enhanced light-absorption structure that comprises a cavity-embedding Si layer, two GLs enclosing the cavity, and a substrate. The coordinate system is shown, where the grating periodicity and grooves/lines are along the x and y axes, respectively, and the layer stacking and light impinging directions are along the z axis.

front air–Si₃N₄/Si₃N₄ with the back SiO₂–Si₃N₄/Si₃N₄ GL encloses the cavity. As shown in Fig. 1, we consider the through-substrate backside illumination (BSI) at normal incidence. Benefiting from the absence of the backside electrical contacts, BSI has the essential advantage of reducing the PD pixel size without decreasing the amount of input light power [26], which is desirable for on-chip applications.

In this paper, we adopt the $\eta = A$ modeling, which simplifies the design to one in which only the optics need to be considered. Herewith, rather than being restricted to a specific PD technology, we analyze the structures optically, characterizing the constitutive layers by complex RIs n + ik, where n are the real RIs and $k \ll n$ are the extinction coefficients. This modeling still has widespread use for initial designs of the electronic PDs [16,18,21,22,26,27]. In such cases, optical modeling can be complemented and refined by an electron-device simulation [18,28,29]; however, it is not expected to disprove the optical modeling outcomes. The n and k spectra for the involved materials are found in Refs. [30–32]; their values at the CDW are given in Table 1.

Here, the design emphasis is put on obtaining the maximal CDW efficiency $\eta_{\text{max}} = \max \eta(\lambda_0)$, while maintaining t_a constant per design, and keeping the back-GL cladding's thickness (similarly to the DBR layers' thicknesses) fixed at the quarterwave value $t_{\text{qw}} \approx 100.2$ nm. Such dimensions as Λ and the

Table 1.RIs at the CDW of the Materials that Are Set inthe Text

$\lambda_0 \ [\mu m]$	Si_3N_4	SiO ₂	Si
0.8	1.9962 + i 0	1.4533 + i 0	3.6925 + i 0.0065

groove width W (constrained for simplicity to be the same for both GLs), etch depths $t_{\rm fg}$ and $t_{\rm bg}$, as well as the thicknesses of the front-grating cladding $t_{\rm Si_3N_4}$ and front and back cavities, $d_{\rm f}$ and $d_{\rm b}$, respectively, are the design variables. We observe the SW regime, limiting *ab initio* the operation wavelengths from below by the Rayleigh wavelength $\lambda_{\rm R} = \Lambda n_{\rm SiO_2}$, which suppresses the non-specular orders of the grating diffraction into the cavity and so prevents losses due to light escaping from the structure's sides. Note that this is not compulsory since the non-specular orders can be suppressed with extra constraints; see, e.g., Refs. [21,33].

3. DESIGN PROCEDURE, OPTIMAL STRUCTURES, AND THEIR EFFICIENCY SPECTRA

We perform the computer-aided designs with an in-house software run in MATLAB environment. The software consists of two modules: "Simulation," which contains the codes of an inhouse recast rigorous coupled-wave analysis (RCWA) for simulations of the structures merging smooth and grating-patterned layers as described in detail elsewhere (see Ref. [21] and references therein); and "Optimization," which calls a multistart optimization algorithm from MATLAB Optimization Toolbox [34–36] and a graphical user interface (GUI), which forces the modules to interact and thus drives the simulations and designs as shown in Fig. 2.

In order that the optimization would not be blind, we are guided by an approach to obtaining initial (start, trial) design parameters that, in modern terms, are geometric-phase control [20]. Homogenizing the gratings as described in Appendix A, we first design the front and back GLs to be standalone and to be highly and lowly reflective (subsections A and B), respectively, at a normal CDW incidence from a semi-infinite SiO₂. In this way, we assess trial values Λ^* , W^* , t_{fg}^* , t_{bg}^* , $t_{Si_3N_4}^*$ for the GLs' parameters shown in Fig. 1. With these, our RCWA code returns the GLs' reflection-amplitudes' phases φ_f and φ_b , the sum of which Φ gives a rough idea on the geometric phase of an unloaded dual-GL cavity. Φ leads



Fig. 2. Sketchy flowchart of the optimization process. The GUI Module inputs the trial parameters, assessed as described in the text, fixed parameters, and constraints to the Optimization Module. There, the trial-and-error multi-start algorithm generates the next start points and inputs them into the Simulation Module, which feeds the algorithm back and loops until attaining an optimum.

Table 2. Parameters of the Structures Designed Shown Below with Two Si Layer Thicknesses Outlined in the Text^a

Structure	t _{bg}	$d_{\rm b}$	d_{f}	$t_{\rm Si_3N_4}$	t_{fg}	Λ	W	$\eta_{ m max}$	Q	$\lambda_{\mathbf{R}}$	N	t
						t_{a1}	= 0.150					
2DBR ₁		0.069	0.030					0.808 (0.995)	303	_	33 (63)	3.779 (7.346)
$2GL - TE_1$	0.120	0.130	1.710	0.792	0.305	0.439	0.252	0.999	197.53	0.642	7	3.307
$2GL - TM_1$	0.173	0.032	1.764	1.380	0.215	0.468	0.185	0.999	1012.6	0.684	7	3.814
-						t_{a2}	= 0.025					
2DBR ₂		0.094	0.053				_	0.353 (0.985)	230	_	29 (71)	3.226 (8.220)
2GL - TE ₂	0.109	0.203	0.968	0.271	0.322	0.443	0.251	0.999	249.22	0.644	7	1.998
$2\text{GL} - \text{TM}_2$	0.586	0.003	1.262	0.047	0.437	0.479	0.175	0.995	5925.9	0.696	7	2.460

 ${}^{a}\eta_{max}$ is the peak efficiency; Q and λ_{R} are the guality factor and the Rayleigh cutoff, respectively, obtained as described in the text. Also, N is the total number of layers (including gratings, if any), t is the thickness of the structure excluding the substrate, and other thicknesses are described in the text and shown in Fig. 1. The parenthesized numbers are related to the 2DBR_{tu} structures described in the text. All dimensional parameters are in micrometers.

us to estimate t_c^* for such cavity's thickness with the roundtrip phase condition $4\pi n_{SiO_2}(\lambda_0)t_c^*/\lambda_0 + \Phi = 2\pi m$; see, e.g., Refs. [16,27]. Though the Si layer is not involved in the start design, the cavity in the final design does adopt it (see Section 2), so given a t_a , we hold the bound $t_c^* > t_a$ by tuning m. Finally, the front and back cavity thicknesses, but not t_c (see Fig. 1), are optimized with initial values d_f^* , d_b^* , chosen compatible with the relation $d_f^* + d_b^* = t_c^*$, where we take $d_b^* = d_f^*$ for simplicity. The start equality proves *a posteriori* far from the relation between d_b and d_f for the optimized dual-GL structures but much closer to that for the optimized dual-DBR structures.

Due to the very small extinction of Si in the operation range [see $k_{\rm Si}(\lambda_0)$ in Table 1], it might be expected that the Si layer would not be too thin; see, e.g., Refs. [3–5]. In fact, due to Ref. [27], we figured out two initial values of t_a for optimizing the dual-DBR structure $\lambda_0/(2n_{\rm Si}) \approx 0.108 \,\mu{\rm m}$ and $\lambda_0/(4n_{\rm Si}) \approx 0.054 \,\mu{\rm m}$. To ultimately maximize $\eta_{\rm max}$, these values were finally refined by the optimization process to $t_{\rm a1} = 0.150 \,\mu{\rm m}$ and $t_{\rm a2} = 0.025 \,\mu{\rm m}$. These were used among the fixed input parameters in optimizing the dual-GL structures (see Fig. 2) to attain $\eta_{\rm max} \sim 100\%$. The results of the computer-aided optimal design, with two Si layer thicknesses indicated

above are shown in Table 2; the $\eta(\lambda)$ spectra are shown in Figs. 3 and 4.

While the planar-interface layers are insensitive to the polarization of normally incident light [37], the 1D gratings (see Appendix A) are strongly sensitive to it. Hence, given a $t_{\rm a}$, the design yields at least two different dual-GL structures, i.e., those for the TE- or TM-polarized irradiation, when the incident electric or magnetic field is overall along the grating grooves, respectively. To have an idea up to what extent they improve the efficiency over the RCE structure, we optimized the latter with the SiO₂/Si₃N₄ DBRs at both sides, using the same technique, while keeping its overall thickness comparable with those of the dual-GL structures. Releasing this constraint, the optimal dual RCE structures reach ~100% CDW absorptance at the cost of a huge number of layers (parenthesized in Table 2), which makes them technologically unfeasible. Further, we refer to the thus optimized structures as 2GL - TE, 2GL - TM, 2DBR, and 2DBR_{tu}.

Table 2 and Figs. 3 and 4 show that the thinner the absorber, the higher the relative gain in η_{max} of the designed dual-GL structures compared to 2DBR₁ and 2DBR₂. It can be inferred from Figs. 3 and 4 that, as regards the full width at half-maximum (FWHM) $\Delta\lambda$, the $\eta(\lambda)$ spectra of the



Fig. 3. Efficiency spectra of the designed structures with a $0.15 \,\mu$ m thick Si layer. Broken lines: red, $2GL - TE_1$; green, $2GL - TM_1$; full blue line, $2DBR_1$. The structural parameters are found in Table 2.



Fig. 4. Efficiency spectra of the designed structures with a 0.025 μ m thick Si layer. Broken lines: red, 2GL – TE₂; green, 2GL – TM₂; full blue line, 2DBR₂. The structural parameters are found in Table 2.

2GL - TE and 2GL - TM are the widest and narrowest, respectively, whereas for the RCE structure, $\Delta \lambda$ is in between them, with the same absorber thickness t_a . It seems that the $\eta(\lambda)$ spectra of the optimized dual-GL structures get sharper and narrower as t_a decreases; the extremely small $\Delta \lambda$ spectrum of the $2GL - TM_2$ in Fig. 4 is notable. Due to the definition of the damped-resonator quality (Q) factor $Q = \lambda_0 / \Delta \lambda$, the Q factors for the designed structures are easily assessed from the data of Figs. 3 and 4. The estimated values are displayed in Table 2 and prove rather high, especially for the TM structures. Note that on-chip Si RCE-PDs were fabricated [5,6] with t_a that are quite larger than those of the 2DBR_{1,2}; see Table 2. An RCE-PD with the MgF2/ZnSe front and SiO2/poly-Si back DBRs [5] showed $\eta_{\text{max}} \sim 34\%$ at λ_0 , and for one with only SiO_2/Si back DBR [6] it was ${\sim}44\%$ at 0.86 $\mu m,$ which is notably smaller than the η_{max} of 2DBR₁; see Table 2.

4. ELECTROMAGNETIC FIELDS AND POWER FLOW

It is worth noting that all the studied structures, laterally unlimited (actually, huge times λ_0 wide) and several times λ_0 thick, are three-dimensional (3D) structures. In this regard, our simulations are 3D, though they are simplified by the independence of the EM fields on *y* for the dual-GL structures (as with, e.g., all RCWA-based simulations), or on *x*, *y* for the dual-DBR ones, which does not imply any two-dimensional (2D) or 1D spatial confinement. The light confinement in the cavity–absorber region, mentioned in the introduction for the RCE structures, and that for the dual-GL ones—both discussed and quantified in this section—is entirely related to the EM fields.

In the RCE structures, the EM confinement is built up due to the optical resonance in a cavity enclosed by plane-parallel mirrors [16,27] and so are 1D in essence. Since the dual-GL structures under study are assemblies of identical vertical (z)top-to-substrate microscale cuts of width Λ from the structure shown in Fig. 1, another physics, which is an EM field resonant recirculation in micro/nanoscale areas or volumes on the subwavelength scales [38], might be more suitable for them. We view the above cut as a "microcavity"; the quotes indicate that it has no real vertical boundaries. Inside the microcavity, the lamella-groove boundary line marks an actual materials interface in the grating regions; continued outside them is a dummy, reminiscent of the above real boundary. The nano/microcavity reasoning has been adopted previously for a simpler gratingbased optical resonator [39] and 2D photonic crystal slabs [40]. While the physics in the RCE devices, behind the EM confinement, is well understood, e.g., Refs. [16,27], its current understanding for the micro/nanocavity devices is much less mature and is more particular per structure. In these cases, visualizing the electric (E) and magnetic (H) fields and Poynting vector (S), computed at the CDW, may shed light on the near-field EM recirculation and thus portray the confinement and power flow ad hoc. Doing that for the dual-DBR structures aims primarily at confirming the adopted vision [16,27]. For any layer, a rigorous relation between A and S states that A is the difference between surface-averaged normal components of **S** (power fluxes) at the layer's exit and entry.

For the 2DBR_{1,2}, in which the only nonzero components E_y , H_x , and S_z depend only on z, we plotted $|E_y|^2$, $|H_x|^2$, and $|S_z|$ versus z in Fig. 5. As is seen, while $|S_z|$ is slightly sub-linear in one half and super-linear in the other half of the absorber of 2DBR₁, it is visibly linear throughout the absorber of 2DBR₂. Overall, sup $|E_y|^2 < \sup |H_x|^2$, which is due to high RI contrast between SiO₂ and Si; see Table 1. In the absorber layers, the maxima of $|E_y|^2$ and minima of $|H_x|^2$, and vice versa, are complementary.

In the dual-GL TE or TM structures, $E_y, H_x, H_z \neq 0$ or $H_y, E_x, E_z \neq 0$, respectively, and $\mathbf{S} \neq \mathbf{0}$. Hence, we plotted the surface graphs of the appropriate squared modules of these fields' components versus (x, z) in Figs. 6 and 7. These show the continuity of **H** [Figs. 6(c), 6(d), 6(e), 6(f), 7(c)], E_y [Figs. 6(a) and 6(b)], and E_x [Fig. 7(a)], and the discontinuity of E_z [Fig. 7(b)] when crossing the cavity–absorber interfaces, which is in fair agreement with the EM boundary conditions. Also, the quiver maps of **S** are plotted in Figs. 8 and 9. Due to the *a priori* x periodicity of $|E_i|, |H_i|$, and $|S_i|$, restricting the x to within any segment of length Λ , not necessarily $[0, \Lambda]$, is sufficient to pick out the microcavity. For reasons of scale and convenience, z is also restricted to a lateral slice of the



Fig. 5. Electric and magnetic fields' amplitudes squared (left ordinate) and Poynting vector modulus (right ordinate), normalized to the input $|S_z|$, across the cavity–absorber range of the dual-DBR structures: (a) 2DBR₁; (b) 2DBR₂. The intersections of the dashed lines with the abscissa mark the Si layer ends. Simulated at the CDW.







Fig. 6. Normalized to the input values, squared amplitudes of the EM fields versus $0 \le x/\Lambda \le 1$ and z as defined in the text. (a), (c), (e) For 2GL – TE₁: 0.22 µm $\le z \le 0.671$ µm; $x/\Lambda = 0.426$ is the internal grating lamella-groove interface derived line; (0, 1) × (0.35 µm, 0.5 µm) is the absorber cut within the microcavity. (b), (d), (f) For 2GL – TE₂: 0.209 µm $\le z \le 0.679$ µm; $x/\Lambda = 0.433$ is the internal grating lamella-groove interface derived line; (0, 1) × (0.312 µm, 0.437 µm) is the absorber cut within the microcavity. Simulated at the CDW.

microcavity. Due to a perfect-mirror action of the front GL, the EM fields strongly decay there, and hence it is overall excluded from the scope. In Figs. 6 and 7, the *z* range covers such parts of the absorber–cavity region, in which the front cavities are top cut to 10% and 25% of their thickness for t_{a1} and t_{a2} , respectively. In Figs. 8 and 9(a), the *z* range is extended to also include

the back GL. The z range in Fig. 9(b) includes a part of the absorber–cavity region, where the front cavity is top cut to retain 5% of its thickness and top 5% of the back grating z range.

It appears that, when simulating **E**, **H**, and **S** for the microcavity frame $[0, \Lambda]$, the $|E_i|^2$, $|H_i|^2$ surfaces and **S** flow maps prove asymmetric as regards their boundaries-adjoined critical



Fig. 7. Normalized squared amplitudes of the (a), (b) electric and (c) magnetic fields' components for the 2GL – TM₁ structure shown versus $0 \le x/\Lambda \le 1$ and 0.273 µm $\le z \le 0.631$ µm as defined in the text. For a simulation convenience, the grating cell was inverted so that $x/\Lambda = 0.396$ is now the grating lamella-groove interface derived line; (0, 1) × (0.305 µm, 0.455 µm) is the absorber cut within the microcavity. The normalization and simulation wavelengths are the same as in Fig. 6.

features (see further in the present section). An analysis shows that the symmetry is restored while preserving the graphs' topology, with applying a rigid lateral shift of the microcavity frame by the distance $\Delta = \Lambda/4 - W/2$; note that $\Delta = 0$

for symmetric grating structures with $W = \Lambda/2$. Table 2 affords the following values of Δ/Λ : -0.03702 (2GL – TE₁), -0.03352 (2GL – TE₂) and 0.06891 (2GL – TM₁), 0.06769 (2GL – TM₂), so the shift is to the right for 2GL – TE_{1,2}, and the shift is to the left for 2GL – TM_{1,2}. Thereby, the EM fields and Poynting vector were recomputed for the shifted frame, and Figs. 6–9 show the resulting $|E_i|^2$, $|H_i|^2$ surface graphs and **S** quivers. In all the following figures and discussions, *x* still denotes the lateral coordinate, and bare values of the microcavity lateral boundaries, i.e., 0, Λ , are assigned to the respective shifted boundaries.

A. Dual Grating-On-Layer TE Structures

1. Electric and Magnetic Field Amplitudes

For 2GL – TE₁ and 2GL – TE₂, the plots of $|E_y|^2$, $|H_x|^2$, and $|H_z|^2$ shown in Figs. 6(a), 6(c), 6(e), and Figs. 6(b), 6(d), 6(f), respectively, display highly peaked surfaces. It is found that the minimum *x*-direction distance between each two peaks is 0.5, in units of Λ , as may be expected from the *x* periodicity. The bell-shaped $|E_y|^2$ and $|H_x|^2$ surfaces are strictly confined inside the microcavity. They have four and two peaks located at the cavity–absorber interface lines and inside the absorber, respectively, for 2GL – TE₁, and vice versa for the 2GL – TE₂.

As seen in Figs. 6(e) and 6(f), the $|H_z|^2$ surfaces show three pairs of peaks for $2GL - TE_1$ and three peaks for $2GL - TE_2$, respectively. One of the peaks (pairwise or singular) is internal, whereas the two others adjoin the vertical boundaries of the microcavity. The surface(s) adjoined to one boundary is (are), as expected, strictly symmetric to the one(s) adjoined to the other boundary, attaining there their maximums. In 2GL – TE₁, the peaks of $|E_{y}|^{2}$ and $|H_{z}|^{2}$ lie on the cavity– absorber interface lines [see Figs. 6(a) and 6(e)]. The peaks of $|H_x|^2$ lie on the midline of the absorber ($z = 0.425 \ \mu m$); see Fig. 6(c). In 2GL – TE₂, the situation is just the opposite; here the absorber is much thinner than in the $2GL - TE_1$, so that in order to provide $\sim 100\%$ resonant absorptance, $|\mathbf{E}|$ tends to be as large as possible, which pushes H out of the absorber. The maximums of $|E_{\gamma}|^2$, $|H_x|^2$ at the vertical microcavity boundaries, over the corresponding z ranges in Fig. 6, are about 1.24, 1.29 and 1.58, 1.50 for $2GL - TE_1$ and 2GL – TE₂, respectively. In 2GL – TE₁, $|E_{\nu}|^2$ has the absolute maxima at the crossing points of two vertical midlines between the lamella-groove interface-derived line and microcavity boundaries with the horizontal absorber-cavity interface lines, while $|H_x|^2$ has the absolute maxima at the crossing points of the same vertical lines with the absorber's midline. In $2GL - TE_2$, the maxima points of these functions are swapped.

In view of the location and sharpness of the $|E_y|^2$ and $|H_x|^2$ peaks, the tight vertical-by-lateral 2D EM fields' confinement occurs in each microcavity of both structures. Moreover, because the above boundary values of $|E_y|^2$, $|H_x|^2$, relative to their global maximums, are about 1.5%, 0.3% for 2GL – TE₁, and 0.5%, 0.9% for 2GL – TE₂, the microcavities in both structures come close to electric and lateral-magnetic isolation. In Figs. 6(e) and 6(f), we observe that the internally peaked $|H_z|^2$ surface's parts are confined similarly to the $|E_y|^2$ and $|H_x|^2$ surfaces. Because of the boundary-peaked parts of the $|H_z|^2$ surface, H_z penetrates from a given microcavity



Fig. 8. Maps of the EM power flow at the CDW in the structures (a) $2GL - TE_1$ and (b) $2GL - TE_2$ (Table 2). The quiver ranges are defined in Section 4, and the horizontal dashed lines show the material interfaces. The structure cut is shown in the inset to (a).

to any one adjacent to it, so there is no vertical magnetic isolation of the microcavities.

2. The EM Power Flow

In Figs. 8(a) and 8(b), we visualize the EM power flow in $2\text{GL} - \text{TE}_1$ and $2\text{GL} - \text{TE}_2$, respectively, using the quiver plots of **S**. The figures display the Poynting vector distributions of those structures, demonstrating vortex-like flows. Both flows consist of two vortices, a central one with counterclockwise **S** circulation and the other, which is split into two symmetric parts adjacent to the microcavity boundaries with clockwise **S** circulation, so that the total circulation appears to be zero. The distance between the vorticity centers is 0.5A, as expected from the A periodicity of $|S_i|$. When moving along the *x* axis through the microcavity, S_x consistently switches its sign at the vertical lines of $x = 0.25A = x_1$ and $x = 0.75A = x_2$. The continuity of S_x dictates that these lines are also the lines

of the $|S_x|$ minima. The inner vortices' centers lie on the midline of the microcavity $x = \Lambda/2$, and the boundary-adjoined vortices are centered at the vertical boundaries of the microcavity. For both structures, the vortices are strictly confined between the lateral boundaries of the chosen microcavity slices. Contrary to the inner vortices, the boundary-adjoined ones are unconfined laterally and percolate continuously from the given microcavity to the neighboring ones.

Both vortices in $2\text{GL} - \text{TE}_1$ [Fig. 8(a)] consist of two linked sub-vortices sustaining the above noted respective **S** circulations, which correlates clearly with the pairwise peaks of $|E_y|^2$ in Fig. 6(a) and of $|H_z|^2$ in Fig. 6(e). Figuratively speaking, the vortices start the circulation at the back cavity–absorber interface, stretch vertically to the absorber layer, and make the counter-circulation turn at the absorber–front cavity interface. Between the vortices and apart from their vicinity, the EM power flow inside the absorber layer is dominantly vertical.



Fig. 9. Maps of the EM power flow at the CDW in the structures (a) $2GL - TM_1$ and (b) $2GL - TM_2$ (Table 2). The quiver ranges are defined in Section 4, and the horizontal dashed lines show the material interfaces.

It crosses the absorber back and forth as a result of the reflection from the GLs. The combination of the vertical flow alongside the lateral one at the vortices' closure seems to be the key mechanism of the ~100% peak absorptance of this structure. Four additional singularities appear at the back grating region, two of which are vortices, and the other two at $x = x_1$ and $x = x_2$ are seemingly flow junctions. The left junction seams to channel a lateral power that flows into it from opposite directions (left and right) to a vertical one that flows away from it, also in opposite directions (up and down), whereas the right junction does exactly the opposite.

The vortices in $2GL - TE_2$ [Fig. 8(b)] do not cluster into sub-vortices, and the S flow they carry circulates entirely inside the back cavity. Between the vortices and apart from their close vicinity, the S flow is dominantly vertical and is pumped into and out from the absorber layer via the vertical junctions, the midlines of which are again $x = x_1$ and $x = x_2$, respectively. Such an EM power transfer produces lateral waveguide-like right and left S flows inside the absorber layer slice within the microcavity. As seen in Fig. 8(b), the right flow occurs between the junctions' entrances and is restricted within the microcavity, whereas the left flow is piecewise inside $[x_2, \Lambda]$ and $[0, x_1]$ and is responsible for the EM power transfer between the periodically arranged microcavities. This lateral back and forth S flow is the main absorption enhancement mechanism for $2GL - TE_2$. The reason why the lateral **S** flow occurs in the absorber layer of this structure and does not in that of $2GL - TE_1$ is the very small thickness of the former layer, 6 times smaller than that in $2GL - TE_1$; it appears that with such a thin Si layer, a planar SiO₂/Si/SiO₂ trilayer will guide only a single light mode. Although the light-absorption enhancement mechanism in the $2GL - TE_2$ structure differs notably from that in the previously discussed $2GL - TE_1$ one, it also provides ~100% peak absorptance. Note that the **S** flow at the back grating region is very similar to that of $2GL - TE_1$, which seems to be due to the closeness of their dimensions. Originating at the back grating, this mixed lateral-vertical flow supports the EM energy confinement and channels the EM power in the microcavity of both TE structures.

A spectral deviation from the optimal CDW or an alternation of the structure's optimal dimensions, such as the grating parameters and the absorber thickness, which result in a change of the optical properties, affects the overall and absorber flows (direction and/or magnitude) and the singular points (location and/or existence). In the 2GL – TE₂ structure, for example, the decrease in absorptance due to these variations is noted as breaking the absorber's balanced lateral optimal flow and the vortices' existence and location. Note that the vortices do not appear only in the optimized structures and not discretely at the CDW. Still, moving further away from peak resonances and confinement wavelengths, by changing the incident wavelength or optimal dimensions, conventional flows of no vorticity are likely to be observed.

B. Dual Grating-On-Layer TM Structures

1. Electric and Magnetic Field Amplitudes

The plots of $|E_x|^2$, $|E_z|^2$, and $|H_y|^2$ for 2GL – TM₁ shown in Figs. 7(a)-7(c), respectively, also display the surfaces that are highly peaked but topologically very different from those for the dual-GL TE structures considered above. It is seen that the $|H_{\nu}|^2$ surface (disregarding the back cavity layer), similarly to the $|E_{y}|^{2}$ one in Fig. 6(a), consists of four highly peaked regions. However, here two of the $|H_{y}|^{2}$ surface peaks are relatively weak and sharp, with maximums located at the absorber-front cavity interface, and two others are strong and less sharply peaked inside the absorber layer. In contrast with the $|E_{\gamma}|^2$ plot, the peaks of the $|H_{\gamma}|^2$ plot in Fig. 7(c) are asymmetric relative to the absorber layer. Nevertheless, with an appropriate interchange $E \leftrightarrow H$, several properties of the present plots in Fig. 7 seem to comply with those in Figs. 6(a), 6(c), and 6(e). Namely, the shortest x-direction distance between each two maximum pairs of both $|H_y|^2$ and $|E_x|^2$, and minima of $|E_z|^2$, is 0.5 Λ . Also, a strong resemblance between the $|E_x|^2$ and $|H_x|^2$ plots can be seen when comparing Fig. 6(c) to Fig. 7(a). Similarly to the 2GL – TE cases, the two $|H_{y}|^{2}$, $|E_{x}|^{2}$ maximums lines are at $x = x_{1}$ and $x = x_{2}$, and the same holds for the $|E_z|^2$ minima lines.

The above asymmetry along the z axis may be attributed to the very small thickness (32 nm) of the back cavity layer (see Table 2), in which the following dissimilarity between the TE and TM structures is observed. While all the above discussed EM field amplitudes in the dual-GL cavity TE structures (Fig. 6) increase when progressing up along the z axis, i.e., moving right to left in the figure inside the back cavity layer towards the absorber interface, the EM fields for $2GL - TM_1$ decrease along the similar path. An additional dissimilarity is found in the above-noted discontinuity of $|E_z|$ as seen in Fig. 7(b). Inside the absorber $|E_z|^2$ drops ~45 times at most, but it still peaks there up to ~20 relative units, which concurrently with high $|E_x|^2$ provides ~100% peak absorptance. Due to a close similarity to the 2GL – TM₁ fields and to avoid repetitions, the 2GL – TM₂ fields are not shown. However, the EM power flows and the absorption mechanisms of both structures, which appear to be quite different, are discussed in the next subsection.

2. The EM Power Flow

The EM power flow quiver of the $2GL - TM_1$ is shown in Fig. 9(a); for convenience, the flow in the absorber and front cavity regions is scaled and displayed in the inset. This time the dominant vortices are vertically centered at the back GL region, one at the x-axis center and the other split between the boundaries, similar to the cavity-absorber region vortices in the TE case (Fig. 8). Inside the chosen microcavity, no strictly confined vortices are observed. In the present structure, the back cavity layer is the thinnest one, and it is not surprising that, similar to the case of the absorber layer in $2GL - TE_2$, a mostly lateral **S** flow occurs within it. A similar flow proceeds at the lower [upper in Fig. 9(a)] part of the absorber layer, though it is about 6 times less intense due to the discontinuity of S_x in z, while crossing the back cavity-absorber interface $z = 0.305 \,\mu\text{m}$, quantitatively resulting from the discontinuity $|E_z|$; see Fig. 7(b). For the same reason, a similar "jump" of S_x occurs when crossing the absorber-front cavity interface z = $0.455 \,\mu\text{m}$, which is seen in Fig. 9(a). This behavior contrasts with the overall continuity of S_x in 2GL – TE₁ and $2GL - TE_2$.

In this case, the strong lateral **S** flow in the back cavity layer originates from the EM power pumping out and into the vortices that persist in the back GL region (not shown here). As seen in Fig. 9(a), the inter-layer transfer of the EM power goes down and up via the above junctions along the $x = x_1$ and $x = x_2$ midlines, respectively. Although the **S** flow over the upper [lower in Fig. 9(a)] part of the absorber is negligibly small as compared to that in the rest, the predominantly lateral **S** flow over the latter proves sufficiently strong to provide the perfect optical absorption.

Focusing mainly at the absorber and its closer regions, the EM power flow of the $2GL - TM_2$ structure is shown in Fig. 9(b). As for the previous structures, a laterally centered and a boundary split vortices are displayed, only this time they are both vertically centered inside the absorber layer. Both vortices are stretched vertically throughout the entire absorber layer and close at the cavity layers. The vortices' closure echoes vertically deep and laterally wide into the back GL and front cavity regions; actually, it appears that the stretched singularity inside the absorber layer. Between the vortices and apart from their closest vicinity, the EM power flows almost completely vertically up and down inside the absorber. So although it is ultrathin, the high electrical fields and the back and forth flow inside the absorber layer fuel the ~100% peak absorptance.

To conclude the section, maximizing the absorptance peak at the CDW dictates the vortices centers' locations and the waveguide-like flows within the structures in general, and inside the active layer in particular, which allows attaining the $A \sim 100\%$. Let us stress a generic difference between the EM confinement and power flow in the dual-DBR (RCE) and dual-GL structures. Namely, in the former, due to the lateral interfaces, light paths are 1D vertical round-trips along the *z* axis; see Fig. 4. The lateral and vertical interfaces in the latter concurrently force light to flow along complex 2D paths in the *x*-*z* plane; see Figs. 8 and 9. For the dual-DBR structures, the zero EM field singularities have little effect on S_z , which is a rather plain function of *z*, as seen in Fig. 5. However, the EM field zeros in the dual-GL structures, seen in Figs. 6 and 7, do complicate the **S** flow, which manifests itself in diverse topological structures, e.g. vortices, shown in Figs. 6 and 7.

5. DESIGN TOLERANCES

The issue of fabrication tolerances is of primary practical importance in the design of the proposed structures. Since in the manufacturing of these devices only few-layer deposition steps are required, the grating fabrication accuracy is critical. For the dual-GL cavity-enhanced photo-absorbers under study, we carried out the performance tolerability tests, varying $t_{\rm fg}$, $t_{\rm bg}$, and W around their optimal values in Table 2. We adopted $\pm 10\%$ relative error in all the etch parameters for the 2GL – TE_{1,2} and 2GL – TM₁ structures. However, with these, and even twice as small etching errors, the 2GLS – TM₂ structure looses its absorptive performance. Due to that, we inquired about the performance of this structure against $\pm 2\%$ relative etching error.

Tables 3 and 4 display the test results for the designed TEand TM-polarization targeted structures, respectively. Table 3 shows that the structures $2\text{GL} - \text{TE}_{1,2}$ sustain high performance, i.e., exhibit a small peak-efficiency deterioration $\Delta\eta_{\text{max}}$ concurrently with a tiny peak-wavelength shift $\Delta\lambda_0$ on a sub-nanometer scale. Particularly, apart from one exceptional case for each structure, such as with $\Delta\eta_{\text{max}} = -3.4\%$ and $\Delta\eta_{\text{max}} = -1.1\%$ for $2\text{GL} - \text{TE}_1$ and $2\text{GL} - \text{TE}_2$, respectively, all other combinations of the errors result in a less than 1% deterioration.

Similar testing of the structures $2GL - TM_{1,2}$ shows that they still support resonant absorption all over the tests, though

Table 3. Performance Changes of the 2GL – TE Structures with Varying Grating Etch Parameters^a

Etching Errors			2GL	– TE ₁	2GL - TE ₂		
$\frac{\Delta W}{W}$	$\frac{W}{W} = \frac{\Delta t_{\rm bg}}{t_{\rm bg}} = \frac{\Delta t_{\rm fg}}{t_{\rm fg}}$		$\Delta \lambda_0$, nm	$\Delta\eta_{\rm max}, \%$	$\Delta \lambda_0$, nm	$\Delta \eta_{\rm max}$, %	
_	_	_	+0.3	+0.0	+0.36	+0.1	
_	_	+	+0.3	+0.0	+0.26	+0.1	
-	+	_	-0.3	-3.4	-0.23	-0.7	
-	+	+	-0.3	-0.6	-0.11	-1.1	
+	_	_	+0.4	-0.3	+0.32	-0.6	
+	_	+	+0.4	-0.3	+0.32	+0.0	
+	+	_	-0.2	-0.3	-0.20	-0.3	
+	+	+	-0.2	-0.2	-0.20	-0.3	

 ${}^{s}\Delta t_{\rm bg}/t_{\rm bg}, \Delta t_{\rm fg}/t_{\rm fg}$, and $\Delta W/W$ are the relative fabrication errors in the groove depths and widths, respectively, allowed for the etching; $\Delta \lambda_0$ is the shift of $\eta(\lambda)$ peak wavelength, and $\Delta \eta_{\rm max}$ is the resulting absolute change of the peak efficiency. The single + or – sign abbreviates the relative error of +10% or –10%, respectively.

Etching Errors			2GL	- TM ₁	2GL - TM ₂		
$\frac{\Delta W}{W}$	$\frac{W}{V} = \frac{\Delta t_{\rm bg}}{t_{\rm bg}} = \frac{\Delta t_{\rm fg}}{t_{\rm fg}}$		$\Delta\lambda_0$, nm $\Delta\eta_{\rm max}$, %		$\Delta \lambda_0$, nm	$\Delta\eta_{\rm max}$, %	
_	_	-	+1.8	-16.5	-0.8	-35.3	
-	-	+	+1.8	-8.3	-0.7	-53.3	
-	+	-	+9.5	-22.9	+3.1	-61.6	
-	+	+	+9.5	-18.9	+3.6	-7.5	
+	-	-	-9.7	-42.4	-3.6	-23.1	
+	-	+	-9.7	-51.4	-3.4	-36.3	
+	+	-	-3.8	-18.2	+0.4	-86.3	
+	+	+	-3.9	-31.2	+0.7	-59.7	

Table 4. Performance Changes of the 2GL – TM Structures with the Varying Grating Etch Parameters^a

'The single + or – sign abbreviates the relative errors of +10% or -10% and of +2% or -2% for 2GL – TM₁ and 2GL – TM₂, respectively. The meaning of the parameters is explained in the caption to Table 3.

with essentially larger $|\Delta\lambda_0|$, as seen in Table 4. It is also seen that with the same relative errors, 2GL - TM₁ deteriorates in the performance one and a half orders of magnitude stronger than $2GL - TE_1$. For $2GL - TM_1$, except for two severe deterioration cases with $\Delta\eta_{\rm max}=-51.4\%$ and -42.4% (at both $\Delta \lambda_0 = -9.7$ nm), at the other errors' combinations, $|\Delta \eta_{\text{max}}| \leq 31.2\%$, which means a still reasonable tolerance. The high performance sensitivity of the 2GL - TM structures, sharper for the 2GL - TM₂, also correlates well with their superior Q factors (see Table 2) and may be attributed to an alternating charging of the line-groove interfaces by the light electric field component E_x that emerges only in the TM polarization. To shed more light on the above noted deterioration cases for 2GL – TM₁, we inquired how a +10% error in W only, with the other dimensions kept at their optimal values, affects the peak absorptance and **S** flow. We found that $\Delta \eta_{\rm max} \approx$ -37%, so just such an error holds a dominant contribution to those cases, while the errors in $t_{\rm bg,fg}$ when switched on concurrently can either enhance or reduce the deterioration as seen in Table 4. Also, at only $\Delta W/W = +10\%$, two vortices, managing the EM power flow back to the absorber at the optimal dimensions, disappear (not shown). Instead, the power continues to flow up, and the optimal confinement is disturbed to a large extent. Table 4 shows that for 2GL – TM₂ the performance tolerance issue is problematic; in a half of the etch errors' combinations $|\Delta \eta_{\text{max}}|$ it exceeds 50%, rising up to ~86% in the worst case. This is not surprising due to its highest Q factor and very specific EM confinement mechanism, which are extra sensitive to the structural dimensions.

6. APPLICATIONS

Due to Table 2, the designed Si-photonics structures appear to be loaded in high-Q-factor and ~100%-peak-absorptance resonators for the chosen λ_0 , and quite similar design can be carried out for any higher NIR wavelength below 1 µm. Hence, due to their high efficiency and ultra-small thickness, these proposed structures can be used as efficient and thin PDs in FPAs, on-chip integrated Si photonics, and precise spectroscopic chemical sensing. When unloaded, the designed double-GL cavities exhibit a ~100% narrowband transmission notch at



Fig. 10. Possible placement of electrical contacts to the photoactive Si layer in the PD applications of the dual-GL structure, sketched in Fig. 1. The placement is appropriate for the BSI considered in this study as described in the text.

 λ_0 , so they may be used as ultra-narrowband optical filters for a wide range of applications.

For the PD applications, electrical contacts need to be connected to the Si layer. Close methods to this end were developed for RCE-PDs, both photodiode and photoconductor ones, e.g., Refs. [3] and [28,41], respectively, and waveguide-coupled PDs, e.g., Refs. [7,23]. Briefly, the technique amounts to *ad hoc* making the part below the front cavity wider than that above the Si layer, on both sides in the *x* direction, and then depositing the poly-Si/metal contacts atop the protruding Si strips as shown in Fig. 10. Since the structures are a huge times λ_0 (in our case, also Λ) wide, the results obtained without the width offset prove valid with it as experimentally verified [26,28,41].

Upon an offset of the light incidence from the normal to an oblique small angle θ , the absorption peaks split into two peaks at some $\lambda_0^- < \lambda_0$ and $\lambda_0^+ > \lambda_0$ for the TE structures and shift to positions at some $\lambda'_0 > \lambda_0$ for the TM structures; see the example in Fig. 11 for 2GL – TM₂. The splitting $\delta_{\text{TE}} = \lambda_0^+ - \lambda_0^-$ and shift $\delta_{\text{TM}} = \lambda'_0 - \lambda_0$ fit accurately with linear and quadratic regressions, respectively, viz. $\delta_{\text{TE}} = a\theta$ and $\delta_{\text{TM}} = b\theta^2$. Here *a* and *b*, each with less than 2% maximum



Fig. 11. Efficiency spectra θ dependence of the 2GL – TM₂ structure. The structural parameters are found in Table 2.

standard relative error, equal 9.818 nm/deg for $2GL - TE_1$, 16.19 nm/deg for $2GL - TE_2$, 2.366 nm/ deg² for $2GL - TM_1$, and 2.395 nm/ deg² for $2GL - TM_2$. Both δ_{TE} and δ_{TM} performances can be easily calibrated, provided that the quantities are measurable. Thus, the considered dual-GL cavity structures may be used for high-precision angular-position sensing.

The above noted $\eta(\lambda)$ sensitivity to a varying incidence angle θ may also be beneficial for the WDM, highly important in contemporary optical communication. The exponential growth in the need for ultra-fast and ultra-wideband communication stimulated the development of the WDM. It provides an increase of the capacity by multiplexing a number of optical carriers using multiple wavelengths onto a single optical guide. With the existing electrical and optical infrastructure for long-haul communication, the WDM approach has proven to be highly cost effective and to benefit, for an efficient operation, from such optical components as ultra-narrowband transmitters, receivers, and filters.

The evolving DWDM requires channel separation that can reach higher than 50 GHz in frequency or lower than 0.8 nm in wavelength. This demands ultra-narrow and stable components acting as demultiplexers to separate each wavelength with minimum crosstalk. Structures such as those proposed and designed in this work can help address this challenge due to their high Q, $\eta_{\rm max}$, and θ sensitivity. A possible approach for separating the wavelength channels is directly by the dependence of the absorptance spectra on θ . For example, as can be inferred from Fig. 11, 2GL - TM₂ provides ~0.1 nm separation around 0.8 µm. Though our structures can function as DWDM devices only for integrated-circuit short- to medium-haul communication in the range 0.8 μ m $\leq \lambda < 1 \mu$ m, principles quite similar to those developed in this work can be applied for a higher efficiency redesign of the detectors currently in use for the conventional communication band's DWDM, such as the monolithic InGaAs.

The emerging field of optical camouflage would also benefit from designs inspired by light-absorbing structures such as those considered in this study, which are ultrathin, subwavelength, and perfectly absorptive. For this purpose, managing the light power to flow laterally in a thin buried absorbing channel, as Figs. 8(b) and 9(a) demonstrate, seems indispensable.

7. CONCLUSIONS

We proposed and theoretically investigated a new type of enhanced light-absorption structures for NIR Si-photonics (see Table 2), comprising only Si-microelectronics-compatible materials (SiO₂ and Si₃N₄) and Si (see Table 1) as the NIR light absorber, in contrast with, e.g., Refs. [5,6]. Inspired by the RCE-PD, this structure contains the cavity–absorber (Si) wrap enclosed with the subwavelength GLs. We argued that the assembly of the RCE-PD with the same materials and topmost absorptance is beyond the scope of feasibility. On the contrary, the subwavelength dual-GL cavity enhanced ultrathin structures, attaining ~100% ultra-narrowband $\eta(\lambda)$ peak at the near IR, were shown to be feasible with the in-house software design.

We extensively shed light on the multiple physical mechanisms behind the record $\eta_{\rm max}$ of the designed structures, visualizing the resonant EM field amplitudes and power flow over the important space regions inside the structures. We concluded that the EM fields in the cavity-absorber region are tightly 2D confined, which combines the sharpened vertical confinement of the resonant-cavity type with lateral confinement set by the GLs' derived microcavity. Also, we gave novel comprehensive and structure-peculiar insights into the EM power transfer, which combines vortex and junction crossed waveguide-like flows. The flows point at optical singularities, at which $\mathbf{E} = \mathbf{0}$, such as the vorticity centers and junctions, which connects this study to singular optics [25]. These combined singular optical phenomena proved to be the keys to the perfect optical absorption of each of the designed and simulated structures. Furthermore, we tested the structures' performance tolerances to the grating fabrication errors, and found them to be superlative for $2GL - TE_{1,2}$; see Table 3. Among the TM-polarization targeted dual-GL enhanced structures, 2GL – TM₁ tolerates most of the errors an order of magnitude weaker than its TE-polarization counterpart, still exhibiting a practical performance.

Finally, in Section 6, we discussed at length possible applications of the proposed structures. To our knowledge, using the metasurfaces proposed and scrupulously theoretically investigated in this study conceptually advances design of topmost optical-absorption structures. Last but not least, this study brings a new focus on striking features of the EM field distribution and Poynting vector flow over the metasurfaces-confined micro-structures. The design principles and subsequent analyses developed in this study may be implemented in other fields of the micro- and nano-optics research and applications.

APPENDIX A: GRATING-ON-LAYER DESIGN

Starting values of the GLs' parameters, needed in the computeraided design input as shown in Fig. 2, are obtained using a trial semianalytical design on the basis of effective-medium and grating-waveguide resonance theories. In these designs, we approximate the true standalone GLs shown in Figs. 12(a) and 13(a) by effective bilayers as shown in Figs. 12(b) and 13(b), respectively, in which the binary-grating layers are described by effective RIs. These RIs n_{ν} depend on one of the principal orthogonal light polarizations, viz. $\nu = s$ (TE), p (TM).

Let permittivity of the grating line be ε_1 and the groove be ε_g ; the line duty cycle is f. Then one can use the effective-



Fig. 12. Front standalone GL: (a) the original; (b) an effective-medium bilayer substitute.

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Fig. 13. Back standalone GL: (a) the original; (b) an effective-medium bilayer substitute.

medium theory (EMT) to obtain $n_{\nu} = \sqrt{\varepsilon_{\nu}}$ via a series in powers of $(\Lambda/\lambda_0)^2$ [42]:

$$\varepsilon_{\nu} = \varepsilon_{\nu,0} + \Delta \varepsilon_{\nu,2} + o((\Lambda/\lambda_0)^2),$$
 (A1)

in which the zeroth-order (achromatic) approximations read

$$\varepsilon_{\rm s,0} = f\varepsilon_{\rm l} + (1-f)\varepsilon_{\rm g}, \qquad \varepsilon_{\rm p,0} = \frac{\varepsilon_{\rm g}\varepsilon_{\rm l}}{f\varepsilon_{\rm g} + (1-f)\varepsilon_{\rm l}}, \quad (A2)$$

and the subsequent second-order corrections are given by

$$\Delta \varepsilon_{s,2} = \frac{\pi^3}{3} f^2 (1 - f)^2 (\varepsilon_{\rm l} - \varepsilon_{\rm g})^2 \left(\frac{\Lambda}{\lambda_0}\right)^2,$$

$$\frac{\Delta \varepsilon_{\rm p,2}}{\Delta \varepsilon_{\rm s,2}} = \frac{\varepsilon_{\rm p,0}^3 \varepsilon_{\rm s,0}}{\varepsilon_{\rm g}^2 \varepsilon_{\rm l}^2}.$$
 (A3)

1. High-Reflection GL Design

In this design, Λ/λ_0 is given, and EMT ε_{ν} , t_1 , and t_2 are estimated with a grating-waveguide resonance theory [43], in which the grating is not completely smoothed out. Rather, it adds to the persistent non-diffracted zeroth-order wave two virtual counter-propagating ± 1 st-order ones with the lateral wavenumbers $k_x = \pm 2\pi/\Lambda$. These waves may be guided in the layers shown in Fig. 12(b), since $n_{\text{Si}_3\text{N}_4} > n_{\text{SiO}_2}$, and for this grating $n_{\text{Si}_3\text{N}_4} > n_{\nu} > 1$. Interference between the non-diffracted guided waves is the mechanism of the high reflectance [43].

For example, at $\nu = s$, complying the diffracted waves with the guided modes yields the following equation:

$$i\beta_{2}(\beta_{1}^{2} + \beta_{0}\beta_{3}) \tan k_{0}\beta_{1}t_{1} + i\beta_{1}(\beta_{2}^{2} + \beta_{0}\beta_{3}) \tan k_{0}\beta_{2}t_{2} + (\beta_{1}^{2}\beta_{3} + \beta_{2}^{2}\beta_{0}) \tan k_{0}\beta_{1}t_{1} \tan k_{0}\beta_{2}t_{2} = \beta_{1}\beta_{2}(\beta_{3} + \beta_{0}),$$
(A4)

where $k_0 = 2\pi/\lambda_0$, and the propagation constants are given by

$$\beta_0 = \sqrt{1 - (\lambda_0/\Lambda)^2}, \qquad \beta_3 = \sqrt{\varepsilon_{\text{SiO}_2} - (\lambda_0/\Lambda)^2},$$

$$\beta_1 = \sqrt{\varepsilon_s - (\lambda_0/\Lambda)^2}, \qquad \beta_2 = \sqrt{\varepsilon_{\text{Si}_3\text{N}_4} - (\lambda_0/\Lambda)^2}.$$
 (A5)

Here ε_s relates to the front grating, β_0 is imaginary, and, concurrently at $n_{\text{SiO}_2} < \lambda_0/\Lambda < n_s$, β_3 is imaginary, and $\beta_{1,2}$ are both real. Then, Eq. (A4) is satisfied by infinitely many pairs of real t_1, t_2 . Solving Eq. (A4) supplies us the trial parameters $t_{\text{fg}}^*, t_{\text{Si}_nN_4}^*$ to be used in the optimization.

2. Low-Reflection GL Design

Due to the theory of optical multilayers [37], for the antireflection of the bilayer with design $t_2 = t_{qw} = \lambda_0/(4n_{Si_3N_4})$, it is uniquely required that $t_1 = t_{qw}$ and $n_{\nu} = n_{Si_3N_4}$. Yet, such RI matching seems physically dubious, whichever EMT approximation for ε_{ν} is adopted. However, given a Λ^* compatible with the solubility of Eq. (A4), one can increase $\varepsilon_{\nu,2}$ towards $\varepsilon_{Si_3N_4}$ using Eqs. (A1)–(A3) by tuning f. This may prove possible since, due to Eq. (A3), the second-order correction increases ε_{ν} beyond the achromatic EMT approximation given by Eq. (2). The assessed Λ^* and $W^* = (1 - f^*)\Lambda^*$, where f^* is an adopted value of f and $t_{bg}^* = t_{qw}$, all obtained with the data of Table 1, are used among the starting values in the optimization procedure.

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