

# ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod array photoanode for dye-sensitized solar cells

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**Abstract** In this paper, ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod arrays were synthesized and used as photoanodes for dye-sensitized solar cells (DSSCs). We first synthesized ZnO nanorod array on fluorine-doped tin oxide (FTO) glasses by a hydrothermal method, and then ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod array was directly obtained via solvothermal reaction in NbCl<sub>5</sub> solution. The scanning electron microscope (SEM) and transmission electron microscope (TEM) images revealed that the ZnO nanorods were uniformly wrapped by Nb<sub>2</sub>O<sub>5</sub> shell layers with a thickness of 30–40 nm. Photovoltaic characterization showed that the device based on ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod photoanode exhibited an improved efficiency of 1.995%, which was much higher than the efficiency of 0.856% for the DSSC based on bare ZnO nanorod photoanode. This proved that the photovoltaic performance of ZnO nanorods could be improved by wrapping with Nb<sub>2</sub>O<sub>5</sub> shells.

**Keywords** ZnO, Nb<sub>2</sub>O<sub>5</sub>, core/shell nanorods, solvothermal, dye-sensitized solar cell (DSSC)

## 1 Introduction

As a promising alternative to conventional silicon-based solar cell, dye-sensitized solar cells (DSSCs) have attracted considerable attention in the last two decades, because of their low manufacturing costs and relatively high energy-conversion efficiencies [1–4]. In DSSC components, the photoanode is a key part that plays an essential role in determining the dye loading and electron transportation, and hence the photon to electron conversion efficiency. To realize highly efficient DSSC, a nanostructured photoanode should possess several favorable intrinsic characteristics, such as large surface area to permit high dye loading, direct electron transport pathways for long electron

diffusion lengths, and compatible energy levels to achieve high electron injection efficiency and high voltage. It is therefore highly desirable to develop a photoanode that meets the above requirements.

There have been many efforts to develop efficient photoanode materials such as TiO<sub>2</sub>, ZnO, SnO<sub>2</sub>, Nb<sub>2</sub>O<sub>5</sub>, and other composites. So far, TiO<sub>2</sub> has been widely investigated as a photoanode material for DSSCs. With a band gap similar to that of TiO<sub>2</sub>, ZnO is another alternative photoanode material, which has attracted much attention due to its high mobility of about 115–155 cm<sup>2</sup>·V<sup>-1</sup>·s<sup>-1</sup>, which is much higher than that of TiO<sub>2</sub> (10<sup>-5</sup> cm<sup>2</sup>·V<sup>-1</sup>·s<sup>-1</sup>) [5,6]. Among the various morphologies, one-dimensional (1D) nanostructures such as nanorods, nanotubes, and nanowires, which can offer more superior electron transport pathways, have been attracting increasing attention. However, the power conversion efficiencies of DSSCs based on ZnO 1D nanostructures are still at relatively low levels, because of the insufficient internal surface area.

To overcome this shortcoming, it is necessary to design a novel core/shell photoanode by applying metal oxide shells such as SiO<sub>2</sub> [7], ZrO<sub>2</sub> [8], Nb<sub>2</sub>O<sub>5</sub> [9–12], and Al<sub>2</sub>O<sub>3</sub> [13] to the core nanostructures. The metal oxide shells synthesized on core nanostructures could increase the specific area of the electrode for dye loading, thereby enhancing the photocurrent density. Moreover, the layer formed by coating with these materials act as an energy barrier that decreases the electron recombination losses, shifts the conduction band downward, which increases the electron injection, and enhances the injection efficiency.

Nb<sub>2</sub>O<sub>5</sub> is a promising metal oxide because it supports good N719 dye loading due to its basic character [14–17], and its conduction band level is more negative than that of TiO<sub>2</sub> [18,19]. Here, we applied a Nb<sub>2</sub>O<sub>5</sub> coating on ZnO nanorod to obtain ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod arrays, which were used as the photoanodes in DSSCs. The results showed that improved efficiencies of 1.609% and 1.995% were obtained for DSSCs based on ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell

nanorod synthesized in low concentration  $\text{NbCl}_5$  solution and in high-concentration  $\text{NbCl}_5$  solution, respectively, compared with the efficiency of 0.856% for the DSSC based on bare ZnO nanorod photoanode.

## 2 Experimental

### 2.1 Synthesis of ZnO nanorod arrays

The ZnO seed layer on fluorine-doped tin oxide (FTO) glasses was prepared using a previously published procedure [20]. Then, 0.455 g  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  and 0.555 g  $\text{NH}_4\text{F}$  were dissolved into 50 mL deionized water. After that,  $\text{NH}_3 \cdot \text{H}_2\text{O}$  was dropped into the above solution while stirring. In this process, the clear solution turned turbid when  $\text{NH}_3 \cdot \text{H}_2\text{O}$  was first dropped into the solution, and then gradually it turned clear with continued stirring. Subsequently, the prepared ZnO seed layer on FTO glass (with the FTO layer facing downwards) was placed into a Teflon-lined autoclave, which contained the above solution, and was then sealed. Finally, the Teflon-lined autoclave was heated at  $70^\circ\text{C}$  for 8 h to obtain ZnO nanorod arrays.

### 2.2 Synthesis of ZnO/ $\text{Nb}_2\text{O}_5$ core/shell nanorod arrays

Figure 1 shows the process of growing ZnO/ $\text{Nb}_2\text{O}_5$  core/shell nanorod arrays on FTO glass. First, 0.135 and 0.270 g  $\text{NbCl}_5$  were dissolved into 50 mL anhydrous ethanol. After that, the obtained ZnO nanorod arrays were placed into the

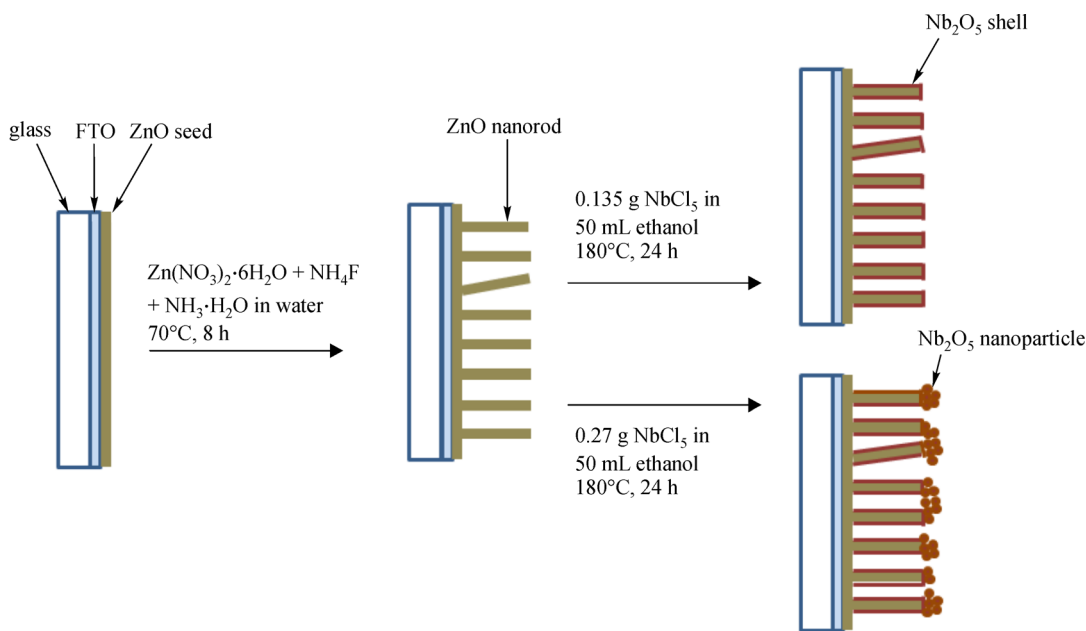
two Teflon-lined autoclaves, each containing one of the above two  $\text{NbCl}_5$  solutions. Then, the Teflon-lined autoclaves were heated at  $180^\circ\text{C}$  for 24 h to obtain ZnO/ $\text{Nb}_2\text{O}_5$  core/shell nanorod arrays with ZnO/ $\text{Nb}_2\text{O}_5(1)$  for 0.135 g  $\text{NbCl}_5$  and ZnO/ $\text{Nb}_2\text{O}_5(2)$  for 0.270 g  $\text{NbCl}_5$ . Finally, the ZnO/ $\text{Nb}_2\text{O}_5$  core/shell nanorod arrays were sintered at  $500^\circ\text{C}$  in air for 1 h.

### 2.3 Fabrication of DSSCs based on ZnO/ $\text{Nb}_2\text{O}_5$ core/shell nanorod arrays

The ZnO/ $\text{Nb}_2\text{O}_5$  core/shell nanorod arrays were dipped in a dye solution containing  $0.5 \text{ mM}^{1)} \text{N719}$  (Dyesol) dye for 24 h. Then, the ZnO/ $\text{Nb}_2\text{O}_5$  electrode was scrapped to obtain an active area of  $25 \text{ mm}^2$ . The counter electrode was prepared by coating a  $0.6 \text{ mM} \text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$  solution in anhydrous ethanol onto the FTO substrate. After that, the ZnO/ $\text{Nb}_2\text{O}_5$  electrode was assembled with the counter electrode by clamping a  $25 \mu\text{m}$  thick polymeric film (Surlyn, DuPont). Then, an electrolyte solution was injected into the gap between the ZnO/ $\text{Nb}_2\text{O}_5$  electrode and the counter electrode. The electrolyte solution contained  $0.05 \text{ M LiI}$  (Sigma-Aldrich),  $0.03 \text{ M I}_2$  (Aldrich), and  $0.5 \text{ M 4-tert-butylpyridine}$  (Aldrich) in a solution containing acetonitrile (Aldrich). Finally, the injecting hole was sealed with an adhesive tape to obtain the completed device.

### 2.4 Characterization

A scanning electron microscope (SEM, Quanta200, FEI,



**Fig. 1** Schematic diagram of the ZnO/ $\text{Nb}_2\text{O}_5$  core/shell nanorod arrays

1)  $1 \text{ mM} = 1 \text{ mmol/L}$

Netherlands) and a transmission electron microscope (TEM, JEOL TEM-2010) were used to measure the structure and morphology of the nanomaterials. The current–voltage characterization was performed using a Keithley 2400 source meter under simulated AM 1.5 sunlight illumination ( $100 \text{ mW} \cdot \text{cm}^{-2}$ ) provided by an Oriol solar simulator (Model 9119X, Newport Co.). The illuminated active area of the photovoltaic measurements was  $0.16 \text{ cm}^2$ . The electrochemical impedance spectra (EIS) of the devices were tested at  $-0.6 \text{ V}$  for the range from 1 MHz to 1 Hz, with an advanced electrochemical system (PAR2273) under dark conditions.

### 3 Results and discussion

#### 3.1 Structural and morphological characterization of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod

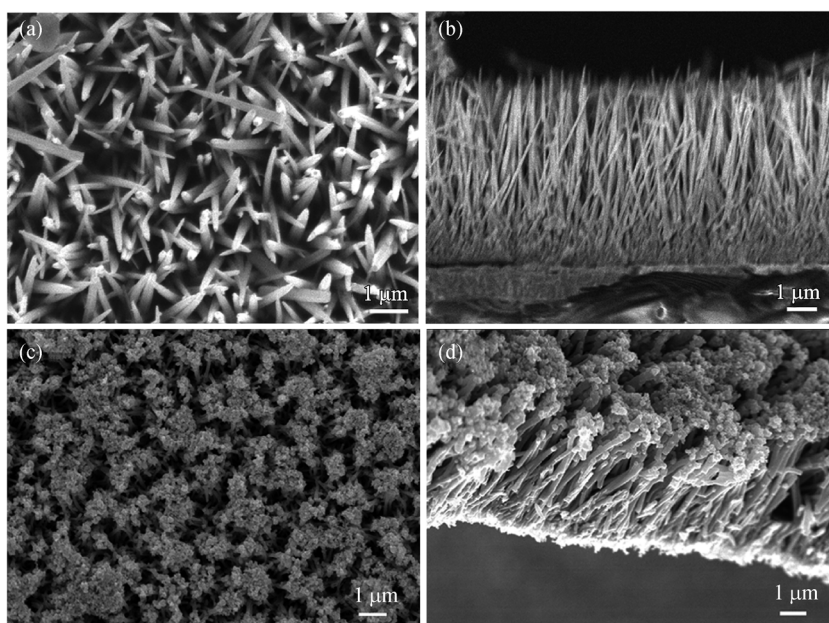
Figures 2(a) and 2(b) show the top-view and cross-view SEM images of the ZnO nanorod arrays, respectively. Figure 2(a) reveals that the top of the nanorods is uniform. From the tilted-view SEM image, we can clearly see that the high-density ZnO nanorods grew vertically on the FTO substrate. Their length is about  $5 \mu\text{m}$ . After the coating of the Nb<sub>2</sub>O<sub>5</sub> shell, the core-shell structure was investigated by SEM. Figure 2(c) shows the top-view SEM image of the ZnO/Nb<sub>2</sub>O<sub>5</sub> core-shell nanorod film. It clearly reveals that the diameter of the core-shell nanorod increased and its surface became rough. The corresponding tilted-view SEM image shows that the core-shell structure has a similar shape as that of the ZnO nanorod shown in Fig. 2(d). To

investigate the coating of Nb<sub>2</sub>O<sub>5</sub>, the samples were synthesized in a low-concentration NbCl<sub>5</sub> solution (0.135 g NbCl<sub>5</sub>) and a higher-concentration NbCl<sub>5</sub> solution (0.270 g NbCl<sub>5</sub>). The results are shown in Fig. 3. By comparison, we could find that upon increasing the concentration of NbCl<sub>5</sub>, the ZnO nanorods became rougher on the side surface and were fully covered with Nb<sub>2</sub>O<sub>5</sub> nanoparticles on the top. This shows that a high concentration of NbCl<sub>5</sub> could facilitate significantly more growth of Nb<sub>2</sub>O<sub>5</sub> on the top of the nanorod than on its side.

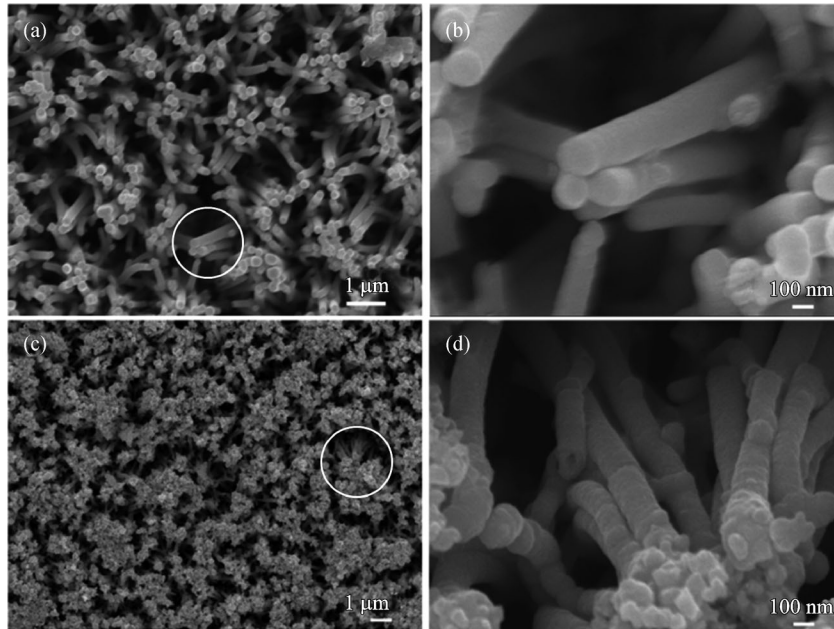
In the TEM image (Fig. 4(a)), obvious differences in terms of the contrast between the center and fringe parts of each individual rod-like structure are observed, which indicates that the rod-like structures are core-shell structures with core diameters of 150–200 nm and shell thicknesses of 30–40 nm. The corresponding selected area electron diffraction (SAED) pattern obtained from the circled area in Fig. 4(a) is shown in Fig. 4(b). We can see single crystal diffraction spots in Fig. 4(b), corresponding to ZnO (103) and (203). In addition, Fig. 4(b) also shows an amorphous diffraction pattern, which is indicated by the dotted line. We measured the radius of the amorphous diffraction ring, and the spacing between the crystal planes ranged from 0.39 to 0.28 nm, which was consistent with the strong peak position of JCPDS (320711) Nb<sub>2</sub>O<sub>5</sub>. Therefore, we inferred that the amorphous diffraction ring was from the amorphous Nb<sub>2</sub>O<sub>5</sub> wrapped on the surface of ZnO.

#### 3.2 Characterization of photovoltaic performances for DSSCs

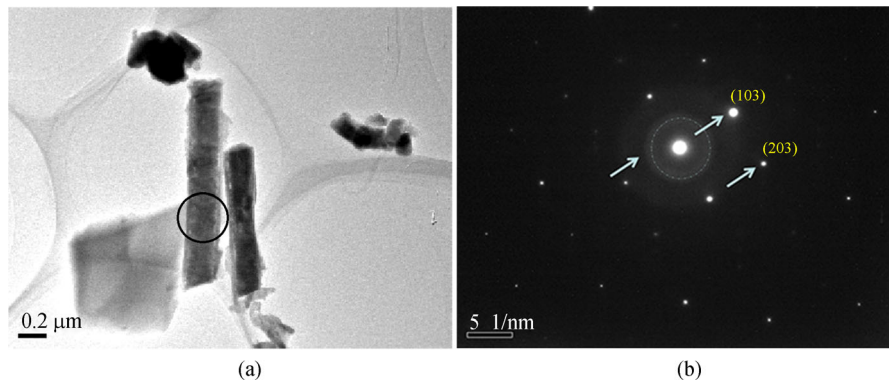
Figure 5 shows the optical absorption spectra of the ZnO



**Fig. 2** Top-view (a) and tilted-view SEM images (b) of bare ZnO nanorod arrays. The top-view (c) and tilted-view SEM images (d) of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod arrays obtained in high concentration NbCl<sub>5</sub> solution (0.270 g NbCl<sub>5</sub>)



**Fig. 3** Top-view SEM images of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod arrays synthesized in low concentration NbCl<sub>5</sub> solution (0.135 g NbCl<sub>5</sub>): (a) low and (b) high magnification. The top-view SEM images of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod arrays synthesized in high concentration NbCl<sub>5</sub> solution (0.270 g NbCl<sub>5</sub>): (c) low and (d) high magnification

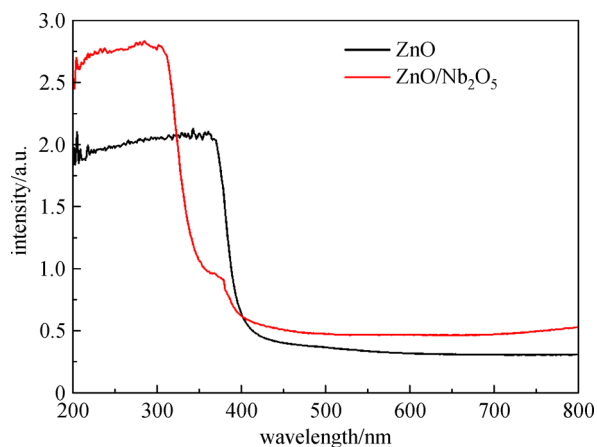


**Fig. 4** (a) TEM image of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod and (b) the corresponding SAED pattern obtained from the circle area in Fig. 4(a)

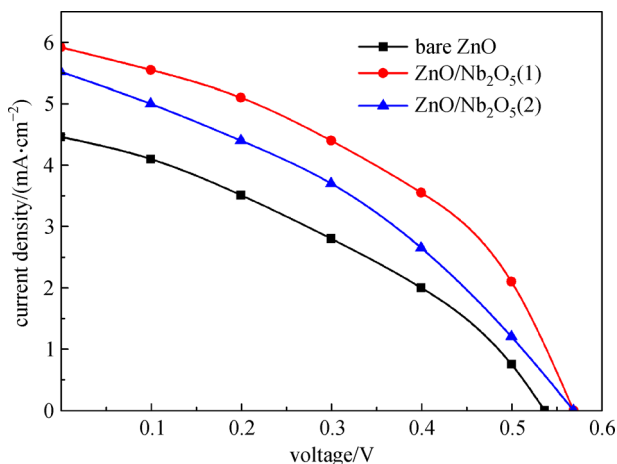
and ZnO/Nb<sub>2</sub>O<sub>5</sub> nanorod films. For the bare ZnO nanorod film, the onset of the band gap transition is at ~420 nm. The presence of Nb<sub>2</sub>O<sub>5</sub> shell increases another adsorption edge at ~370 nm relative to Nb<sub>2</sub>O<sub>5</sub>. Compared with the ZnO nanorod arrays, the coating of the Nb<sub>2</sub>O<sub>5</sub> moved the onset of absorption to a lower wavelength, and the light absorption intensity was enhanced. The larger band gap of Nb<sub>2</sub>O<sub>5</sub> semiconductor is related to the more negative conduction band potential and the larger open-circuit photovoltage of the Nb<sub>2</sub>O<sub>5</sub> cell. This result is consistent with the following photovoltaic performances.

The photocurrent density-voltage curves of DSSCs based on the photoanodes of bare ZnO nanorods, ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) core/shell nanorod (0.135 g NbCl<sub>5</sub>), and ZnO/

Nb<sub>2</sub>O<sub>5</sub>(2) core/shell nanorod (0.270 g NbCl<sub>5</sub>) are shown in Fig. 6. The photovoltaic performances of the three devices are listed in Table 1. The device based on the photoanode of bare ZnO nanorods showed poor performances with short-circuit current density ( $J_{sc}$ ), open-circuit voltage ( $V_{oc}$ ), fill factor ( $FF$ ), and power conversion efficiency (PCE) of 4.46 mA·cm<sup>-2</sup>, 537 mV, 0.357, and 0.856%, respectively. This efficiency was similar to the efficiencies of DSSCs based on bare ZnO nanorod arrays reported by other groups [21,22]. The device based on ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) core/shell nanorod exhibited improved performances with  $J_{sc}$ ,  $V_{oc}$ ,  $FF$ , and PCE of 5.52 mA·cm<sup>-2</sup>, 569 mV, 0.512, and 1.609%, respectively. The increased  $J_{sc}$  is mainly attributed to the increased surface area of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/



**Fig. 5** Optical absorption spectra of ZnO and ZnO/Nb<sub>2</sub>O<sub>5</sub> nanorod arrays



**Fig. 6** Photocurrent density-voltage curves of DSSCs based on the photoanodes of bare ZnO nanorods, ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) core/shell nanorod (0.135 g NbCl<sub>5</sub>) and ZnO/Nb<sub>2</sub>O<sub>5</sub>(2) core/shell nanorod (0.270 g NbCl<sub>5</sub>)

shell nanorod, which was caused by the small Nb<sub>2</sub>O<sub>5</sub> nanoparticles wrapped on the ZnO nanorod. The increased  $V_{oc}$  is attributed to the fact that the Fermi level of ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod is higher than that of bare ZnO nanorod, since Nb<sub>2</sub>O<sub>5</sub> has a higher conduction band than ZnO. This increases the  $V_{oc}$  value because the maximum  $V_{oc}$  of DSSC is mainly decided by the difference between

the Fermi level of the photoanode and the redox potential of  $I^-/I_3^-$ . In addition to the increased  $J_{sc}$  and  $V_{oc}$ , the device based on ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) core/shell nanorod exhibited a higher  $FF$  of 0.512 than the device based on ZnO nanorod (0.357). This indicates that the Nb<sub>2</sub>O<sub>5</sub> shell provided another charge-transporting channel other than the ZnO nanorod, resulting in decreased series resistance. As for the device based on ZnO/Nb<sub>2</sub>O<sub>5</sub>(2) core/shell nanorod, the  $J_{sc}$ ,  $FF$ , and PCE were further increased to 5.95 mA·cm<sup>-2</sup>, 0.592, and 1.995%, respectively. Compared with the ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) core/shell nanorod, the ZnO/Nb<sub>2</sub>O<sub>5</sub>(2) core/shell nanorod had a thicker diameter and had additional Nb<sub>2</sub>O<sub>5</sub> particles on top of the nanorod. Such a core/shell nanorod structure makes the ZnO/Nb<sub>2</sub>O<sub>5</sub> nanorod photoanode advantageous in the following two aspects. On the one hand, since the protons that are released from the dye molecules in the ethanolic solution dissolved ZnO to generate Zn<sup>2+</sup>-dye aggregates, the structure of the ZnO crystals was easily destroyed after loading the Ru-complex dyes. However, after being coated with Nb<sub>2</sub>O<sub>5</sub>, the recombination was suppressed by passivating its centers on the ZnO nanostructure surface. On the other hand, the electrolyte was suppressed presumably due to the energy barrier formed at the ZnO/Nb<sub>2</sub>O<sub>5</sub> interface. Thus,  $J_{sc}$  and  $V_{oc}$  of the ZnO/Nb<sub>2</sub>O<sub>5</sub> nanorod photoanode are much higher than that of the bare ZnO photoanode.

## 4 Conclusions

In summary, ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorods were successfully developed by solvothermal synthesis and applied as photoanodes for DSSCs. ZnO nanorod core was first synthesized by a hydrothermal process, and then Nb<sub>2</sub>O<sub>5</sub> shell was directly synthesized on the ZnO nanorod core by solvothermal reaction in NbCl<sub>5</sub> solution. SEM and TEM images revealed that the ZnO nanorods were uniformly wrapped by the Nb<sub>2</sub>O<sub>5</sub> shell layers with a thickness of 30–40 nm. Photovoltaic characterization showed that improved efficiencies of 1.609% and 1.995% were obtained for DSSCs based on ZnO/Nb<sub>2</sub>O<sub>5</sub> core/shell nanorod synthesized in low-concentration NbCl<sub>5</sub> solution and high-concentration NbCl<sub>5</sub> solution, respectively, compared with the efficiency of 0.856% for the DSSC based on bare ZnO nanorod photoanode. Our work offers a facile strategy to develop functional composite nanomaterials for photovoltaic devices.

**Table 1** Photovoltaic performance of DSSCs based on photoanodes of ZnO nanorods, ZnO/Nb<sub>2</sub>O<sub>5</sub>(1) nanorod and ZnO/Nb<sub>2</sub>O<sub>5</sub>(2) nanorod under AM1.5 conditions 100 mW·cm<sup>-2</sup>. The length of three nanorods in the devices are all around 5 μm

photoanode	$J_{sc}/(\text{mA} \cdot \text{cm}^{-2})$	$V_{oc}/\text{mV}$	$FF$	PCE/%
ZnO	4.46	537	0.357	0.856%
ZnO/Nb <sub>2</sub> O <sub>5</sub> (1)	5.52	569	0.512	1.609%
ZnO/Nb <sub>2</sub> O <sub>5</sub> (2)	5.92	569	0.592	1.995%



**Acknowledgements** The authors acknowledge the financial support from the National Natural Science Foundation of China (Grant Nos. 11647073, 11547263) and the financial support from High Education Natural Science Research Project of Jiangsu Province (No. 15KJB430033).

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