

# 大角度双波段探测成像分光器件的研制

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**摘要** 在微光夜视与红外成像融合的光学系统中,光通过55°放置的分光镜分成两束光,其中反射光被微光探测器 接收进行微光夜视成像,透射光被红外探测器接收进行红外成像,通过图像融合技术来提高系统的成像分辨率。针 对分光镜的参数要求,笔者选用Ge、ZnS和YbF。作为沉积材料,采用离子源辅助沉积技术在多光谱ZnS基底上制备 了0.6~0.9 μm波段高反、3.7~4.8 μm波段高透的分光膜。通过对膜系结构的优化以及沉积工艺参数的调整,解决 了膜层牢固度和面形精度等问题,实现了0.6~0.9 μm波段反射率为90.77%、3.7~4.8 μm波段透射率为91.15%的 分光指标。附着力测试、摩擦力测试、高低温测试、恒温恒湿测试结果显示所制备的双面膜可以满足使用要求,但该 膜的短波反射率和长波透过率仍有一定的提升空间。

关键词 光学器件;分光镜;微光夜视;中波红外成像;面形精度 中图分类号 O484 **文献标志码** A

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## 1引言

随着我国军用光学技术的不断更新迭代,单一波 段光谱已远远不能满足特定场景的使用需求,需要根 据不同波段成像质量的优劣选择多波段成像,从成像 技术上不断提高传感器的质量。因此,图像融合技术 越来越受到关注,并对光学系统中的光学元件提出了 更高要求<sup>[1-2]</sup>。

目前,多光谱融合技术已被广泛应用于军事作战、 医学影像分析等领域<sup>[34]</sup>。2014—2018年,我国科研人员<sup>[58]</sup>研究了 0.5~9.3 µm 波段范围内的双波段或三波 段增透膜系,其在该波段的平均透过率能达到 95% 以 上。2018年,Fleming等<sup>[9]</sup>采用脉冲磁控溅射的方式制 备了 4.2 µm 带通滤光膜。2021年,Guo等<sup>[10]</sup>研制了 0° 入射时,在 0.3~0.9 µm 波段的平均透射率低于 1% 且 在 3.7~4.8 µm 波段的平均透射率超过 92% 的滤 光片。

综上所述,国内外在红外图像融合领域进行了广 泛研究,但目前针对可见/近红外、中红外双波段大角 度分光镜的研究还鲜有报道,该研究对于双波段红外 探测成像技术具有重要的参考意义。

## 2 材料选取

辐射传输常用的红外基底有 Ge、Si、ZnS、ZnSe、 蓝宝石(Al<sub>2</sub>O<sub>3</sub>)等。分析后认为 ZnS 更适合作为基 底。相较于普通的 ZnS,多光谱 ZnS 具有透射波段更 宽、吸收更小、透过率更高的特点<sup>[11]</sup>,如图 1(a)所示。 红外波段常见的镀膜材料的可选择性较小,大部分在 可见波段有吸收。能满足覆盖可见到中红外波段的 低折射率薄膜材料主要有 ThF<sub>4</sub>、YbF<sub>3</sub>和 YF<sub>3</sub>,其中: ThF<sub>4</sub>有放射性;YbF<sub>3</sub>和 YF<sub>3</sub>的光学性质相似,但对于 采用同种工艺镀制的相同厚度的 YbF<sub>3</sub>和 YF<sub>3</sub>来说, 后者的应力更大,更容易出现脱膜裂膜现象。因而, 本次实验选用 YbF<sub>3</sub>作为低折射率材料。采用 Macleod 对 YbF<sub>3</sub>单层膜进行拟合得到了其折射率曲 线,如图 1(b)所示。

考虑到薄膜的物理特性以及基底本身就是ZnS, 分光面上的高折射率材料选择ZnS。Ge的折射率高, 且与ZnS匹配度高,故增透膜中的高折射率材料选择 Ge。采用Macleod对单层膜进行拟合得到的薄膜的折 射率如图2所示。

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图 2 薄膜的折射率。(a) Ge;(b) ZnS Fig. 2 Refractive index of two films. (a) Ge; (b) ZnS

## 3 膜系设计

分光镜是探测系统中的关键部件,其薄膜的主要 技术参数如表1所示。

对表1所示技术指标进行分析可知,该膜系短波 高反长波高透,类似于一个基础结构为(0.5HL0.5H)<sup>s</sup> 的长波通。对称膜堆用矩阵方法进行计算,公式<sup>[12]</sup>为

$$\boldsymbol{M}_{\mathrm{H}} = \begin{bmatrix} \cos \delta_{\mathrm{H}} & \frac{\mathrm{i} \sin \delta_{\mathrm{H}}}{\eta_{\mathrm{H}}} \\ \mathrm{i} n_{\mathrm{H}} \sin \delta_{\mathrm{H}} & \cos \delta_{\mathrm{H}} \end{bmatrix}, \qquad (1)$$

$$\boldsymbol{M}_{\mathrm{L}} = \begin{bmatrix} \cos \delta_{\mathrm{L}} & 1 \sin \delta_{\mathrm{L}} \\ i n_{\mathrm{L}} \delta \sin \delta_{\mathrm{L}} & \cos \delta_{\mathrm{L}} \end{bmatrix}, \qquad (2)$$

$$\boldsymbol{M} = \boldsymbol{M}_{\mathrm{H}} \boldsymbol{M}_{\mathrm{L}} \boldsymbol{M}_{\mathrm{H}} \cdots \boldsymbol{M}_{\mathrm{H}} \boldsymbol{M}_{\mathrm{L}} \boldsymbol{M}_{\mathrm{H}} = \begin{bmatrix} m_{11} & m_{12} \\ m_{21} & m_{22} \end{bmatrix}, \quad (3)$$

表1 薄膜的技术指标 Table 1 Technical specifications of film

Parameter	Indicator		
Spectrum range /µm	0.6-0.9, 3.7-4.8		
Reflectivity $R$	>90%		
Transmittance $T$	>90%		
Incident angle /(°)	$55\pm3$		
Peak-valley (PV) value of spectral surface $/\lambda$	≪0.2		
PV value of antireflection surface $/\lambda$	≪0.4		

式中: $\delta_{H}$ 、 $\delta_{L}$ 分别表示高折射率材料和低折射率材料的相位差; $n_{H}$ 、 $n_{L}$ 分别表示高低折射率材料的折射率; $\eta$ 表示材料的折射率(包括斜入射情况)。通过矩阵运算可求得

$$m_{11} = m_{22} = \cos(2\delta_{\rm H})\cos\delta_{\rm L} - \frac{1}{2}\left(\frac{\eta_{\rm L}}{\eta_{\rm H}} + \frac{\eta_{\rm H}}{\eta_{\rm L}}\right)\sin(2\delta_{\rm H})\sin\delta_{\rm L}, \qquad (4)$$

$$m_{12} = \frac{\mathrm{i}}{\eta_{\mathrm{H}}} \left[ \sin(2\delta_{\mathrm{H}}) \cos \delta_{\mathrm{L}} + \frac{1}{2} \left( \frac{\eta_{\mathrm{H}}}{\eta_{\mathrm{L}}} + \frac{\eta_{\mathrm{L}}}{\eta_{\mathrm{H}}} \right) \cos(2\delta_{\mathrm{H}}) \sin \delta_{\mathrm{L}} + \frac{1}{2} \left( \frac{\eta_{\mathrm{H}}}{\eta_{\mathrm{L}}} + \frac{\eta_{\mathrm{L}}}{\eta_{\mathrm{H}}} \right) \sin \delta_{\mathrm{L}} \right], \tag{5}$$

$$m_{21} = \frac{\mathrm{i}}{\eta_{\mathrm{H}}} \left[ \sin\left(2\delta_{\mathrm{H}}\right) \cos\delta_{\mathrm{L}} + \frac{1}{2} \left(\frac{\eta_{\mathrm{H}}}{\eta_{\mathrm{L}}} + \frac{\eta_{\mathrm{L}}}{\eta_{\mathrm{H}}}\right) \cos\left(2\delta_{\mathrm{H}}\right) \sin\delta_{\mathrm{L}} - \frac{1}{2} \left(\frac{\eta_{\mathrm{H}}}{\eta_{\mathrm{L}}} - \frac{\eta_{\mathrm{L}}}{\eta_{\mathrm{H}}}\right) \sin\delta_{\mathrm{L}} \right]_{\circ}$$
(6)

当|m<sub>11</sub>|>1时,等效导纳为虚数,即当膜层足够多时该波段为截止带;当|m<sub>11</sub>|<1时,等效导纳为实数,该

波段为透射带;当|m<sub>11</sub>|=1时,表示该波段是透射带向 截止带过渡的波段,如图3所示。



图 3 对称周期膜系在垂直入射时的等效折射率*E*<sup>[13]</sup> Fig. 3 Equivalent refractive index *E* of symmetrical periodic film systems under vertical incidence<sup>[13]</sup>

长波通光谱曲线的反射带波数宽度∆g为

$$\Delta g = \frac{2}{\pi} \arcsin\left(\frac{n_{\rm H} - n_{\rm L}}{n_{\rm H} + n_{\rm L}}\right),\tag{7}$$

波长宽度为

$$\Delta \lambda = 2 \Delta g \lambda_0, \tag{8}$$

式中: $n_{\rm H}$ 、 $n_{\rm L}$ 表示 0°入射时高低折射率材料的折射率; $\lambda_0$ 表示参考波长。

当高折射率材料(H)为ZnS,低折射率材料(L)为

YbF<sub>3</sub>时,长波通截止带波长宽度约为100 nm,考虑55<sup>°</sup> 入射,需要4个膜堆叠加,故膜系基础结构设计为Sub[1.4(0.5HL0.5H)<sup>10</sup> 1.26(0.5HL0.5H)<sup>10</sup> 1.14(0.5HL 0.5H)<sup>10</sup> (0.5HL0.5H)<sup>10</sup> [Air,其中Sub表示多光谱ZnS基板,Air表示空气。采用膜系设计软件进行优化,得到膜系结构为Sub[0.494H1.0149L2.38H0.563L···· 0.467L0.77H2.8L0.482H]Air,其理论光谱曲线如图4(a)所示,其总层数为80层,总厚度约为10  $\mu$ m。





Fig. 4 Theoretical spectra of spectral surface and antireflection surface with 55° incidence. (a) Spectral surface; (b) antireflection surface

基板的另一面需要设计  $3.7 \sim 4.8 \ \mu m 增透膜。依$ 据增透膜设计理论,选择基础膜系 Sub|(HL)<sup>2</sup> M|Air进行优化,得到膜系结构为 Sub|<math>0.27H2.04L0.147H0.452 L2.188M|Air,其中H表示高折射率材料 Ge,L表示中 折射率材料 ZnS,M表示低折射率材 YbF<sub>3</sub>。经过实验 验证,将 YbF<sub>3</sub>作为最外层会出现划痕、龟裂和脱膜等 问题,因此,在最外层增加 ZnS层,这样既能起到保护 作用还能降低 YbF<sub>3</sub>的厚度。最终的设计膜系为 Sub| 0.414H1.742L0.508H 1.666M 0.361L|Air,设计光谱曲 线如图 4(b)所示。

双面设计的理论光谱曲线如图 5 所示,其中 0.6~ 0.9 μm 波段的平均反射率为 98.8%, 3.7~4.8 μm 波段





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的	平均透射率为96.3%。	该设备配备霍尔离子源和 XTC/3 晶体膜厚控制
4	镀膜实验	系统,Ge薄膜采用电子束蒸发制备,ZnS、YbF。薄膜采用电阻蒸发制备。Ge、ZnS、YbF。薄膜的沉积
	镀膜实验在OTFC-1300蒸发镀膜机上完成。	工艺参数如表2所示。

	表2	Ge、ZnS	S、YbF₃薄	膜的沉淀	积工	艺参	数	
Table 2	Dep	osition p	arameters	of Ge,	ZnS	and	YbF <sub>3</sub>	film

Film layer	Deposition rate/ $(nm \cdot s^{-1})$	Chamber pressure/ $(10^{-4} \text{ Pa})$	Rotation rate/ $(rad \cdot min^{-1})$	Deposition temperature /°C
Ge	0.5	4	25	160
ZnS	2	4	25	160
$YbF_3$	0.4	4	25	160

镀膜前恒温1h,采用离子源清洗基板15 min。为 保证薄膜材料与基板的结合度,同时保证二者应力匹 配,全程使用离子源辅助沉积。ZnS和YbF<sub>3</sub>在不同蒸 发条件下的应力值不同。与电子束加热蒸发相比,电 阻蒸发制备的薄膜的应力较小,折射率更低,因此采用 电阻蒸发的方式沉积。真空退火工艺对应力、膜层牢 固度、膜层表面形貌均有很大影响。镀膜结束后在 200℃恒温2h,随后自然降温至90℃,放气<sup>[14+6]</sup>。离子 源沉积工艺参数如表3所示

表3 离子源沉积工艺参数

Table 3	e 3 Process parameters of ion source deposition				
Material	Anodic voltage /V	Ion beam current /mA	Flow of argon gas /(mL•min <sup>-1</sup> )		
Ge	150	1	20		
ZnS	150	1	20		
YbF <sub>3</sub>	200	5	20		
Cleaning	180	3	25		

# 5 测试与分析

## 5.1 应力检测与分析

蒸发镀制的ZnS薄膜中的应力通常为压应力,而

蒸发镀制的YbF<sub>3</sub>薄膜中的应力通常为张应力。依据 控制变量的方法,通过改变沉积工艺来调整相应厚度 ZnS、YbF<sub>3</sub>单层膜的应力值。由于膜系是高低折射率 材料交替沉积而成的,张应力和压应力可以相互抵消, 因此相应厚度的高低折射率材料的单层膜的应力值越 接近,基板的形变量就越小。根据测得的应力分析膜 系结构,在设计上考虑双面薄膜应力能够相互抵消或 趋于更小。应力计算需要借助Stoney公式,其表达 式<sup>[17]</sup>为

$$R = \frac{1}{2} \left( \frac{r^2}{h} + h \right) \approx \frac{r^2}{2h},\tag{9}$$

$$\sigma = \frac{E_z d_z^2}{6(1-v^2)d_f} \left(\frac{1}{R_f} - \frac{1}{R_0}\right), \quad (10)$$

式中:R为曲率半径; $\sigma$ 为薄膜的应力值;h为干涉仪测 得的 power 值; $E_z$ 为基板的杨氏模量; $d_z$ 为基板的厚 度;v为基板的泊松比; $d_i$ 为薄膜厚度; $R_o$ 为镀膜前基板 的曲率半径; $R_i$ 为镀膜后基板的曲率半径;r为基板 半径。

利用激光干涉仪测量两片Ø30 mm圆形硫化锌基 底沉积单层薄膜前后的面形,测试结果和计算结果如 表4所示,其中 $\lambda$ =632.8 nm。

	表4 单层膜的应力变化表	
Table 4	Stress variation table of single-layer membra	ne

Film layer	Film thickness /nm	Powe	er /λ	Stress /MPa		
		Before evaporation	After evaporation	Before process adjustment	After process adjustment	
ZnS	4261.6	0.138	-1.24	180	208	
YbF <sub>3</sub>	6127.4	0.172	1.47	276	222	

分析工艺调整前的参数可知,单层ZnS的应力值 与单层YbF<sub>3</sub>的应力值不匹配。调整ZnS和YbF<sub>3</sub>的沉 积速率以及离子源辅助工艺,将ZnS的沉积速率由 2 nm/s调整为1.5 nm/s,YbF<sub>3</sub>离子源辅助沉积参数由 200 V、5 A调整为220 V、5 A。调整工艺参数后,单层 ZnS与单层YbF<sub>3</sub>的应力值接近。 基底镀膜前,测得表面面形的峰谷值(PV值)为 0.136λ;长波通镀制结束后,测得PV值为0.63λ。镀膜 前后的面形差异较大,因此需要对增透面进行应力补 偿。计算得分光膜系的应力为326 MPa,增透膜系的 应力为162 MPa,随后进行双面膜镀制实验。测得双 面膜分光面的 PV 值为0.182λ,增透面的 PV 值为

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膜应力、优化表面面形的目的。

0.228\,如图6所示。可见,工艺调整后达到了降低薄

图 6 基板、分光面和增透面的面形。(a)基板;(b)分光面;(c)增透面

Fig. 6 Surface shape of substrate, spectral surface and antireflection surface. (a) Substrate; (b) spectral surface; (c) antireflection surface

#### 5.2 光谱检测结果分析

使用 Cary 7000 分光光度计与 Spectrum Two 傅 里叶光谱仪分别对可见/近红外和中红外光谱进行 测试。依据所获得的薄膜材料的折射率与消光系 数设计分光膜的光谱,对设计光谱与实际光谱进行 比较后发现实际光谱整体右移,长波透过率偏低, 如图 7 所示。

将光谱数据导入 Macleod 进行拟合后发现, ZnS 的实际厚度是设计厚度的 1.08倍, YbF<sub>3</sub>的实际厚度是 设计厚度的 95%。调整膜层厚度比例后沉积双面膜, 其分光面透射光谱如图 8 所示,光谱整体上与设计光 谱拟合得较好, 无明显偏差。在单面镀膜条件下, 3.7~4.8 μm 波段的理论平均透过率为 79.5%, 实测平 均透过率为 77%。对实测光谱进行拟合,结果显示光 谱性能变差不是由膜层厚度失配导致的。分析后发



图7 理论设计光谱与实际光谱的对比

Fig. 7 Comparison of theoretically designed spectrum and actual spectrum

现,在调节应力匹配过程中增加离子源辅助沉积导致 ZnS与YbF<sub>3</sub>膜层的吸收增大,但仍满足技术指标。





Fig. 8 Designed and actual spectra comparison for spectral surface. (a) Visible/near-infrared spectra; (b) mid-infrared spectra

在多光谱ZnS基底上完成双面膜的制备后进行光 谱测试。当入射角为55°时,双面膜在0.6~0.9 μm波 段的反射率为90.77%,在3.7~4.8 μm波段的透过率 为91.15%,如图9所示,可以满足设计要求。样片实 物图如图10所示。

#### 5.3 根据使用要求对样品进行环境测试

附着力测试:将3mm宽的拉力胶带粘贴在膜层

表面,快速垂直将胶带撕下,重复5次,无脱膜现象 发生。

摩擦力测试:用无尘布紧裹住橡皮擦,加1kg砝码,在膜层表面沿同一轨迹重复摩擦25次,无擦痕出现。

高低温测试:将样片放入冷热对冲箱内,先在低 温-62℃±2℃下保持2h,而后在高温85℃±2℃下



图 9 双面膜样品的反射率和透射率。(a)0.6~0.9 µm 波段的反射率;(b)3.7~4.8 µm 波段的透过率

Fig. 9 Reflectivity and transmittance of double-sided film samples. (a) Reflectivity in band 0.6-0.9 μm; (b) transmittance in band 3.7-4.8 μm





图 10 样片实物图。(a)分光面;(b)增透面 Fig. 10 Sample physical photos. (a) Spectral surface; (b) antireflection surface

保持2h,进行4个程式测试后观察,膜层没有明显 变化。

恒温恒湿测试:将样片放入湿热箱内,温度为 45℃±2℃,相对湿度为95%~100%,放置24h后取 出观察,膜层没有明显变化。

### 6 结 论

采用控制变量法对温度、沉积方法、离子源辅助沉 积参数进行调整,利用 Stoney 公式计算膜层应力,进 而对特定厚度的单层膜 ZnS、YbF<sub>3</sub>中的应力进行调 整,使单层膜 ZnS、YbF<sub>3</sub>的应力相匹配。利用增透面 的应力补偿分光面的应力,分光面的面形精度(PV值) 从 0.63λ降到 0.182λ,解决了分光镜表面面形差的问 题。通过对薄膜沉积结果进行逆向反演分析,调整膜 层厚度,提升了薄膜在 55°入射时的反射率和透射率, 其在 0.6~0.9 μm 波段的平均反射率为 90.77%,在 3.7~4.8 μm 波段的平均透过率为 91.15%。

制备的薄膜基本满足系统元件的使用需求,但短 波反射率和长波透过率仍有提升空间,在后续研究中 将以提高光谱性能为主要工作。

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## Fabrication of Large-Angle Dual-Band Detection Imaging Splitter

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#### Abstract

**Objective** There are significant differences in the actual optical characteristics of specific targets in different spectral bands. Thus, the two imaging systems shared an aperture for detection and identification. The internal light was separately imaged after passing through an optical device that separated the visible near-infrared and infrared light. In this way, the light from the two bands could be used to synchronously detect and image the target, allowing it to track the target more effectively. Thus, it could meet the all-day, wide-range, and high-resolution detection requirements. There have been few research reports on visible near-infrared and mid-infrared dual-band large-angle spectroscopy. Therefore, this research has an important reference value for dual-band infrared detection and imaging technology.

**Methods** The width and reflectance of the cut-off band at large angles were studied using the evaporation coating method. After practical calculation, it was concluded that at least four groups of film stacks should be used as the initial film system. A spectral curve that met the requirements could be obtained based on this optimization of the initial film system. The stress of ZnS film is generally compressive, while the stress of YbF<sub>3</sub> film is generally tensile. The control variable method was used to optimize the deposition process and adjust the corresponding thickness to change the stress of the ZnS and YbF<sub>3</sub> single-layer films. Because the film system was formed by depositing alternate layers of high and low refractive index materials, the tensile stress of the high and low refractive index materials with corresponding thicknesses were closer. The film system structure was analyzed based on the measured stress results, which showed that the two sides of the film could offset each other and tended to be smaller in the design.

**Results and Discussions** Based on a study of the characteristics of the high and low refractive index materials, the deposition process parameters of the film material (Table 2) were selected as the auxiliary parameters of the ion source (Table 3). Before coating, a constant temperature was 1 h. Thereafter, the ion source was used to clean the substrate for 15 min. A constant temperature of 200 °C was maintained for 2 h after coating, then the temperature was naturally cool to 90 °C for venting. The stress changes before and after coating are listed in Table 4. It can be seen from the analysis of the parameters before and after the process adjustment that the stress of the single-layer ZnS did not match that of the single-layer YbF<sub>3</sub>. The ZnS and YbF<sub>3</sub> deposition rates and auxiliary process parameters of the ion source were adjusted. The ZnS deposition rate was adjusted from 2 nm/s to 1.5 nm/s, and the deposition parameters of the YbF<sub>3</sub> ion source was adjusted from 200 V and 5 A to 220 V and 5 A. After adjusting the process parameters, the stress of the single-layer ZnS was close to that of the single-layer YbF<sub>3</sub> (Table 4). The antireflective film coated on the reverse side

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presented a tensile stress state, which could compensate for the stress on the spectral surface. The antireflective film coated on the reverse side met the requirements (Fig. 6). The spectrum of the deposited film was obtained and analyzed (Fig. 7), and the spectral data were imported into the film system design software for fitting. It was found that the actual thickness of the ZnS was 1.08 times the design thickness, and the actual thickness of the YbF<sub>3</sub> was 0.95 times the design thickness. After adjusting the film thickness ratio, the transmission spectrum curve of the spectral surface after deposition (Fig. 8) showed that the overall spectrum was well fitted with the design without obvious deviation (transmittance of  $3.7-4.8 \,\mu\text{m}$  under the condition of the single-sided coating). The theoretical average transmittance in the band of  $3.7-4.8 \,\mu\text{m}$  was 79.5%, and the average measured transmittance was 77%. The measured spectrum was fitted, and the fitting results showed that the spectral performance was not caused by the mismatch of the film thickness. After analysis, it was found that the addition of ion source-assisted deposition in the adjustment of the stress matching resulted in an increase in the absorption of the ZnS and YbF<sub>3</sub> films but still met the technical specifications.

**Conclusions** In this study, the temperature, deposition method, and ion source parameters were adjusted using the control variable method. The film stress was calculated using the Stoney formula. Then, the stresses of specific-thickness single-layer ZnS and YbF<sub>3</sub> were adjusted to make these stresses match. When the stress of the antiantireflection surface was used to compensate for the stress of the spectral surface, the surface accuracy (PV value) of the spectral surface decreased from  $0.63\lambda$  to  $0.182\lambda$ . Then, the problem of the poor surface shape of the spectroscope could be solved. Based on a reverse analysis of the film deposition results, the film thickness was adjusted to improve the reflectivity and transmittance at an incidence of 55°. In the  $0.6-0.9 \,\mu$ m wave band, the average reflectivity was 90.77%. In the  $3.7-4.8 \,\mu$ m wave band, the average transmittance was 91.15%. The prepared film basically met the use requirements of the system components.

Key words optical devices; spectroscope; low light level night vision; medium wave infrared imaging; surface flatness