

中国激光

亚波长双曲超材料板表面反射光 Goos-Hänchen 位移

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摘要 研究了亚波长双曲超材料板表面反射光 Goos-Hänchen (G-H) 位移增大、方向转换以及临界波长调制方法。基于等效介质理论和稳态相位方法系统计算了入射波长、填充因子以及背景介电常数等对 G-H 位移特性的影响。研究表明亚波长双曲超材料板不仅可以明显增大反射光的 G-H 位移, 而且可通过改变入射波长实现 G-H 位移方向的转换。在正负 G-H 位移之间存在临界波长, 入射波长小于此临界波长时 G-H 位移为正, 反之为负; 入射波长越接近该临界波长, G-H 位移越大, 反之越小。研究还发现临界波长可以由双曲超材料板的填充因子和背景介电常数调制, 增大填充因子和背景介电常数可以使临界波长发生蓝移, 反之发生红移。以上研究表明基于亚波长双曲超材料板表面的 G-H 位移在光隔离器、光学传感器以及集成光电器件应用中具有广阔的应用前景。

关键词 表面光学; 亚波长结构; 双曲超材料; Goos-Hänchen 位移

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1 引言

1947 年, 德国科学家 Goos 和 Hänchen^[1] 在实验室发现当有限尺寸的入射光束在两介质界面上发生全内反射时, 实际反射点相对于入射点会沿入射面发生一段横向偏移, 这一偏移即所谓的 Goos-Hänchen(G-H) 位移。此后, 人们对 G-H 位移的物理本质和实际应用展开了大量研究^[2-4]。有限宽度的入射光束可以等效成一系列不同方向的平面波, 这些平面波在两介质界面上发生不同的反射, 观察到的结果是所有反射光的叠加效果^[5]。所有反射光强度和相位上的差异会使入射光束在横向发生一定偏移, 即产生 G-H 位移。G-H 位移在光隔离、光学传感以及集成光学等领域表现出很大的潜在应用价值^[5-6]。在一般材料的交界面, G-H 位移很小, 一般仅为波长的几倍, 这不利于观察测量和实际应用。研究人员提出了包括稳态相位、能量传输以及平面波光线膨胀等模型来计算反射光的 G-H 位移。实验室中通常采用位置敏感探测器、弱测量方法以及干涉测量法观测 G-H 位移^[7]。近年来, 人们在对超材料表面 G-H 位移的研究中发现了很多新奇的结果。例如, Kang

等^[8] 发现运用石墨烯基超材料可实现 G-H 位移的有效放大和调节。Wong 等^[9] 在光子晶体系统中实现了正负 G-H 位移的增大。Asiri 等^[10] 则指出可用外加泵浦驱动场来控制 G-H 位移。但这些研究大多集中在红外波段, G-H 位移约为入射波长的几十倍, 无法用于可见光波段的器件设计。

双曲超材料(HMM)是一类高度各向异性的单轴材料, 因其色散曲面呈现双曲面形状而得名^[11]。非磁性介质 HMM 的介电系数为二阶张量形式。当介电系数垂直分量为正、平行分量为负时, 其色散曲面为双叶型双曲面, 称为 I 型 HMM。当介电系数的垂直分量为负、平行分量为正时, 其色散曲面为单叶型双曲面, 称为 II 型 HMM^[12]。HMM 在光场局域、增强自发辐射、亚波长成像等领域都有重要应用^[13-19]。自然界中存在的 HMM 非常少, 而且天然材料呈现双曲色散特性的频率多在远红外和太赫兹波段, 无法用于可见光波段的器件设计。而由人工设计的亚波长电磁周期性结构则可以实现光波段的 HMM, 但要满足材料的光轴方向和垂直光轴方向的介电函数符号相反^[20-23]。目前, 比较典型的人工 HMM 设计方法主要包括亚波长厚度的金属和电介质堆叠的多层结构和

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嵌入电介质内部的金属纳米线阵列。多层 HMM 的设计取决于目标光谱范围、损耗和阻抗匹配。为了实现阻抗匹配,电介质和金属的介电常数不能相差超过一个数量级。金属银(Ag)在紫外区和可见光区域的损耗有限,此外,它还可以与弱吸收介质(二氧化钛或氧化铝)的介电常数相匹配,因此本文所选取的多层结构由金属银和二氧化钛构成^[24-26]。

本文根据等效介质理论和稳态相位方法,对由金属银和二氧化钛构成的亚波长 HMM 板表面反射光束的 G-H 位移特性进行了系统的研究。详细分析了亚波长 HMM 板表面反射光束的 G-H 位移的大小和方向特性,分别研究了入射波长、填充因子以及背景介电系数对系统 G-H 位移的影响。研究结果表明亚波长 HMM 板表面反射光束的 G-H 位移约为传统材料 G-H 位移的 100 倍,此外,还可通过改变入射波长实现 G-H 位移方向的转换。正负 G-H 位移之间存在临界波长,入射波长小于临界波长时 G-H 位移为正,反之为负。入射波长越接近该临界波长,G-H 位移越大,反之越小。研究还发现增加填充因子和背景介电常数可以使临界波长发生蓝移,反之发生红移。

2 亚波长 HMM 板等效介质理论及其表面 G-H 位移

本文采用金属银和二氧化钛(TiO_2)薄膜交替排列构成亚波长 HMM 结构,如图 1 所示。HMM 是具有强各向异性介电常数张量或磁导率张量的介质,该张量的一个主分量和其余两个主分量符号相反,具体形式为

$$\hat{\boldsymbol{\epsilon}} = \begin{bmatrix} \epsilon_{xx} & 0 & 0 \\ 0 & \epsilon_{yy} & 0 \\ 0 & 0 & \epsilon_{zz} \end{bmatrix}, \quad \hat{\boldsymbol{\mu}} = \begin{bmatrix} \mu_{xx} & 0 & 0 \\ 0 & \mu_{yy} & 0 \\ 0 & 0 & \mu_{zz} \end{bmatrix}, \quad (1)$$

式中: $\hat{\boldsymbol{\epsilon}}$ 和 $\hat{\boldsymbol{\mu}}$ 分别为介电常数张量和磁导率张量; ϵ_{xx} 、 ϵ_{yy} 、 ϵ_{zz} 分别为 x 、 y 、 z 方向的介电常数主分量; μ_{xx} 、 μ_{yy} 、 μ_{zz} 分别为 x 、 y 、 z 方向的磁导率主分量。面内分量 $\epsilon_{xx} = \epsilon_{yy} = \epsilon_{\perp}$, 面外分量 $\epsilon_{zz} = \epsilon_{\parallel}$, 下标 \perp 和 \parallel 分别表示与光轴垂直和平行,本文仅考虑电 HMM,因此,磁导率张量为单位张量。根据等效介质理论^[27], 垂直和平行于各向异性轴的有效介电系数分量可表示为

$$\epsilon_{\parallel} = \frac{\epsilon_{Ag}\epsilon_{TiO_2}}{f\epsilon_{TiO_2} + (1-f)\epsilon_{Ag}}, \quad (2)$$

$$\epsilon_{\perp} = f\epsilon_{Ag} + (1-f)\epsilon_{TiO_2}, \quad (3)$$

式中: f 是金属银的填充因子; ϵ_{Ag} 和 ϵ_{TiO_2} 分别表示金属银和二氧化钛的介电系数。 ϵ_{Ag} 可由 Drude 模型给出^[28]:

$$\epsilon_{Ag} = \epsilon_{\infty} - \frac{\omega_p^2}{\omega(\omega + i\tau^{-1})}, \quad (4)$$

式中: ω_p 为等离子体频率, $\omega_p = 1.4 \times 10^{16}$ rad/s; $\epsilon_{\infty} = 3.7$; τ 为弛豫时间, $\tau = 0.45 \times 10^{-14}$ s。二氧化钛的介电系数可表示为^[29]

$$\epsilon_{TiO_2} = 5.913 + \frac{0.2441}{\lambda^2 - 0.0803}. \quad (5)$$

本文以横磁(TM)波为例,根据 Fresnel 公式得到其反射系数为

$$r_{TM} = \frac{r_I + r_{II} e^{2ik_{m,z}d}}{1 + r_I r_{II} e^{2ik_{m,z}d}}, \quad (6)$$

式中: $k_{m,z}$ 为材料板波矢量的垂直分量; d 为 HMM 板的厚度; r_I 和 r_{II} 分别为光波在界面 I 和界面 II 的反射系数,可分别表示为

$$r_I = \frac{k_{i,z}/\epsilon_i - k_{m,z}/\epsilon_{\perp}}{k_{i,z}/\epsilon_i + k_{m,z}/\epsilon_{\perp}}, \quad (7)$$

$$r_{II} = \frac{k_{m,z}/\epsilon_{\perp} - k_{t,z}/\epsilon_t}{k_{m,z}/\epsilon_{\perp} + k_{t,z}/\epsilon_t}, \quad (8)$$

式中: ϵ_i 、 ϵ_t 分别为入射区和透射区的介电系数; $k_{i,z}$ 、 $k_{t,z}$ 分别为入射区和透射区波矢量的垂直分量。首先将该结构置于空气中,此时 $\epsilon_i = \epsilon_t = 1$, $k_{i,z} = k_{t,z} = k_0 \cos \theta$, k_0 是光在真空中的波数, θ 为入射角。将 HMM 介电常数张量 $\hat{\boldsymbol{\epsilon}}$ 代入麦克斯韦方程,根据电场、磁场以及波矢量的关系,可得 HMM 中波矢的垂直分量 $k_{m,z}$ 为^[20]

$$k_{m,z} = \sqrt{\epsilon_{\perp} \left(k_0^2 - \frac{\beta^2}{\epsilon_{\parallel}} \right)}, \quad (9)$$

式中: $\beta = k_0 \sin \theta$ 为传播常数,即入射波矢量的平行分量。将(7)~(9)式代入(6)式可以求得 HMM 板的反射系数。根据稳态相位法^[30],材料板表面的 G-H 位移(图 1)可表示为

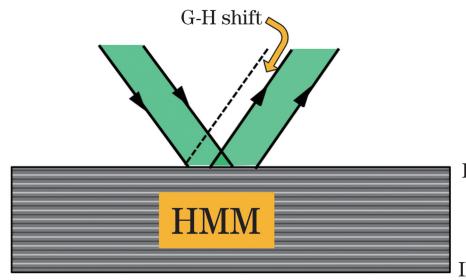


图 1 双曲超材料板表面反射光 G-H 位移示意图

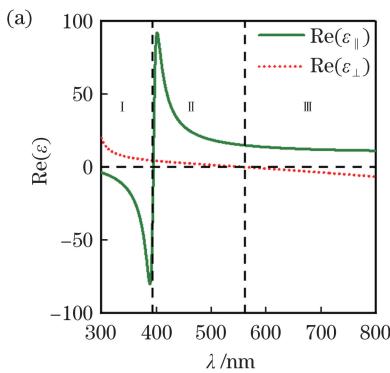
Fig. 1 Diagram of G-H shift of reflection light at surface of hyperbolic metamaterial slab

$$D = -\frac{\lambda}{2\pi} \frac{\partial \varphi}{\partial \theta}, \quad (10)$$

式中: $\varphi = -i \ln r_{TM}$ 表示反射相位, 可由反射系数给出, r_{TM} 为 TM 波的反射系数。

3 亚波长 HMM 板表面 G-H 位移特性分析

本节由稳态相位法计算亚波长 HMM 板表面的 G-H 位移, 并对其变化机制和影响因素加以讨论分析。HMM 板的厚度 $d = 120 \text{ nm}$, $f = 0.35$ 。首先, 根据(2)~(5)式, 计算了材料板等效介电系数实部和虚部随入射波长的变化, 如图 2 所示。从图 2(a)可以看出: 对于区域 I, $\epsilon_{\perp} > 0$ 且 $\epsilon_{\parallel} < 0$,



即当入射波长在 $300 \sim 395 \text{ nm}$ 之间时, 该结构表现为 I 型 HMM; 对于区域 II, $\epsilon_{\perp} > 0$ 且 $\epsilon_{\parallel} > 0$, 即当入射波长在 $395 \sim 551 \text{ nm}$ 之间时, 该结构表现为椭圆形 HMM; 对于区域 III, $\epsilon_{\perp} < 0$ 且 $\epsilon_{\parallel} > 0$, 即当入射波长在 $551 \sim 800 \text{ nm}$ 之间时, 该结构表现为 II 型 HMM。随着入射波长的增大, 该多层材料的色散曲面由双叶型双曲面转变为椭圆面再转变为单叶型双曲面, 这种转变是由金属银和二氧化钛的色散耦合导致的。图 2(b)表明: 对于 I 型 HMM, 介电系数的平行分量的损耗非常大, 且随波长呈近线性增加, 垂直分量的损耗则很小; 对于 II 型 HMM, 介电系数的平行和垂直分量的损耗都较小。

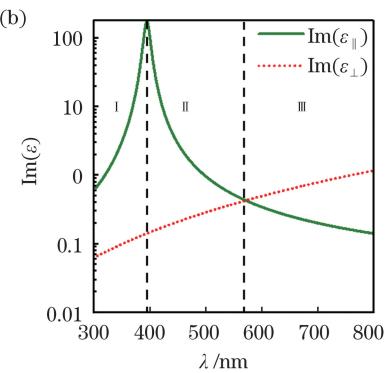


图 2 $f = 0.35$ 时材料板等效介电系数实部和虚部随入射波长的变化。(a) $\text{Re}(\epsilon_{\parallel})$ 和 $\text{Re}(\epsilon_{\perp})$; (b) $\text{Im}(\epsilon_{\parallel})$ 和 $\text{Im}(\epsilon_{\perp})$

Fig. 2 Variations of real part and imaginary part of material's effective permittivity with incident wavelength for $f = 0.35$. (a) $\text{Re}(\epsilon_{\parallel})$ and $\text{Re}(\epsilon_{\perp})$; (b) $\text{Im}(\epsilon_{\parallel})$ and $\text{Im}(\epsilon_{\perp})$

结合图 2 和(6)~(10)式, 首先计算了不同类型亚波长 HMM 板表面的 G-H 位移随入射角度的变化, 计算结果如图 3 所示。需要指出当 $f = 0$ 时, HMM 板变为传统的 TiO_2 材料板[图 3(a)], 当入射波长不同时 HMM 板的类型也不同[图 3(b)~(d)]。图 3(a)为入射波长 $\lambda = 360 \text{ nm}$ 的 TiO_2 材料板表面 G-H 位移随入射角度的变化。图 3(b)为入射波长 $\lambda = 360 \text{ nm}$ 的 I 型 HMM 板表面 G-H 位移随入射角度的变化。图 3(c)为入射波长 $\lambda = 430 \text{ nm}$ 的椭圆形 HMM 板表面 G-H 位移随入射角度的变化。图 3(d)为入射波长 $\lambda = 630 \text{ nm}$ 的 II 型 HMM 板表面 G-H 位移随入射角度的变化。图 3 表明, 在同样的入射参数条件下[图 3(a)和 3(b)], HMM 板表面 G-H 位移是传统材料的 100 倍, 这归因于 HMM 独特的色散特性。同样的结构参数条件下[图 3(b)~(d)], I 型 HMM 板表面的 G-H 位移比 II 型和椭圆形 HMM 板表面的 G-H 位移的增大更明显。因此, 后面的计算只关注 I 型 HMM 板

表面 G-H 位移特性。

计算了入射波长分别为 325 nm 和 350 nm 时 G-H 位移随入射角度的变化值, 计算结果如图 4(a)所示, 图中点线和实线分别对应入射波长为 325 nm 和 350 nm 。当波长为 325 nm 的入射光以 69.9° 入射时, 对应的 G-H 位移为 $+52.8 \mu\text{m}$ 。当波长为 350 nm 的入射光以 66.4° 入射时, 对应的 G-H 位移为 $-39.1 \mu\text{m}$ 。图 4(a)中计算结果表明: 对于 HMM 板, 仅改变入射波长即可获得方向相反的 G-H 位移, 这是其他单一材料系统所不能实现的^[31-32]。当 TM 波反射系数达到最小时反射相位发生突变, 而反射系数相位的突变是引起 G-H 位移的原因。图 4(b)给出了 HMM 板表面反射相位随入射角度的变化值, 图中点线和实线分别对应入射波长为 325 nm 和 350 nm 。可以看到图 4(b)的相位突变点和图 4(a)中的 G-H 位移峰是对应的, 且当相位突然变小时 G-H 位移为正, 反之为负。

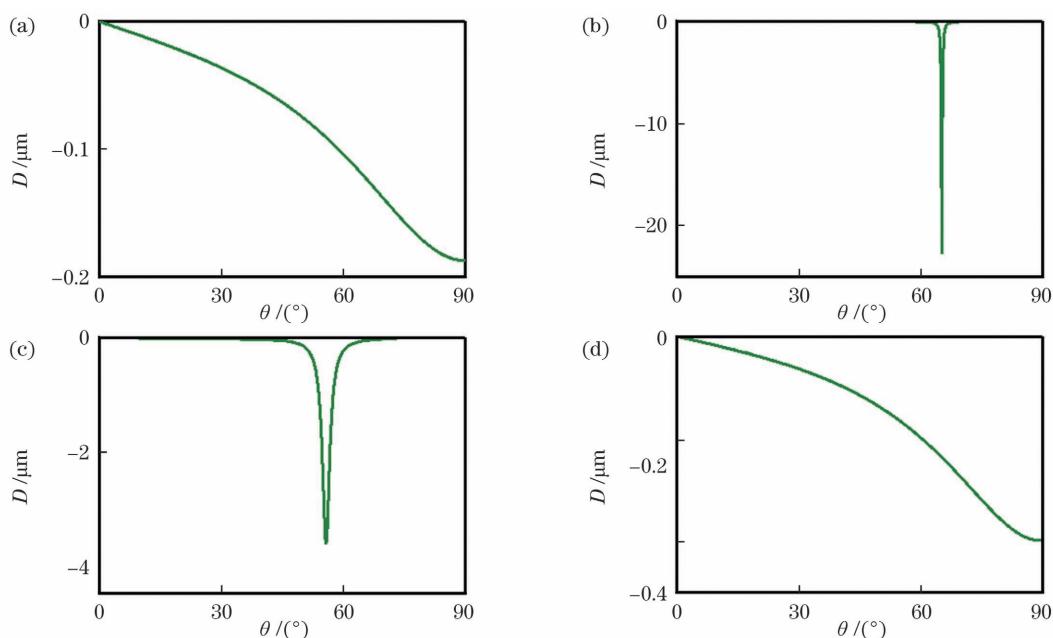


图3 不同类型HMM板G-H位移随入射角度的变化。(a) $\lambda=360\text{ nm}, \text{TiO}_2$ 材料板;(b) $\lambda=360\text{ nm}$, I型HMM板;
(c) $\lambda=430\text{ nm}$,椭圆形材料板;(d) $\lambda=630\text{ nm}$, II型HMM板

Fig. 3 Variation in G-H shift of different types of HMM slab with incident angle. (a) $\lambda = 360 \text{ nm}$, TiO_2 material slab; (b) $\lambda = 360 \text{ nm}$, HMM slab of type I; (c) $\lambda = 430 \text{ nm}$, material slab of elliptical shape; (d) $\lambda = 630 \text{ nm}$, HMM slab of type II

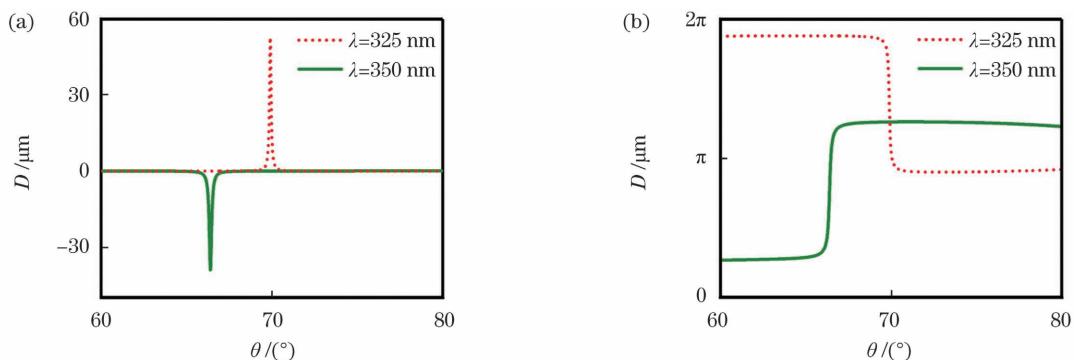


图4 HMM板表面G-H位移和反射相位随入射角度的变化。(a)HMM板表面G-H位移;(b)HMM板表面反射相位
Fig. 4 Variations of G-H shift and reflection phase of HMM slab with incident angle. (a) G-H shift of HMM slab;
(b) reflection phase of HMM slab

图5为HMM板表面最大G-H位移与入射波长的对应关系,计算结果表明入射波长从310 nm变化到360 nm的过程中,HMM板表面最大G-H位移先增大后减小,在此过程中G-H位移的偏移方向发生转换。此外,正负G-H位移间存在一个临界波长(约为335.5 nm),当入射波长小于临界波长时G-H位移为正,反之为负。入射波长越接近临界波长,G-H位移越大,且表现出一定的对称性。这与Yin等发现的最佳金属层厚度现象类似,该结果是由共振模中的内在损耗与辐射损耗关系所决定的,当共振模中的内在损耗大于辐射损耗时,G-H位移为负,反之为正,且二者之差越小,G-H位移越大^[33-34]。

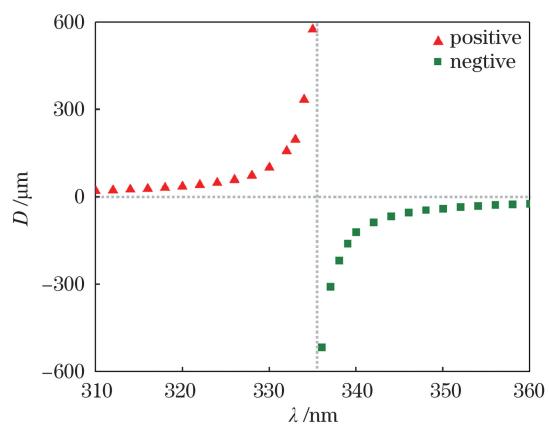


图5 入射波长对HMM板表面G-H位移的影响
Fig. 5 Effect of incident wavelength on G-H shift of HMM slab

为了进一步揭示HMM板表面G-H位移特性,计算了金属银填充因子 f 和背景介质介电常数 ϵ 对G-H位移大小和位置的影响,计算结果如图6所示。图6(a)和(b)给出了入射波长分别为325 nm和350 nm时HMM板金属银的填充因子 f 对G-H位移的影响,显然填充因子 f 是通过影响HMM介电常数系数间接影响G-H位移的。从图6(a)的计算结果可以看出,对于波长为325 nm的入射波(小于临界波长),随着 f 的增加,正向的G-H位移不断增加,但当填充因子 $f=0.55$,G-H位移的偏移方向发生改变,且随着 f 的增加,负向的G-H位移不断减小。对于波长为350 nm(大于临界波长)的入射波[图6(b)],随着 f 的增加,负向的G-H位移则不断减小。同样地,在图6(c)和

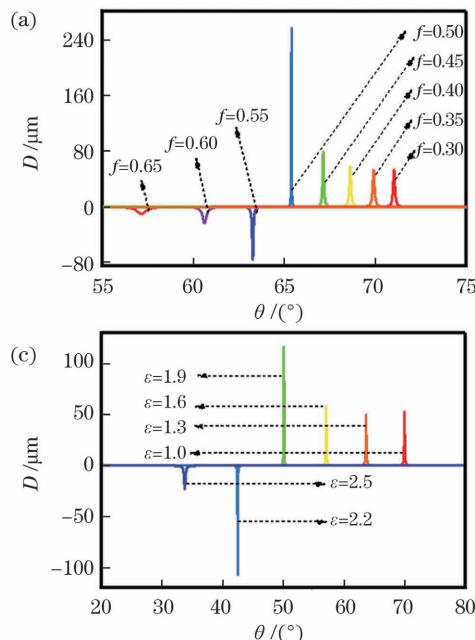


图6 金属银填充因子 f 和背景介质介电常数 ϵ 对HMM板G-H位移的影响。(a)(c) $\lambda=325$ nm; (b)(d) $\lambda=350$ nm

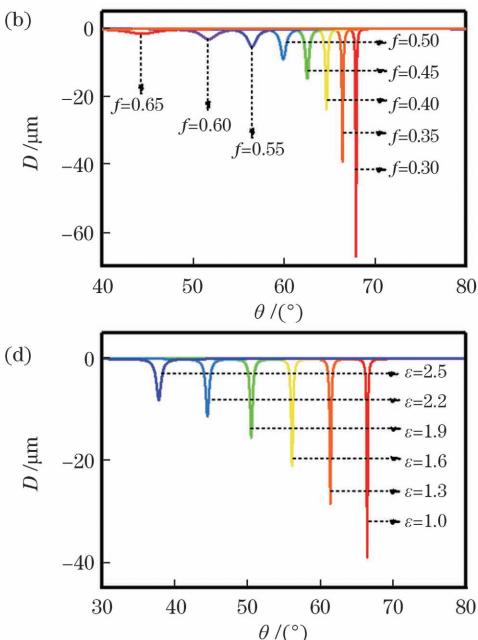
Fig. 6 Effects of filling factor and background permittivity ϵ on G-H shift of HMM slab. (a)(c) $\lambda=325$ nm;

(b)(d) $\lambda=350$ nm

4 结 论

基于等效介质理论和稳态相位方法,研究了亚波长HMM板表面反射光G-H位移的大小和方向特性。研究结果表明:与传统材料相比,HMM板表面G-H位移得到显著增大,并且仅通过改变入射波长即可实现G-H位移方向的转换。正负G-H位移之间存在临界波长,当入射波长小于此临界波长时,G-H位移为正,反之为负。入射波长越接近临界波长,G-H位移越大,正负G-H位移关于临界波长表现出一定的对称性。此外还发现利用HMM的填

(d)中可以看出入射波长分别为325 nm和350 nm时HMM板背景介质介电常数 ϵ 对G-H位移的影响。对于波长为325 nm的入射波[图6(c)],随着 ϵ 的增加,正向的G-H位移不断增加,但当介电常数 $\epsilon=2.2$,G-H位移的偏移方向发生改变,且随着 ϵ 的增加,负向的G-H位移不断减小。对于波长为350 nm的入射波[图6(d)],随着 ϵ 的增加,负向的G-H位移不断减小。结合图4可得增加填充因子和背景介质介电常数实际上等效于临界波长蓝移。因此,当入射波长小于临界波长时G-H位移方向发生改变,而当入射波长大于临界波长时则G-H位移方向不变。对临界波长的调节将为光隔离器、光传感器等新型光学器件的设计提供帮助。



充因子和背景介质的介电常数可以调节临界波长位置,随着填充因子和背景介质的介电常数的增加,临界波长发生蓝移。以上结论对基于G-H位移效应的光隔离器、光学传感器、集成光电器件等的设计有一定的参考价值。

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Goos-Hänchen Shift of Reflected Light at Surface of Subwavelength Hyperbolic Metamaterial Slab

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Abstract

Objective Goos and Hänchen discovered in 1947 that when an incident beam of finite size undergoes total internal reflection on the interface of two media, the actual reflection point shifts laterally along the incident plane relative to the incident point, and the shift is known as the Goos-Hänchen (G-H) shift. A finite-width incident beam can be compared to a series of plane waves travelling in different directions. These plane waves have different reflections at the interface of the two media. After the superposition of the differences in intensity and phase of all reflected light, the incident beam shifts to a certain extent in the transverse direction. G-H shift has great potential applications in optical isolation, optical sensing, and integrated optics. However, in general, the G-H shift at the material interface is very small, which is only a few times the wavelength. Therefore, it is not conducive to observation, measurement, and practical application. Hyperbolic metamaterials (HMMs) are a type of highly anisotropic uniaxial material named after their hyperbolic dispersion relations. HMMs have a wide range of applications, including light field localization, enhanced spontaneous emission, and subwavelength imaging. In this paper, we present the enhancement, direction transformation, and critical wavelength modulation of the G-H shift on the surface of the subwavelength HMM slab.

Methods Researchers proposed models including steady-state phase, energy transfer, and plane wave linear expansion functions to calculate the G-H shift of reflected light. In the laboratory, position-sensitive detectors, weak measurement, and interferometry are usually used to observe the properties of G-H shift. For nonmagnetic media,

the dielectric coefficient of HMM is in the form of a second-order tensor. When the vertical component of the dielectric coefficient is positive and the parallel component is negative, its dispersion surface is a hyperboloid of bilobate type, which is called I-type HMM. When the vertical component of a dielectric coefficient is negative and the parallel component is positive, the dispersion surface is a hyperboloid of univalent type, which is called II-type HMM. Currently, the most common artificial HMMs include a multilayer structure of metal and dielectric stacked with subwavelength thickness and a metal nanowire array embedded in a dielectric. The multilayer HMM design is determined by the target spectral range, loss, and impedance matching. We investigate the effects of incident wavelength, filling factor, and background permittivities on the properties of G-H shift using the effective medium theory and the stationary-phase method.

Results and Discussions First, we calculated the real and imaginary parts of the equivalent permittivity of the material slab with the incident wavelength (Fig. 2) and determine the types of HMM materials at different incident wavelengths. Then, the variation of the G-H shift of different types of the subwavelength HMM slab with incident angle is calculated (Fig. 3). Figs. 3(a)–(d) present the bulk material, I-type HMM, elliptical HMM, and II-type HMM, respectively. Fig. 3 shows that the G-H shift of the HMM slab is 100 times that of bulk material under the same incident parameters. In addition, under the same structural parameters, the G-H shift of I-type HMM is more significant than that of II-type and elliptical HMM slabs.

Second, we calculated the G-H shift with the increasing incident angle when the incident wavelengths are 325 and 350 nm, respectively [Fig. 4(a)], which shows that the G-H shift in the opposite direction can be obtained only by changing the incident wavelength. Fig. 4(b) shows the variation of the reflection phase of the surface of the HMM slab with the incident angle. The phase mutation point corresponds to the G-H shift peak, and when the phase suddenly decreases, the G-H shift is positive, and otherwise, it is negative. Moreover, the relationship between the maximum G-H shift and incident wavelength is calculated (Fig. 5). We discovered that a critical wavelength exists between the positive and negative G-H shifts. The G-H shift is positive when the incident wavelength is less than the critical wavelength, and otherwise, it is negative. The greater the G-H shift is, the closer the incident wavelength is to the critical wavelength, indicating symmetry.

Finally, we studied the effects of the silver filling factor and background dielectric constant on the G-H shift characteristics (Fig. 6). We found that increasing the filling factor and the dielectric constant of the background medium is equivalent to the blue shift of the critical wavelength. When the incident wavelength is less than the critical wavelength, the direction of the G-H shift will change. When the incident wavelength is greater than the critical wavelength, the direction of the G-H shift remain unchanged.

Conclusions In the present study, the intensity and direction characteristics of G-H shift of the subwavelength HMM slab have been revealed. It shows that except enhancing the value of the G-H shift, the HMM slab can achieve the direct transformation of the G-H shift under different incident wavelengths. A critical wavelength exists between the positive and negative G-H shifts. The G-H shift is positive (negative) when the incident wavelength is less (greater) than the critical wavelength. At the same time, the closer the incident wavelength is to the critical wavelength, the larger the G-H shift is. We also discovered that the filling factor and background permittivities can be used to tune the critical wavelength. The critical wavelength presents blue-shift as the filling factor or background permittivities increases (decreases) (red-shifts). We believe that the G-H shift at the surface of the subwavelength HMM slab is very promising for potential applications in novel all-optical isolators, optical sensing, and integrated optoelectronic devices considering these intriguing discoveries.

Key words optics at surfaces; subwavelength structure; hyperbolic metamaterials; Goos-Hänchen shift

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