# **Optical bistability in a silicon nitride microring integrated with 2D PtSe<sub>2</sub>**

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Optical bistability can be used to explore key components of all-optical information processing systems, such as optical switches and optical random memories. The hybrid integration of emerged two-dimensional layered PtSe<sub>2</sub> with waveguides is promising for the applications. We demonstrated the optical bistability in the PtSe<sub>2</sub>-on-silicon nitride microring resonator induced by a thermo-optic effect. The fabricated device has a resonance-increasing rate of 6.8 pm/mW with increasing optical power. We also established a theoretical model to explain the observation and analyze the device's performance. The study is expected to provide a new scheme for realizing all-optical logic devices in next-generation information processing systems.

Keywords: Optical bistability; waveguide-integrated PtSe2 device; nonlinear optics

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# 1. Introduction

The integration of two-dimensional (2D) materials, such as graphene and transition metal dichalcogenides (TMDCs), with photonic integrated circuits has been emerging for applications in optical modulation <sup>[1]</sup>, detection <sup>[2,3]</sup>, and nonlinear signal processing <sup>[4,5]</sup>, owing to the unique optical and electronic properties [6,7] of 2D materials, the giant interaction between 2D materials and light in waveguide devices [8,9], as well as, the CMOS-compatible device fabrication processes. PtSe<sub>2</sub> is a newly discovered group-ten TMDC with a tunable bandgap, carrier mobility of more than 16300 cm<sup>2</sup>V<sup>-1</sup>s<sup>-1 [10]</sup>, and stable physical and chemical properties<sup>[11,12]</sup>, which makes  $PtSe_2$  promising for various optoelectronic applications <sup>[7,12]</sup>. To date, waveguideintegrated PtSe<sub>2</sub> devices have been explored for optoelectronic applications of light modulation <sup>[13]</sup> and photodetection <sup>[14,15]</sup>.

On the other hand, optical bistability can be used to construct all-optical switches, optical memories, and alloptical logic gates, which are believed to break through the information bottleneck of exchange in future communication and computing systems, playing a crucial role as the core of the next-generation communication and signal processing systems <sup>[16]</sup>. Optical bistability was realized in silicon waveguide devices at the beginning of this century <sup>[17]</sup>. To further enhance the on-chip optical bistability, graphene-on-waveguide structures have been applied for developing bistable devices. For example, by integrating graphene on the waveguide-integrated Fabry-Perot resonator <sup>[18]</sup> and microring resonator (MRR) <sup>[19]</sup>, the photo-induced Joule heating in graphene can lead to enhanced effective thermal nonlinear index compared with the bare silicon waveguides. Besides, optical bistability with low input power requirement induced by the Kerr effect in graphene has been theoretically studied in the graphene-onsilicon slot microring resonator <sup>[20]</sup>. However, the optical bistability has seldom been reported based on waveguide-integrated  $PtSe_2$  devices.

In this paper, we studied an integrated nonlinear optical device based on 2D few-layer PtSe<sub>2</sub>-on-silicon nitride MRR. We measured the transmission spectra of the silicon nitride MRRs before and after covering the PtSe<sub>2</sub> film and calculated the absorption coefficients of the fabricated devices. We selected a representative device to measure its transmission curve while gradually increasing the optical power. The resonance shifts to longer wavelengths linearly with the input power, and a clear optical bistability effect was observed. Besides, we used the time-domain coupled mode theory (CMT) to simulate the thermo-optic effect in the device, which agrees well with the experimental results. The subsequent finite element method (FEM) simulation also shows that the integration of PtSe<sub>2</sub> greatly enhances the photo-thermal conversion in the waveguide device. The study is expected to provide a useful reference for developing subsequent nonlinear devices for all-optical signal processing applications.



**FIG. 1.** Schematic of the PtSe<sub>2</sub>-on-silicon nitride MRR. (a) Three-dimensional view of the device. (b) Top view of the device. (c) Cross-section view of the waveguide.

# 2. Results and discussion

We designed and fabricated the PtSe<sub>2</sub>-on-silicon nitride MRR based on the standard nanofabrication technology and home-developed wet transfer method. We first fabricated silicon nitride MRRs by using electron beam lithography and ICP-DRIE. The low-pressure chemical vapor deposited (CVD) top silicon nitride layer is 720 nm thick (H1), the buried oxide (BOX) is 4 µm thick (H3), and the etching depth of the silicon nitride layer is 400 nm (H2), as shown in Fig. 1. The gap between the bus waveguide and the microring is 390 nm (g). The width of the waveguide (W) and the diameter (D) of the MRR are 1.2 µm and 200 µm. A pair of focusing one-dimensional grating couplers is used for the fiber-to-waveguide optical coupling. Next, we covered the MRR with commercial CVD-grown five-layer PtSe<sub>2</sub> film on the sapphire substrate (Six Carbon Technology Shenzhen). To transfer the material, a layer of polymethyl methacrylate (PMMA) was spin-coated on the surface of the PtSe<sub>2</sub> sample, and the PMMA layer was heated and cured to form a solid protection layer. Then the sample was immersed in the KOH solution to separate the PMMA-on-PtSe<sub>2</sub> layer from the sapphire substrate. After cleaning in the deionized water, the PMMA-on-PtSe<sub>2</sub> layer was transferred onto the silicon nitride chip. Then the chip was drying in the air. Finally, the PMMA was removed with an acetone solution. The flowchart for the fabrication process is shown in Fig. S1 in Supplementary Material.



**FIG. 2.** Characterization of the five-layer  $PtSe_2$  on the surface of the silicon nitride chip. (a) The AFM characterization of the  $PtSe_2$  film. (b) The Raman spectrum of the  $PtSe_2$  film measured by using a Raman spectrometer with a pump wavelength of 785 nm. (c) The measurement result of the XPS full spectrum. (d) and (e) are the Gaussian fitting curves of the Pt 4f peak and the Se 3d peak in (c).

Next, we characterized the five-layer  $PtSe_2$  film on the surface of the silicon nitride chip. The thickness of the  $PtSe_2$  film was measured to be ~2.5 nm by using atomic force microscopy (AFM), as shown in Fig. 2(a). Fig. 2(b) shows the Raman spectrum of the  $PtSe_2$  film on the surface of the silicon nitride chip. Three major Raman peaks appeared at  $177 \text{cm}^{-1}$ ,  $206 \text{cm}^{-1}$ , and  $225 \text{cm}^{-1}$ , related to the  $E_g$ ,  $A_{1g}$ , and LO vibration modes of Se atoms respectively, and the locations

of Raman peaks were consistent with the previous report <sup>[21]</sup>. X-ray photoelectron spectroscopy (XPS) analysis was also performed to characterize the PtSe<sub>2</sub> film further. As shown in Fig. 2(c), C, O, Pt, and Se peaks mainly existed in the film. The peaks of  $C_{1s}$  and  $O_{1s}$  come from the air absorbed by the surface of the PtSe<sub>2</sub> film. Two peaks of 72.72eV and 76.02eV corresponding to Pt 4f<sub>7/2</sub> and Pt 4f<sub>5/2</sub> were obtained by Gaussian fitting of the Pt 4f peak, as shown in Fig. 2(d). The fitting of the Se 3d peak in Fig. 2(e) also results in two main peaks located at 54.02eV and 54.88eV, corresponding to Se  $3d_{5/2}$  and Se  $3d_{3/2}$ . The atomic numbers of Pt and Se were calculated based on the measured spectra, as shown in the inset table of Fig. 2(c), which was consistent with the theoretical values <sup>[21]</sup>.



**FIG. 3.** Characterization of the PtSe<sub>2</sub>-on-silicon nitride MRRs. (a) and (d) are the SEM images of two MRRs with the PtSe<sub>2</sub> film coverage, with estimated lengths of the material covered of 125  $\mu$ m and 471  $\mu$ m. (b) and (c) are the transmission spectra of the MRR in (a) before and after the PtSe<sub>2</sub> film transfer. (e) and (f) are the transmission spectra of the MRR in (d) before and after the PtSe<sub>2</sub> film transfer.

Next, we characterized the optical absorption of the PtSe<sub>2</sub>on-silicon nitride MRRs. The scanning electron microscopy (SEM) images of the devices are shown in Fig. 3(a) and (d). The wet transfer process resulted in PtSe<sub>2</sub> film breakages on the MRRs. The material-covered lengths on the MRRs were estimated as  $125 \,\mu\text{m}$  (Fig. 3(a)) and  $471 \,\mu\text{m}$  (Fig. 3(d)) from the contrast of the images. For the MRR in Fig. 3(a), from the Lorentz fittings of the fullwidth at half maximum (FWHM) of the measured resonance  $(\lambda_{res})$  spectra in Fig. 3(b) and (c), the Q factor is obtained as  $Q = \lambda_{res}$ / FWHM, with 12300 and 5800 before and after the  $PtSe_2$  film coverage. The transmission spectra of the MRR in Fig. 3(d) before and after the  $PtSe_2$  film integration are shown in Fig. 3(e) and (f). The quality (Q) factor decreases from 16000 to 2500 after the  $PtSe_2$  film integration. Combined with the estimated PtSe<sub>2</sub> film covered length, and the measured transmission spectra before and after the PtSe<sub>2</sub> film integration, the optical absorption coefficients of the PtSe<sub>2</sub>-on-silicon nitride waveguide were calculated [22] as 172dB/cm (Fig. 3(a)) and 171dB/cm (Fig. 3(d)). The resonances have redshifts after the PtSe<sub>2</sub> film integration due to the effective refractive index change induced by the PtSe<sub>2</sub> film. We also simulated the optical absorption coefficient of the PtSe<sub>2</sub>-on-silicon nitride waveguide to be ~179 dB/cm, by using the FEM simulator (COMSOL Multiphysics) with the complex dielectric constant of the 5-layer PtSe<sub>2</sub> in the previous study <sup>[21]</sup>, which was measured by using the spectroscopic ellipsometry method. The simulation result agrees well with the experimental estimation.

To measure the nonlinear property of the fabricated device, we set up an experimental system as shown in Fig. 4(a). We used a variable optical attenuator (VOA) to vary the input optical power, and the 1% output of the 1:99 optical coupler to monitor the input power with a power meter (PM1). The coupling efficiency of the grating coupler at 1550 nm wavelengths is ~-5.5 dB. The input power in the following experiment refers to the optical power coupled into the bus waveguide from the grating coupler. We chose the device shown in Fig. 3(a) for the optical-bistability exploration. We started by setting the output power of the erbium-doped fiber amplifier (EDFA) to a high value that the device can withstand, while the VOA attenuation was set to the maximum attenuation. In this way, we can start the measurement at a low input power, ensuring a stable and continuous tuning of the input power. The measured transmission spectra for different input powers are shown in Fig. 4(b). As the input power increases, the resonance shifts to longer wavelengths, with a linearly fitted slope of  $\sim 6.8$  pm/mW, as shown in Fig. 4(c). Besides, as the optical power increases, the resonance curve is no longer symmetrical, and eventually, there is a significant power jump on the longer wavelength side, which indicates the emergence of the bistable state.



**FIG. 4.** Optical nonlinearity measurement and simulation results. (a) Schematic of the experimental setup. (b) Measured transmission spectra under different input power levels. (c) The resonant wavelength at different input powers. (d) Simulated transmission spectra under different input powers using the time-domain CMT method. (e) Simulated resonant wavelengths at different input powers. (f)-(h) Hysteresis loop simulations for the input wavelengths

of 1541.55 nm, 1541.56 nm, and 1541.57 nm. Arrows in (f)-(h) indicate the directions of the input power variations.

To further evaluate the nonlinear effect in the PtSe<sub>2</sub>-onsilicon nitride MRR, we used the time-domain CMT <sup>[23,24]</sup> method to calculate the energy distribution  $(|a|^2)$  in the microring cavity with the input power in the bus waveguide  $(P_{in})$ , and the transmission spectrum of the MRR ( $T_{trans}$ ). We established the CMT model as follows <sup>[25,26]</sup>,

$$\frac{1}{Q_0} = \frac{1}{Q_{inst}} + \frac{1}{Q_{ext}} \tag{1}$$

$$\frac{2}{\tau_{ext(inst)}} = \frac{\omega'_0}{Q_{ext(inst)}}$$
(2)

$$2\pi c$$
 (p)

$$b_0 - \frac{1}{\lambda_0 + \Delta \lambda_{therm} + \Delta \lambda_{Kerr}}$$
(3)

$$\lambda_{therm} = \frac{2|a|^2 \tau_{linear}^{-1} \lambda_0 R \partial n}{n_0 \quad \partial T} \tag{4}$$

Δ

$$\Delta\lambda_{Kerr} = \frac{|a|^2 n_2 \lambda_0 c}{n_0 n_g V} \tag{5}$$

$$a|^{2} = \frac{2\tau_{ext}P_{in}}{\left(\omega - \omega_{0}'\right)^{2} + \left(\tau_{ext}^{-1} + \tau_{inst}^{-1} + \tau_{linear}^{-1}\right)^{2}}$$
(6)

$$T_{\rm trans} = \frac{\left(\omega - \omega_0'\right)^2 + \left(\tau_{ext}^{-1} + \tau_{linear}^{-1} - \tau_{inst}^{-1}\right)^2}{\left(\omega - \omega_0'\right)^2 + \left(\tau_{ext}^{-1} + \tau_{inst}^{-1} + \tau_{linear}^{-1}\right)^2}$$
(7)

where  $Q_0$  is the loaded Q factor of the MRR,  $\tau_{ext}$  and  $Q_{ext}$  are related to the intra-cavity mode reduction rate caused by the light coupling output,  $\tau_{inst}$  and  $Q_{inst}$  are related to the dissipation rate caused by the intrinsic loss in the microring cavity,  $\tau_{inst}$  can be regarded as the time that the mode amplitude attenuates to 1/e of the initial amplitude under the effect of the intrinsic loss in the microring cavity.  $\omega'_0$  is the resonant frequency, considering the thermo-optic effect  $(\Delta \lambda_{therm})$  and Kerr effect  $(\Delta \lambda_{Kerr})$ .  $\lambda_0$  is the resonance without nonlinear effect (i.e., at low input powers),  $\tau_{linear}$  is the amplitude loss rate caused by the linear optical loss, R is the thermal resistance of the MRR,  $n_0$  is the refractive index of the waveguide,  $\frac{\partial n}{\partial T}$  is the thermo-optic coefficient,  $n_2$  is the Kerr coefficient of silicon nitride, c is the speed of vacuum light,  $\omega$  is the angular frequency of the input light.  $n_g$  and V are the group index and the mode volume in the MRR cavity, which are calculated by the frequencydifference time-domain (FDTD) method (Lumerical FDTD). The values of the parameters are shown in Table 1. Specifically, parameters of  $Q_{ext}$ ,  $Q_{inst}$ ,  $\tau_{linear}$ , and R were obtained by fitting the measurement of the transmission spectrum of the PtSe<sub>2</sub>-on-silicon nitride MRR at the low input power of -15.5 dBm. The simulation results are shown in Fig. 4(d). The transmission spectrum becomes unsymmetrical at high input powers, with a resonance increasing slope of 6.85 pm/mW (shown in the linear fit in Fig. 4(e)), which is consistent with the measurement result, indicating that our model can well describe the nonlinear effects in the device. Besides, it can be known from Eqs. (4)

and (5) that the resonance shift due to the thermo-optic effect dominates, while the Kerr effect is almost negligible. We also calculated the power transmission of the device for different input wavelengths based on the time-domain CMT model. As shown in Fig. 4(f), with an input wavelength of 1541.55 nm, there is always only one solution to the equations with the change of the input power, which means that the MRR has only one steady state, which is the same to the shorter wavelengths. At the longer input wavelength of 1541.56 nm, a hysteresis loop appears and represents the appearance of bistability (Fig. 4(g)). With further increasing the input wavelength to 1541.57 nm, the hysteresis loop is extended and the gap between the two steady states increases (Fig. 4(h)). Besides, the stronger thermal-optic effect can lead to the redshift of the bistable threshold wavelength and decrease of the bistable threshold power while the reduction in width and area of the hysteresis loop. The device itself works like an optical static random-access memory. Decreasing or increasing the input optical power can be regarded as writing data, after which the device can hold the written data by setting the input power between the two edges of the hysteresis loop, and the data can be read by detecting the output of the device. We also measured the switching time of the bistable state to be  $\sim 30 \ \mu s$  by synchronously recording the input and output powers with the two power-sensor ports of the optical power meter (N7744A) while tuning the input power.

**Table 1.** Parameters and sources used in CMT simulation.

Parameter	Value	Source
$Q_0$	5800	[Measurement]
$Q_{ext}$	40324	[Measurement, CMT <sup>[27,28]</sup> ]
$Q_{inst}$	6774	[Measurement, CMT <sup>[27,28]</sup> ]
$ au_{linear}$ (s)	$1.6 \times 10^{-9}$	[Measurement, CMT <sup>[27,28]</sup> ]
R (K/mW)	105	[Measurement, CMT <sup>[27,28]</sup> ]
$n_0$	1.9965	[Reference <sup>[29]</sup> ]
$n_2 ({ m m^{2}/W})$	$2.6 \times 10^{-19}$	[Reference <sup>[30]</sup> ]
$n_g$	2.05	[FDTD simulation]
V (m <sup>3</sup> )	$8.4754 \times 10^{-16}$	[FDTD simulation]
<i>∂n/∂T</i> (1/K)	$2.51 \times 10^{-5}$	[Reference <sup>[31]</sup> ]
(1/K)	2.51×10 <sup>-5</sup>	[Reference <sup>[31]</sup> ]



**FIG. 5.** Electromagnetic thermal simulation results of the waveguide cross-section. (a) and (b) are temperature distributions of the PtSe<sub>2</sub>-on-silicon nitride waveguide cross-section and the bare waveguide cross-section.

Finally, we performed a simulation of the photothermal effect in the waveguide using the FEM simulator (COMSOL Multiphysics) with the electromagnetic heating module. We modeled the waveguide cross-section and simulated the cases with and without the  $PtSe_2$  film integration. In the

simulation, the electromagnetic loss was used as the heat source for heat transfer calculations and then the temperature distribution can be obtained. We gave the model an initial input light source of about 30 mW and set the ambient temperature to 293.15 K. The thermal conductivity of air, silicon nitride, PtSe2, and BOX were chosen as 0.026 m·K, 29 m·K, 51 m·K  $^{\rm [32]}\!,$  and 1.4 m·K. The conductivity of  $PtSe_2$  film was set to  $1 \times 10^{-5}$  S/m <sup>[33]</sup>. The simulation results are shown in Fig. 5. The average temperature of the waveguide region is about 366.2 K, which is a significant improvement compared with the case of bare silicon nitride waveguide with an average temperature of 297.9 K. This result also indicates that the application of PtSe<sub>2</sub> film can enhance the photothermal conversion in the waveguide. The results demonstrate the application and potential of PtSe<sub>2</sub>-based bistable devices in all-optical signal processing.

## 3. Conclusion

In summary, we have studied the optical bistability in the five-layer PtSe<sub>2</sub>-on-silicon nitride MRR. The absorption coefficient of the PtSe<sub>2</sub>-on-silicon nitride waveguide was obtained from the transmission spectra of the MRR with/without PtSe<sub>2</sub> film coverage at the low input optical power. With increasing the input power, the resonance of the PtSe2-on-silicon nitride MRR has redshifts, and the transmission spectrum becomes unsymmetrical, showing the optical bistability. We have also theoretically studied the nonlinear effects in the device through developing the time-domain CMT model, the results were consistent with the measurement results. Besides, the FEM simulation also shows that the addition of the five-layer PtSe<sub>2</sub> film significantly enhances the photothermal conversion in the waveguide device. The results obtained in this study provide some references to the combination of PtSe<sub>2</sub> with waveguide devices and its mechanism of action, which will help us to advance the application of TMDCs thin film materials in integrated photonics, as well as provide a new direction of exploration for integrated optoelectronic devices.

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